

# SYNTHESIS OF LOW-LOSS YTTRIUM IRON GARNET (Y<sub>3</sub>Fe<sub>5</sub>O<sub>12</sub>) VIA THE MICROEMULSION TECHNIQUE AND ITS SYNTHESIS VARIATION EFFECTS ON ITS PROPERTIES

**MASNI BINTI MANAP** 

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By

MASNI BINTI MANAP

Thesis Submitted to the School of Graduate Studies, Universiti Putra Malaysia, in Fulfillment of the Requirement for the Degree of Master of Science

July 2013

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# UPM

This thesis is dedicated to my late father Allahyarham Haji MANAP BIN ABDULLAH, my beloved mother Hajah KULSHAM BINTI PONIMIN, my brothers and sisters, MAHIDAN, MARIAH, MAHADIR, MARIYANTI, MARIZA, and MUHAMAD, in appreciation of their love and supports. Abstract of thesis presented to the Senate of Universiti Putra Malaysia in fulfillment of the requirement for the degree of Master of Science

## SYNTHESIS OF LOW-LOSS YTTRIUM IRON GARNET (Y<sub>3</sub>Fe<sub>5</sub>O<sub>12</sub>) VIA THE MICROEMULSION TECHNIQUE AND ITS SYNTHESIS VARIATION EFFECTS ON ITS PROPERTIES

By

## MASNI BINTI MANAP

#### July 2013

Chairman : Associate Professor Mansor Hashim, PhD Faculty : Science

Yttrium iron garnet,  $Y_3Fe_5O_{12}$  (YIG), is a material widely used in electronic devices for the microwave region from 300 MHz to 100 GHz range. Even though the technology involving ferrites is advancing, there is still a lack of understanding and systematic examinations on how losses in ferrites occur and how to control them especially at UHF or higher frequencies. In this work, YIG powders were prepared via the microemulsion technique and different approaches of bulk preparation were employed to attempt production of extremely low loss YIG. The loss of conventional-uniaxial sample, monodisperse-uniaxial sample and monodisperse-CIP sample was studied for samples sintered at relatively low temperature and up to fairly high temperature. YIG was produced by the microemulsion technique as the growth of the precursor was controlled by a water-in-oil emulsion. The aqueous solution consists of (Y(NO<sub>3</sub>)<sub>3</sub>·6 H<sub>2</sub>O) and (Fe(NO<sub>3</sub>) •9 H<sub>2</sub>O). Cetyltrimethyammonium Bromide (CTAB) was used as the surfactant and n-octane as the oil phase. Ammonium hydroxide solution which acts as a reducing agent was added to the aqueous solution to form precipitates. Then, the

precipitates were separated by centrifugation, washed with ethanol and then dried at 80 °C for 12 h. The dried precipitate was calcined at 600 °C for 2 h and ground into powder form. Three torroidal samples were prepared which is conventional-uniaxial sample, monodispersion-uniaxial sample, and monodisperse-CIP sample were sintered at different temperatures. The particle size was confirmed by Transmission Electron Microscopy (TEM), the thermal analysis was performed using a Thermal Thermogravimetric Analyzer (TGA), the phase was characterized using X-ray diffraction (XRD) and morphology was observed by Field Emission scanning electron microscopy (FeSEM). The permeability and rf energy loss of the samples was studied using an impedance material analyzer. The TEM results show that the particles are in the nanometer range with an average of 24 nm. The crystallization temperature of the sample can be deduced to be at 1145°C as observed from the TGA curve. The XRD results show that the full phase of YIG is formed at 1200°C. FeSEM micrographs and grain size distributions for the samples with different preparation techniques show the evolution of microstructure as the grain size increases with the increase of the sintering temperature. The micrographs clearly illustrate the evolution of the particle constituent to the formation of necks which lead to grains development over the sintering temperature range. The conventionally prepared and monodisperse sample show the correlation between magnetic loss, phase purity and grain size as it decrease with the decrease of grain size and phase purity. The monodisperse-CIP sample shows a different trend where it has a big grain size but low loss. We speculate that the monodisperse-CIP samples have significant numbers of pores that can act as pinning centers to the domain wall movements thus decreasing the magnetic loss of the sample. All the samples exhibit lower loss with tan  $\delta$  lower than  $10^{-1}$  comparable to previous research results.

