

UNIVERSITI PUTRA MALAYSIA

MICROSTRUCTURAL, OPTICAL AND MAGNETIC PROPERTIES OF BARIUM HEXAFERRITE AND NICKEL ZINC FERRITE SYNTHESIZED VIA MECHANOCHEMICAL PROCEDURE

LOW ZHI HUANG

ITMA 2018 15



MICROSTRUCTURAL, OPTICAL AND MAGNETIC PROPERTIES OF BARIUM HEXAFERRITE AND NICKEL ZINC FERRITE SYNTHESIZED VIA MECHANOCHEMICAL PROCEDURE



Thesis submitted to the School of Graduate Studies, Universiti Putra Malaysia, in Fulfilment of the Requirements for the Degree of Doctor of Philosophy

May 2018

COPYRIGHT

All material contained within the thesis, including without limitation text, logos, icons, photographs and all other artwork, is copyright material of Universiti Putra Malaysia unless otherwise stated. Use may be made of any material contained within the thesis for non-commercial purposes from the copyright holder. Commercial use of material may only be made with the express, prior, written permission of Universiti Putra Malaysia.

Copyright © Universiti Putra Malaysia



Abstract of thesis presented to the Senate of Universiti Putra Malaysia in fulfillment of the requirement for the degree of Doctor of Philosophy

MICROSTRUCTURAL, OPTICAL AND MAGNETIC PROPERTIES OF BARIUM HEXAFERRITE AND NICKEL ZINC FERRITE SYNTHESIZED VIA MECHANOCHEMICAL PROCEDURE

By

LOW ZHI HUANG

May 2018

Chairman: Assoc. Prof. Chen Soo Kien, PhD Institute: Institute of Advanced Technology

Mechanochemical process is a powder processing technique that utilises mechanical energy to grind down bulk materials. Mechanochemical process has received a lot of interest for producing technologically important ferrites because it is a solvent-free technique and hence green process. Throughout the centuries, the applications of mechanochemical process are limited to diminution of particles because the lack of systematic studies on the process mechanisms of mechanochemical process. The immediate objective of this research is devoted to this subject by developing a systematic study on top-down approach mechanochemical process (referring to the production of nanoparticles by mechanochemical process) and mechanochemical activation-based synthesis (referring to mechanochemical process, used to activate the starting powders, before a sintering step to induce the formation of final product). For top-down approach mechanochemical process, starting bulk materials were mechanically treated for different milling time ranging from 1 to 20 hours at room temperature, for the preparation of nanoparticles. Evidence of the presence of single phase ferrites was identified by XRD. Rietveld refinement analysis suggested the deformation of a mechanically triggered polyhedral in the magnetoplumbite structure of BaFe₁₂O₁₉ and spinel structure Ni_{0.5}Zn_{0.5}Fe₂O₄. Three distinct stages of the mechanochemical mechanism were observed when the milling time was extended. The average crystallite size decreased at different rate during the first stage and the intermediate stage, and increased during the final stage of the mechanochemical process. FESEM micrographs showed the particle size decreased from 432.96 nm to 81.43 nm for BaFe₁₂O₁₉ and 371.68 nm to 158.49 nm for Ni_{0.5}Zn_{0.5}Fe₂O₄ during the first stage and the intermediate stage. In the final stage, particle size increased to 134.15 nm for BaFe₁₂O₁₉ and 193.60 nm for Ni_{0.5}Zn_{0.5}Fe₂O₄. HRTEM micrographs suggested the formation of a non-uniform nanostructure shell surrounding the ordered core materials. The thickness of the shell extended up to 12 nm during the first and intermediate stages, and diminished to approximately 3 nm during final stage. VSM results showed a mixture of ferromagnetic, superparamagnetic, and paramagnetic