Abstrak tesis yang dikemukakan kepada senat Universiti Putra Malaysia sebagai memenuhi keperluan untuk Ijazah Master Sains

# PENYEDIAAN YTTRIUM IRON GARNET (Y<sub>3</sub>Fe<sub>5</sub>O<sub>12</sub>) YANG MEMPUNYAI KEHILANGAN TENAGA YANG SANGAT SEDIKIT MELALUI TEKNIK MICROEMULSION DAN KESAN VARIASI PENYEDIAAN TERHADAP CIRI-CIRINYA

Oleh

## MASNI BINTI MANAP

#### Februari 2013

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Yttrium Iron Garnet, Y<sub>3</sub>Fe<sub>5</sub>O<sub>12</sub> (YIG), merupakan bahan yang digunakan secara meluas dalam alat-alat elektronik bagi rantau gelombang mikro daripada 300 MHz hingga 100GHz. Walaupun teknologi ferit semakin maju, masih terdapat kekurangan pemahaman dan pemeriksaan yang sistematik tentang bagaimana kehilangan tenaga dalam ferit berlaku dan bagaimana untuk mengawalnya terutama pada frekuensi yang sangat tinggi atau frekuensi yang lebih tinggi. Dalam kajian ini, serbuk YIG telah disediakan melalui teknik mikroemulsi dan pendekatan yang berbeza bagi teknik penyediaan pukal telah dilakukan dalam percubaan untuk menghasilkan kehilangan YIG amat rendah. Kehilangan tenaga bagi sampel proses penyediaan biasa-unipaksi, sampel monodisperse-unipaksi dan sample monodisperse-CIP dikaji pada suhu pensinteran yang agak rendah dan sehingga ke suhu pensinteran yang tinggi. YIG telah dihasilkan oleh teknik mikroemulsi dengan pertumbuhan prekursor telah dikawal oleh emulsi air dalam minyak. Larutan akueus terdiri Y(NO<sub>3</sub>)<sub>3</sub> dan Fe(NO<sub>3</sub>), Cetyltrimethyammonium bromida (CTAB) telah digunakan sebagai surfaktan dan n-oktana sebagai fasa minyak. Larutan Ammonium hidroksida yang bertindak sebagai agen penurunab ditambah kepada larutan akueus bagi membentuk mendakan. Kemudian, mendakan telah dipisahkan oleh sentrifugasi, dibasuh dengan etanol dan kemudian dikeringkan pada 80°C selama 12 jam. Mendakan kering telah dikalsin pada 600°C selama 2 jam dan dihancurkan kepada bentuk serbuk. Tiga sampel berbentuk torroid iaitu sampel proses. penyediaan biasa-unipaksi, sampel monodisperse-unipaksi dan sample monodisperse-CIP telah disinter pada suhu yang berbeza. Saiz partikel telah disahkan oleh mikroskop transmisi elektron (TEM), analisis terma telah dilakukan menggunakan analisia terma termogravimetri (TGA), fasa dicirikan menggunakan pembelauan sinar-X (XRD) dan morfologi telah diperhatikan oleh mikroskop imbasan elektron pelepasan medan (FeSEM ). Kebolehtelapan dan kehilangan tenaga rf sampel telah dikaji menggunakan penganalisis bahan impedans. Keputusan TEM menunjukkan bahawa partikel dalam julat nanometer dengan purata sebanyak 24 nm. Suhu pembentukan hablur boleh disimpulkan pada 1145°C seperti yang dilihat dari lengkung TGA. Keputusan XRD menunjukkan bahawa fasa penuh YIG telah terbentuk pada 1200°C. Mikrograf FeSEM dan taburan saiz butiran untuk sampel bagi semua teknik penyediaan menunjukkan evolusi mikrostruktur apabila saiz butiran meningkat dengan peningkatan suhu pensinteran. Mikrograf jelas menggambarkan evolusi butiran kepada pembentukan leher yang membawa kepada pembangunan butiran seiring dengan suhu pensinteran. Sampel penyediaan biasa dan monodisperse menunjukkan korelasi antara kehilangan tenaga, ketulenan fasa dan saiz butiran kerana ia berkurangan dengan penurunan saiz butiran dan ketulenan fasa. Sampel CIP menunjukkan tren yang berbeza di mana ia mempunyai saiz butiran yang besar tetapi mempunyai kehilangan tenaga yang rendah. Kami berspekulasi bahawa sampel CIP mempunyai bilangan liang yang signifikan yang boleh menyematkan pusat untuk pergerakan dinding domain sekali gus mengurangkan kehilangan tenaga sampel. Semua sampel mempamerkan kehilangan tenaga yang lebih rendah yang mempunyai tan  $\delta$  lebih rendah daripada  $10^{-1}$  jika dibandingkan dengan penyelidikan sebelumnya.



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# TABLE OF CONTENTS

			Page
DEDICATI	ON		ii
ABSTRACT	Γ		iii
ABSTRAK			V
ACKNOWI	ACKNOWLEDGEMENTS		
APPROVAL	L		X
DECLARA	ΓΙΟΝ		xii
TABLE OF	CON	ΓΕΝΤS	xiii
LIST OF TA	ABLES	8	XV.
LIST OF FI	GURE	28	XVI
LIST OF AI	BBRE	VIATIONS	XIX
CHAPTER			
1	INTI	RODUCTION	1
1	1 1 B	Background of the study	1
	1.1 E	errite for Microwaye Application	2
	1.3 P	roblem Statement	4
	1.4 C	Diectives	4
	1.5 T	Thesis Outline	5
			0
2	LITH	ERATURE REVIEW	6
	2.1	Introduction	6
	2.2	Low Loss Ferrites	6
	2.3	Synthesis Methods for YIG	10
		2.3.1 Solid State Reaction Route	10
		2.3.2 Sol-Gel Route	11
		2.3.3 High Energy Ball Milling	12
		2.3.4 Microemulsion	14
3	THE	ORY	15
	3.1	Introduction	15
	3.2	Fundamentals of Magnetization	15
	3.3	Type of Magnetism	17
	3.4	The Ferrimagnet Garnet Structure	19
	3.5	Loss Mechanisms	20
		3.5.1 Losses Due to The Oscillating Electric Field	20
		3.5.2 Losses Due to The Oscillating Magnetic Field	21
		3.5.2.1 Hysteresis Losses	23
		3.5.2.2 Eddy Current Losses	24
	2 -	3.5.2.1 Residual Losses	24
	3.6	Microemulsion Synthesis	25
		3.6.1 Properties of Microemulsion	25
		3.0.2 Microemuision Synthesis Technique	30

<b>4</b> I	ATERIALS A	AND METHODS	32
	1 Introduct	ion	32
	.2 Research	Design	34
	.3 Raw Mat	erials	34
	.4 Synthesis	s of Yttrium Iron Garnet	35
	.5 Toroid Pi	reparation	37
	.6 Cold Isos	static Pressing	39
	.7 Sintering		41
	.8 Materials	Characterization Measurement	42
	4.8.1 Stru	actural Measurement	42
	4.8.	1.1 X-ray Diffraction (XRD)	42
	4.8.	1.2 Thermogravimetric Analyzer (TGA)	44
	4.8.	1.3 Transmission Electron Microscope (TEM)	45
	4.8.	1.4 Field Emission Scanning Electron	47
		Microscope (FESEM)	
	4.8.	1.5 Materials Density Measurement	47
	4.8.2 Mag	gnetic Properties Measurement	49
	4.8	.2.1 Impedance/ material Analyzer	49
	.9 Error Est	imation	51
		DISCUSSION	50
5	ESULTS ANI	DISCUSSION	52
	I Introduct	ion	52
-	2 Microstru	icture-related Analysis	53
	5.2.1 Part	ticle Size Analysis	53
	5.2.2 The	ermal Analysis	54
	5.2.3 Pha	se Analysis	55
	5.2.4 Mo	Proportion A polysis	62 75
	5 Magnetic	ial Parmashility	75
	5.3.2 Los	s Factor	79
	5.3.2 Los	s Tangent	84
	5.5.5 203	s rangent	04
6 (	ONCLUSION	IS AND SUGGESTIONS	89
0	1 Conclusio	ons	89
	2 Suggestic	ons	90
REFERENCE			91
APPENDICES			96
BIODATA OF STUDENT			98
LIST OF PUB	ICATIONS		99

LI LICES	
APPENDICES	96
BIODATA OF STUDENT	98
IST OF PUBLICATIONS	99

# LIST OF TABLES

Table		Page
4.1	Estimated Error of measurements	52
5.1	Phase identification for conventional-uniaxial sample sintered at various temperatures. Note that $Fe_2O_3$ is Iron (III) Oxide, YFeO <sub>3</sub> is Yttrium Orthoferrite and $Y_3Fe_5O_{12}$ is YIG phase.	60
5.2	Phase identification for monodisperse-uniaxial sample sintered at various temperatures. Note that Fe <sub>2</sub> O <sub>3</sub> is Iron (III) Oxide, YFeO <sub>3</sub> is Yttrium Orthoferrite and Y <sub>3</sub> Fe <sub>5</sub> O <sub>12</sub> is YIG phase.	60
5.3	Phase identification for monodisperse-CIP sample sintered at various temperatures. Note that $Fe_2O_3$ is Iron (III) Oxide, YFeO <sub>3</sub> is Yttrium Orthoferrite and Y <sub>3</sub> Fe <sub>5</sub> O <sub>12</sub> is YIG phase.	61
5.4	Average grain size and density for conventional-uniaxial sample at different sintering temperatures.	73
5.5	Average grain size density for monodisperse-uniaxial sample at different sintering temperatures.	73
5.6	Average grain size and density for monodisperse-CIP sample at different sintering temperatures.	74
5.7	Initial permeability, $\mu$ ' at 10 MHz at room temperature for conventional-uniaxial sample, monodisperse-uniaxial sample and monodisperse-CIP sample.	79
5.8	Loss Factor, $\mu$ " at 10 MHz at room temperature for conventional-uniaxial sample, monodisperse-uniaxial sample, and monodisperse-CIP sample.	83