behaviours attributed to the defects, distorted polyhedra, and non-equilibrium amorphous layers induced by the mechanical energy. The observed spectral shift from UV-Vis spectra was ascribed to the competition between quantum confinement effects and structural disorder bandgap narrowing effect. For mechanochemical activationbased synthesis, mechanochemical process on the starting powders and subsequent sintering was carried out to synthesize $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles. The XRD results indicated an improvement of crystallinity with increasing sintering temperature. Single phase ferrites were observed at 1100 °C for BaFe₁₂O₁₉ and 700 °C for Ni_{0.5}Zn_{0.5}Fe₂O₄. FESEM micrographs showed the particle size increased from 42.24 nm to 913.96 nm for BaFe₁₂O₁₉ and 66.39 nm to 1084.27 nm for Ni_{0.5}Zn_{0.5}Fe₂O₄ when sintering temperature were elevated from 600 °C to 1200 °C. Morphological studies showed three stages of sintering with distinct microstructure features. By sintering from 600 °C to 1200 °C, a dependence of magnetic properties on sintering temperature was found. Maximum magnetization at 10 kOe improved with elevating sintering temperature. The optical bandgap values decreased with increasing crystallite size, showing the dominancy of quantum confinement effects. It can be concluded top-down approach mechanochemical process is capable of producing single phase nanoparticles; and mechanochemical activation-based synthesis has significantly reduced the sintering temperature required for the formation of final product. The systematic studies on the process mechanisms of top-down approach mechanochemical process and mechanochemical activation-based synthesis developed a fundamental knowledge to tailor nanoparticles with specific properties according to its possible industrial applications.

Abstrak tesis yang dikemukakan kepada Senat Universiti Putra Malaysia sebagai memenuhi keperluan untuk Ijazah Doktor Falsafah

SIFAT-SIFAT MIKROSTRUKTURAL, OPTIKAL DAN MAGNETIK FERIT HEKSA BARIUM DAN FERIT NIKEL ZINK YANG DISINTESIS MENGGUNAKAN PROSEDUR MEKANOKIMIA

Oleh

LOW ZHI HUANG

Mei 2018

Pengerusi: Prof. Madya Dr. Chen Soo Kien, PhD Institut: Institut Teknologi Maju

Proses mekanokimia adalah teknik pemprosesan serbuk yang menggunakan tenaga mekanikal untuk menghaluskan bahan-bahan pukal. Proses mekanokimia telah menerima banyak sambutan bagi menghasilkan ferit-ferit penting berteknologi tinggi kerana proses ini bebas pelarut dan dengan itu proses ini adalah proses hijau. Sepanjang abad yang lalu, penggunaan proses mekanokimia terbatas kepada pengecilan saiz zarah kerana kurangnya kajian sistematik ke atas mekanisme proses mekanokimia. Objektif utama kajian ini adalah untuk menumpukan kepada perkara ini dengan membangunkan sebuah kajian sistematik ke atas pendekatan proses mekanokimia atas ke bawah (merujuk kepada penghasilan zarah nano melalui proses mekanokimia) dan proses sintesis berdasarkan pengaktifan mekanokimia (berdasarkan proses mekanokimia, digunakan untuk mengaktifkan serbuk-serbuk permulaan sebelum langkah pensinteran untuk mendorong pembentukan produk akhir). Bagi pendekatan proses mekanokimia atas ke bawah, bahan-bahan pukal permulaan telah dirawat secara mekanikal pada masa pengisaran berbeza dari 1 hingga 20 jam pada suhu bilik untuk penyediaan zarah nano. Bukti kehadiran fasa tunggal ferit-ferit telah dikenalpasti menggunakan XRD. Analisis perbaikan Rietveld mencadangkan bahawa ubah bentuk polihedral yang dicetuskan secara mekanikal dalam struktur magnetoplumbit BaFe₁₂O₁₉ dan struktur spinel Ni_{0.5}Zn_{0.5}Fe₂O₄. Tiga peringkat berbeza bagi mekanisme mekanokimia telah diperhatikan apabila masa pengisaran dipanjangkan. Purata saiz kristal berkurangan pada kadar yang berbeza semasa peringkat pertama dan peringkat pertengahan, dan meningkat pada peringkat akhir proses mekanokimia. Mikrograf-mikrograf FESEM menunjukkan saiz zarah berkurang dari 432.96 nm kepada 81.43 nm untuk BaFe₁₂O₁₉ dan dari 371.68 nm kepada 158.49 nm untuk $Ni_{0.5}Zn_{0.5}Fe_2O_4$ semasa peringkat pertama dan peringkat pertengahan. Dalam peringkat akhir, saiz zarah meningkat kepada 134.15 nm bagi BaFe₁₂O₁₉ dan 193.60 nm bagi Ni_{0.5}Zn_{0.5}Fe₂O₄. Mikrograf-mikrograf HRTEM mencadangkan pembentukan kerangka nanostruktur tidak seragam yang mengelilingi bahan-bahan teras yang tersusun. Ketebalan lapisan kerangka adalah sehingga 12 nm semasa peringkat pertama dan pertengahan, dan berkurang kepada sekurang-kurangnya

3 nm semasa peringkat akhir. Keputusan-keputusan VSM menunjukkan campuran sifat-sifat ferromagnetik, superparamagnetik dan paramagnetik yang disebabkan oleh kecacatan, polyhedra terherot, dan ketidakseimbangan lapisan-lapisan amorfus yang didorongkan oleh tenaga mekanikal. Anjakan spektrum dari spektra UV-Vis yang diperhatikan adalah disebabkan oleh persaingan diantara kesan-kesan kurungan kuantum dan kesan penyempitan jurang jalur struktur terganggu. Bagi sintesis berdasarkan pengaktifan mekanokimia, proses mekanokimia ke atas serbuk-serbuk permulaan dan seterusnya pensinteran telah dilakukan untuk mensintesis zarah nano BaFe₁₂O₁₉ dan Ni_{0.5}Zn_{0.5}Fe₂O₄. Keputusan XRD menunjukkan penambahbaikan penghabluran dengan peningkatan suhu pensinteran. Fasa tunggal ferit telah diperhatikan pada suhu pensinteran 1100 °C dan 700 °C masing-masing bagi BaFe₁₂O₁₉ dan Ni_{0.5}Zn_{0.5}Fe₂O₄. Mikrograf-mikrograf FESEM menunjukkan saiz zarah menambah dari 42.24 nm kepada 913.96 nm untuk BaFe₁₂O₁₉ dan dari 66.39 nm kepada 1084.27 nm untuk Ni_{0.5}Zn_{0.5}Fe₂O₄ apabila suhu persinteran meningkat dari 600 °C ke 1200 °C. Kajian-kajian morfologi menunjukkan tiga tahap pensinteran dengan ciri-ciri mikrostruktur yang berbeza. Dengan melakukan pensinteran dari 600 °C ke 1200 °C, suatu kebergantungan sifat-sifat magnetik ke atas suhu pensinteran telah dijumpai. Kemagnetan maksimum pada 10 kOe ditingkatkan dengan peningkatan suhu pensinteran. Nilai-nilai jalur tenaga optikal berkurang dengan peningkatan saiz kristal, menunjukkan dominasi kesan-kesan kurungan kuantum. Dapat disimpulkan bahawa pendekatan proses mekanokimia atas ke bawah mampu menghasilkan fasa tunggal zarah nano; dan sintesis berdasarkan pengaktifan mekanokimia telah merendahkan suhu pensinteran dengan ketara bagi menghasilkan produk akhir. Kajian-kajian sistematik ke atas mekanisme proses pendekatan mekanokimia atas ke bawah dan sintesis berdasarkan pengaktifan mekanokimia telah membangunkan pengetahuan asas untuk mengubahsuai zarah nano dengan sifat-sifat tertentu bergantung kepada aplikasiaplikasi industri yang berkemungkinan.

ACKNOWLEDGEMENTS

First of all, I would like express my sincere thanks and deep appreciation to my supervisor, Associate Prof. Dr. Chen Soo Kien for his guidance, encouragement, support, and patience throughout the course of my study. I'm always grateful to have Dr. Chen as my supervisor, supporting me during the hardest period, bringing the best out of me, and always run whole heartedly with me till the finishing line. We solve and get over problems together, and she always advise and encourage me to be a better researcher. It is my pleasure to be able to learn from Dr. Chen and good to work with him, always.