(C)

# LIST OF FIGURES

Figure		Page
3.1	Schematic diagram of different magnetic behaviors.	18
3.2	The garnet structure: $Fe^{3+}$ ( $\circ$ ) in tetragonal site at; $Fe^{3+}$ ( $\bullet$ ) in octahedral site at (0 $\frac{1}{4}\frac{3}{8}$ ) and $Y^{3+}$ ( $\circ$ ) in dodecahedral sites at ( $\frac{1}{4}\frac{1}{8}\frac{1}{2}$ ) and (0 $\frac{1}{4}\frac{5}{8}$ ).	19
3.3	Water-in-oil microemulsion (Reversed Micelle).	26
3.4	Oil-in-water microemulsion (Normal Micelle).	26
3.5	A linear surfactant molecule.	28
3.6	Schematic representation of decreased surface tension of two droplets as a result of repulsion between the surfactant molecules (Lopez-Quintela, 2003).	29
3.7	Mechanism synthesis of microemulsions.	31
4.1	Flowchart for sample preparation.	32
4.2	Mixture of oil-surfactant-co-surfactant (a) unstable cloudy solution (b) transparent stable solution of microemulsion	35
4.3	Synthesis route of microemulsion.	36
4.4	Toroidal shape.	38
4.5	Toroid mold.	38
4.6	Vacuumize torroidal sample.	39
4.7	Cold isostatic press set-up.	40
4.8	Schematic Diagram of the XRD (Cullity and Stock, 2001).	43
4.9	Schematic Principle of TGA Measurement.	44
4.10	An Optical Image Showing (a) TEM copper Grid covered with a Lacey Carbon Film, (b) A Lacey Carbon Film.	45

4.11	Schematic of Electrons from Primary Beam of TEM Interacting with Sample.	45
4.12	Density measurement by Archimedes principle/method.	48
4.13	The signal flow of permeability measurement.	49
5.1	Transmission electron microscopy image of as-prepared YIG.	53
5.2	TG curves of the as-prepared YIG sample by microemulsion.	54
5.3	Multi-plot of X-ray diffraction data for conventional-uniaxial sample; YIG before and after sintered at various temperatures. Note that Fe2O3 is Iron (III) Oxide, YFeO3 is Yttrium Orthoferrite and Y3Fe5O12 is YIG phase.	57
5.4	Multi-plot of X-ray diffraction data for monodisperse-uniaxial sample; YIG before and after sintered at various temperatures. Note that Fe2O3 is Iron (III) Oxide, YFeO3 is Yttrium Orthoferrite and Y3Fe5O12 is YIG phase.	58
5.5	Multi-plot of X-ray diffraction data for monodisperse-CIP sample; YIG before and after sintered at various temperatures. Note that $Fe_2O_3$ is Iron (III) Oxide, YFeO_3 is Yttrium Orthoferrite and Y <sub>3</sub> Fe <sub>5</sub> O <sub>12</sub> is YIG phase.	59
5.6	SEM micrographs and grain size distribution of conventional- uniaxial sample for YIG sintered at sintering temperature of (a) 800°C, (b) 900°C, (c) 1000°C, (d) 1100°C, and (e) 1200°C.	66
5.7	SEM micrographs and grain size distribution of monodisperse- uniaxial sample for YIG sintered at sintering temperature of (a) 700°C, (b) 750°C, (c) 800°C, (d) 850°C, (e) 900°C, (f) 950°C, (g)1000°C, (h) 1050°C, (i) 1100°C, (j) 115 0°C and (k) 1200°C.	70
5.8	SEM micrographs and grain size distribution of monodisperse- CIP sample for YIG sintered at sintering temperature of (a) 800°C, (b) 900°C, (c) 1000°C, (d) 1100°C and (e) 1200°C.	72
5.9	Initial permeability measured at room temperature in range of 10 MHz to 1 GHz for (a) conventional-uniaxial sample; (b) monodisperse-uniaxial sample; (c) monodisperse-CIP sample.	77

5.10	Initial permeability against grain size at (a) 10 MHz and (b) 100 MHz for conventional-uniaxial sample, monodisperse-uniaxial sample and monodisperse-CIP sample.	78
5.11	Loss factor measured at room temperature in range of 10 MHz to 1 GHz for (a) conventional-uniaxial sample; (b) monodisperse-uniaxial sample; (c) monodisperse-CIP sample.	81
5.12	Loss factor against grain size at (a) 10 MHz and (b) 100 MHz for conventional-uniaxial sample, monodisperse-uniaxial sample, and monodisperse-CIP sample.	82
5.13A	Tan $\delta$ measured at room temperature in range of 10 MHz to 1 GHz for (a) conventional-uniaxial sample; (b) monodisperse- uniaxial sample; (c) monodisperse-CIP sample.	86
5.13B	Tan $\delta/\mu$ ' deduced from Figure 5.13A.	87
5.14	Tan $\delta$ against grain size at (a) 10 MHz and (b) 100 MHz for conventional-uniaxial sample, monodisperse-uniaxial sample, and monodisperse-CIP sample.	88

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# LIST OF ABBREVIATIONS

ffrequencykHzkilohertzMHzmegahertzGHzgigahertzμ'real permeabilityμ'magnetic lossμ*complex permeabilityρdensityYFeO3Yttrium OrthoferriteFe_O3Iron OxideXRDX-ray diffractionSEMScanning Electron MicroscopynmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 heta degree	YIG	Yttrium Iron Garnet
kHzkilohertzMHzmegahertzGHzgigahertzµ*real permeabilityµ*magnetic lossµ*complex permeabilitypdensityYFeO3Yttrium OrthoferriteFe2O3Iron OxideXRDX-ray diffractionSEMSeanning Electron MicroscopynmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 theta degree	f	frequency
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Fe2O3Iron OxideXRDX-ray diffractionSEMScanning Electron MicroscopyTEMTransmission electron microscopynmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 theta degree	YFeO <sub>3</sub>	Yttrium Orthoferrite
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SEMScanning Electron MicroscopyTEMTransmission electron microscopynmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 theta degree	XRD	X-ray diffraction
TEMTransmission electron microscopynmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 theta degree	SEM	Scanning Electron Microscopy
nmnanometerMUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit202 theta degree	TEM	Transmission electron microscopy
MUTMaterial under testJCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit2θ2 theta degree	nm	nanometer
JCPDSJoint Committee on Power Diffraction Standarda.uArbitrary unit2θ2 theta degree	MUT	Material under test
a.u Arbitrary unit 20 2 theta degree	JCPDS	Joint Committee on Power Diffraction Standard
20 2 theta degree	a.u	Arbitrary unit
	20	2 theta degree

## **CHAPTER 1**

### INTRODUCTION

## 1.1 Background of the study

In the earlier part of electrical technology development, iron and its alloys were used as magnetic materials to supply the need of the electrical industry for a long time. However, with the introduction of higher frequencies, the standard techniques of using lamination or iron powder cores, reducing eddy current losses, were no longer cost effective or efficient. This realization stimulated a renewed interest in "magnetic insulators" since first reported by Hilpert in Germany (1909). The high electrical resistivity of oxides could be combined with desired magnetic characteristics and a magnetic material would result that was well suited for high frequency operation.

Various laboratories all over the world have done research to develop such a material, such as by V. Kato, T. Takei, and N. Kawai in the 1930's in Japan and by Snoek of the Philips' Research Laboratories in the period 1935-45 in the Netherlands (McLyman and McLyman, 2004). By 1945 Snoek had laid down the basic fundamentals of the physics and technology of practical ferrite materials. In 1948, the Neel theory (1948) of ferromagnetic provided the theoretical understanding of this type of magnetic material. These ferrites are ceramic, homogeneous materials composed of various oxides with iron oxide as their main constituent.

Generally, ferrites are classified into three classes based on three different crystal types which are:

- Soft Ferrite with spinel cubic structure, for example; Nickel Zinc
   Ferrite and Manganese Zinc Ferrite.
- ii) Soft Ferrite with garnet structure, for example yttrium-based garnets that are used in microwave devices.
- iii) Hard Ferrite with magnetoplumbite (hexagonal) structure. The hexagonal ferrites develop high coercivity and are an important member of the permanent magnet family.

Among most common representatives of soft ferrite with the garnet structure is  $Y_3$  Fe<sub>5</sub> O<sub>12</sub> (YIG) which has been widely investigated as it is an interesting ferrimagnetic material. This material has high resistivity and low magnetic loss at high frequency which has made it among the best UHF magnetic materials. It is widely used in the microwave frequency range and optical-communication devices and other applications (Vaqueiro et. al, 1997).

## **1.2 Ferrite for Microwave Application**

Microwave technology is moving up to higher frequencies and higher bandwidths, into the mm wave range, up to 100 GHz. Nonconducting materials are essential to ensure total penetration of electromagnetic fields. Ferrite materials are unique because they are one of a few classes of insulating magnetic oxides that possess moderate value of magnetization, high permeability, moderate to high permittivity, and low losses at frequencies from dc to sub-millimetre wavelengths. These properties add to them a great value in high frequency devices that require strong coupling to electromagnetic signals while experiencing low losses (Harris et al., 2009). Ferrite elements are widely used in microwave devices such as isolators, circulators, and phase shifters. For applications requiring nonreciprocal operation, as in circulators and isolators, there is no alternative to magnetic devices. Due to their very high specific resistance, remarkable flexibility in tailoring the magnetic properties, ease of preparation, and, last but not the least, price and performance considerations, ferrites are the first choice materials for microwave applications. However, the frequency range of operation, the power handling capacity and the temperature sensitivity of ferrite devices should be improved.

Nanostructured materials have a number of desirable electromagnetic and mechanical properties. Electromagnetic absorption properties can be controlled by changing the particle size distribution in nano-materials and application-specific, tailored materials can be produced. Eddy -current and magnetic losses are minimized in nano-materials and very sharp resonances can be set up leading to high-Q filter characteristics. This can be directly exploited in antenna technology. Few key aspects that make nano-materials very attractive candidates for antenna technology development are:

- i. Physical properties different from bulk and often superior
- ii. Superior mechanical properties
- iii. Better control of microstructure, porosity
- iv. Selective enhancement of desirable parameter

## **1.3 Problem Statement**

To appreciate the value of ferrites in microwave applications, it is important to understand the basic physical phenomena that are involved in successful device operation. Even though the technology in ferrites is advancing, there is still a lack of understanding and systematic investigation on how losses in ferrites occur and how to control them especially at UHF or higher frequencies.

Some magnetic properties of a ferrite depend critically on the structure and microstructure of the material. The dependence towards microstructure of the materials leads to the development of techniques to produce garnets with a strict control of the composition, homogeneity, size and particle shape (Vaqueiro et. al, 1997). This work would attempt to obtain extremely low loss nanometer-sized YIG particles by the water-in-oil microemulsion technique.

#### 1.4 Objectives

The goal of this research is to synthesize a high quality yttrium iron garnet with extremely low electromagnetic loss via the microemulsion technique. In order to accomplish this work, an experimental investigation was carefully conducted. The objectives are as follows:

 To prepare uniform and monodisperse Yttrium Iron Garnet nanoparticles from microemulsion technique.

- To study the effect of phase purity, crystal structure and microstructure on EM energy losses.
- To synthesize a garnet that can transmit and receive microwave energy with very little EM energy loss, employing synthesis variation effects.

## **1.5 Thesis Outline**

The earliest chapter of this thesis gives an introduction on ferrites, ferrites for microwave application and some research questions. Chapter two presents aspects on the related literature on low loss ferrites, synthesis methods, and some microstructural of ferrites. Chapter three reports the basic theories of magnetism, the ferromagnetic structure of garnet, and microemulsion synthesis. The preoccupations in chapter four are methodologies employed for the preparations and the characteristics measurement of the prepared YIG samples. The discussion of the results obtained forms chapter five. Chapter six summarizes and concludes the research findings, in addition to some suggested recommendations. The list of publications by the author is attached at the end of the thesis, preceded by the author's biography, appendices and references/bibliographies.

#### REFERENCES

- Bae, S., Hong, Y. K., Lee, J. J., Jalli, J., Abo, G. S., Lyle, A., Seong, W. M., et al. (2009). Low loss Z-type barium ferrite (Co<sub>2</sub>Z) for terrestrial digital multimedia broadcasting antenna application. *Journal of Applied Physics*, 105(7), 07A515.
- Chen, H., Wang, J. M., Pan, T., Xiao, H. M., Zhang, J. Q., & Cao, C. N. (2003). E ects of high-energy ball milling (HEBM) on the structure and electrochemical performance of nickel hydroxide, 28, 119-124.
- Cullity, B. D., & Stock, S. R. (2001). Elements of x-ray diffraction. Prentice Hall.
- Curri, M. ., Agostiano, a, Mavelli, F., & Della Monica, M. (2002). Reverse micellar systems: self organised assembly as effective route for the synthesis of colloidal semiconductor nanocrystals. *Materials Science and Engineering: C*, 22(2), 423-426.
- Drmota, a., Drofenik, M., & Žnidaršič, a. (2012). Synthesis and characterization of nano-crystalline strontium hexaferrite using the co-precipitation and microemulsion methods with nitrate precursors. *Ceramics International*, 38(2), 973-979.
- Fu, Y.-P., Lin, C.-H., Tay, K.-W., & Yao, Y.-D. (2007). Yttrium iron garnet ceramic prepared from microwave-induced combustion. *Journal of Electroceramics*, 21(1-4), 677-680.
- Gao, Y., Zhao, Y., Jiao, Q., & Li, H. (2013). Microemulsion-based synthesis of porous Co–Ni ferrite nanorods and their magnetic properties. *Journal of Alloys and Compounds*, 555, 95-100.
- Goldman, A. (1999). Handbook of Modern Ferromagnetic Materials (p. 646). Springer.
- Gupta, N., Dimri, M. C., Kashyap, S. C., & Dube, D. C. (2005). Processing and properties of cobalt-substituted lithium ferrite in the GHz frequency range. *Ceramics International*, 31(1), 171-176.

- Halliday, D., Resnick, R., & Walker, J. (1997). Fundamentals of Physics Without Softlock Cd-Physics, 2.0 (p. 1232). John Wiley & Sons.
- Han, D. Y., Yang, H. Y., Shen, C. B., Zhou, X., & Wang, F. H. (2004). Synthesis and size control of NiO nanoparticles by water-in-oil microemulsion. *Powder Technology*, 147, 113-116.
- Harris, V. G., Geiler, A., Chen, Y., Yoon, S. D., Wu, M., Yang, A., Chen, Z., et al. (2009). Recent advances in processing and applications of microwave ferrites. *Journal of Magnetism and Magnetic Materials*, 321(14), 2035-2047.
- Hosseini Vajargah, S., Madaah Hosseini, H. R., & Nemati, Z. A. (2006). Synthesis of nanocrystalline yttrium iron garnets by sol-gel combustion process: The influence of pH of precursor solution. *Materials Science and Engineering: B*, *129*(1-3), 211-215.
- Idza, I. R., Hashim, M., Rodziah, N., Ismayadi, I., & Norailiana, A. R. (2012).
  Influence of evolving microstructure on magnetic-hysteresis characteristics in polycrystalline nickel–zinc ferrite, Ni0.3Zn0.7Fe2O4. *Materials Research Bulletin*, 47(6), 1345-1352.
- Ismayadi, I (2012). An Elucidating Study on The Parallel Evolving Morphology, Magnetic Properties And Their Relationships In Ni<sub>0.5</sub>Zn<sub>0.5</sub>Fe<sub>2</sub>O<sub>4</sub>. Unpublisehed doctoral dissertation, Universiti Putra Malaysia, Malaysia.
- Ji, J. K., Ahn, W. K., Kum, J. S., Park, S. H., Kim, G. H., & Seong, W. M. (2010). Miniaturized T-DMB Antenna With a Low–Loss Ni–Mn–Co Ferrite for Mobile Handset Applications. *IEEE Magnetics Letters*, 1, 5000104-5000104.
- Karami, M. a., Shokrollahi, H., & Hashemi, B. (2012). Investigation of nanostructural, thermal and magnetic properties of yttrium iron garnet synthesized by mechanochemical method. *Journal of Magnetism and Magnetic Materials*, 324(19), 3065-3072. Elsevier.
- Kingery, W. D., Bowen, H. K., Uhlmann, D. R. (1976). Introduction to ceramics. United States: *Wiley*.

- Kotnala, R. K., Dar, M. A., Verma, V., Singh, A. P., & Siddiqui, W. A. (2010). Journal of Magnetism and Magnetic Materials Minimizing of power loss in Li – Cd ferrite by nickel substitution for power applications. *Journal of Magnetism* and Magnetic Materials, 322(22), 3714-3719. Elsevier.
- Lam, U. T., Mammucari, R., Suzuki, K., & Foster, N. R. (2008). Processing of Iron Oxide Nanoparticles by Supercritical Fluids. *Society*, 599-614.
- Lee, W. E., & Rainforth, W. M. (1994). Ceramic Microstructures: Property Control by Processing (p. 590). Springer.
- Lopez-quintela, M. A. (2003). Synthesis of nanomaterials in microemulsions formation mechanisms and growth control . *Interface Science*, 8, 137-144.
- Mathew, D., & Juang, R. (2007). Role of alcohols in the formation of inverse microemulsions and back extraction of proteins/enzymes in a reverse micellar system. *Separation and Purification Technology*, 53(3), 199-215.
- McLyman, W.T., McLyman, C. W. T. (2004). *Transformer and Inductor Design Handbook* (p. 556). CRC Press.
- Minghao, F., Juntong, H., Zhaohui, H., Yangai, L., Bin, J., & Peng, P. (2008). Solid Phase Synthesis and Sintering Properties of Yttrium Iron Garnet. *Key Engineering Materials*, 372, 588-590.
- Nasr Isfahani, M. J., Isfahani, P. N., Da Silva, K. L., Feldhoff, A., & Šepelák, V. (2011). Structural and magnetic properties of NiFe<sub>2-x</sub> Bi<sub>x</sub>O<sub>4</sub> (x=0, 0.1, 0.15) nanoparticles prepared via sol–gel method. *Ceramics International*, *37*(6), 1905-1909.
- Otsuki, E., Yamada, S., Otsuka, T., Shoji, K., & Sato, T. (1991). Microstructure and physical properties of Mn-Zn ferrites for high-frequency power supplies. *Journal of Applied Physics*, 69(8), 5942.
  - Rajendran, M., Deka, S., Joy, P. a., & Bhattacharya, A. K. (2006). Size-dependent magnetic properties of nanocrystalline yttrium iron garnet powders. *Journal of Magnetism and Magnetic Materials*, 301(1), 212-219.

- Rodziah, N., Hashim, M., Idza, I. R., Ismayadi, I., Hapishah, A. N., & Khamirul, M. A. (2012). Dependence of developing magnetic hysteresis characteristics on stages of evolving microstructure in polycrystalline yttrium iron garnet. *Applied Surface Science*, 258(7), 2679-2685. Elsevier B.V.
- Shinde, T. J., Gadkari, A. B., & Vasambekar, P. N. (2008). DC resistivity of Ni–Zn ferrites prepared by oxalate precipitation method. *Materials Chemistry and Physics*, 111(1), 87-91.
- Shirakata, Y.,Hidaka, N., Ishitsuka, M. A. T. and T. O. (2009). Low-Loss Composite Material Containing Fine Zn – Ni – Fe Flakes for High-Frequency Applications. *IEEE Transactions on Magnetics*, 45(10), 4337-4340.
- Snelling, E. C. (1988). Soft Ferrites: Properties and Applications, 2<sup>nd</sup> ed. Boston,
  M.A: Butterworths.
- Vaqueiro, P., M. A. L.-Q. and J. R. (1997). Synthesis of yttrium iron garnet nanoparticles via coprecipitation in microemulsion. *Powder Diffraction*, 7, 501-504.
- Waje, S. B., Hashim, M., & Ismail, I. (2011). Effects of sintering temperature on grain growth and the complex permeability of Co<sub>0.2</sub> Ni<sub>0.3</sub> Zn<sub>0.5</sub> Fe<sub>2</sub>O<sub>4</sub> material prepared using mechanically alloyed nanoparticles. *Journal of Magnetism and Magnetic Materials*, 323(11), 1433-1439. Elsevier.
- Xiang, C. C. (2011). Low-loss Z-type hexaferrite with Sr-substitution for microwave antenna miniaturization. *IEEE*, 1-4.
- Xu, Z., Yu, Z., Sun, K., Li, L., Ji, H., & Lan, Z. (2009). Microstructure and magnetic properties of Sn-substituted MnZn ferrites. *Journal of Magnetism and Magnetic Materials*, 321(18), 2883-2889. Elsevier.
- Yang, H. K., & Jeong, J. H. (2010). Synthesis , Crystal Growth , and
   Photoluminescence Properties of YAG□: Eu<sup>3+</sup> Phosphors by High-Energy Ball
   Milling and Solid-State Reaction. *Society*, *12*, 226-230.

- Yang, H. K., Chung, J. W., Moon, B. K., Jeong, J. H., Jang, K.-wan, Lee, H. S., & Yi, S. S. (2009). Photoluminescence investigations of YAG:Eu nanocomposite powder by high-energy ball milling. *Current Applied Physics*, 9(2), e86-e88. Elsevier B.V.
- Yang, Q., Zhang, H., Liu, Y., Wen, Q., & Jia, L. (2008). The magnetic and dielectric properties of microwave sintered yttrium iron garnet (YIG). *Materials Letters*, 62(17-18), 2647-2650.
- Yasushi Shirakata, Nobuhiro Hidaka, Masayuki Ishitsuka, A. T. and T. O. (2009). Low-Loss Composite Material Containing Fine Zn – Ni – Fe Flakes for High-Frequency Applications. *IEEE Transactions on Magnetics*, 45(10), 4337-4340.
- Yu, H., Zeng, L., Lu, C., Zhang, W., & Xu, G. (2011). Synthesis of nanocrystalline yttrium iron garnet by low temperature solid state reaction. *Materials Characterization*, 62(4), 378-381. Elsevier Inc.