Also, I would like to extend my gratitude to my ex-supervisors, Associate Prof. Dr. Mansor Hashim. There are many things that I feels but unable to put into words, because the impact Dr. Mansor had on my life is one that is unable to be comprehended. Thank you for making me a passionate researcher, as an incredible educator, an honourable mentor, and a friend. May you rest in peace, and hope one day, I will live a life that will make you feel proud.

Not to forget my co-supervisors, Dr. Ismayadi Ismail, Dr. Josephine Liew Ying Chyi, Dr. Tan Kim Song, and Dr. Yap Wing Fen for their valuable ideas, encouragement and continuous support, making this research better. They are generous in sharing knowledge, spending their valuable time for discussion and guiding me the best that they could. I'm proud to have such energetic supervisory committee members.

Not forgetting to thank the entire staff of the Institute of Advanced Technology, UPM for their advices and assistances in my experimental and analyses work. In addition, I would like to thank my friends and colleagues Kak Idza, Kak Rod, Kak Faz, Kak Pisha, Kak Nora, Kak Dilah, Abang Shamsul, Farah, Diyah, Kriss Kumar, Wan, Misbah for sharing with me their friendships, knowledge and assistances throughout this project.

Moreover, I feel deeply grateful to MyBrainSc scholarship programme under the Ministry of Higher Education, Malaysia for the financial supports throughout my Ph.D.

Last but not least, I would like to give my heartfelt and special thanks to my family for their endless support, understanding, and love. They are always my best stories listeners no matter what, when and where.

I could not have completed this project without anyone of them. Thank you very much, all.

I certify that a Thesis Examination Committee has met on 31 May 2018 to conduct the final examination of Low Zhi Huang on his thesis entitled "Microstructural, Optical and Magnetic Properties of Barium Hexaferrite and Nickel Zinc Ferrite Synthesized via Mechanochemical Procedure" in accordance with the Universities and University Colleges Act 1971 and the Constitution of the Universiti Putra Malaysia [P.U.(A) 106] 15 March 1998. The Committee recommends that the student be awarded the Doctor of Philosophy.

Members of the Thesis Examination Committee were as follows:

Halimah binti Mohamed Kamari, PhD

Professor Faculty of Science Universiti Putra Malaysia (Chairman)

Abdul Halim bin Shaari, PhD Professor Faculty of Science Universiti Putra Malaysia (Internal Examiner)

Khamirul Amin bin Matori, PhD

Associate Professor Faculty of Science Universiti Putra Malaysia (Internal Examiner)

Xiaoding Qi, PhD

Professor National Cheng Kung University Taiwan (External Examiner)

RUSLI HAJI ABDULLAH, PhD Professor and Deputy Dean School of Graduate Studies Universiti Putra Malaysia

Date: 30 July 2018

This thesis was submitted to the Senate of Universiti Putra Malaysia and has been accepted as fulfillment of the requirement for the degree of Doctor of Philosophy. The members of the Supervisory Committee were as follows:

Chen Soo Kien, PhD Associate Professor Faculty of Science Universiti Putra Malaysia (Chairman)

Ismayadi Ismail, PhD Research Officer Institute of Advanced Technology Universiti Putra Malaysia (Member)

Josephine Liew Ying Chyi, PhD Senior Lecturer

Faculty of Science Universiti Putra Malaysia (Member)

Yap Wing Fen, PhD

Senior Lecturer Faculty of Science Universiti Putra Malaysia (Member)

Tan Kim Song, PhD

Head of Programme Rubber Research Institute Malaysia Malaysia (Member)

ROBIAH BINTI YUNUS, PhD

Professor and Dean School of Graduate Studies Universiti Putra Malaysia

Date:

Declaration by graduate student

I hereby confirm that:

- this thesis is my original work;
- quotations, illustrations and citations have been duly referenced;
- this thesis has not been submitted previously or concurrently for any other degree at any other institutions;
- intellectual property from the thesis and copyright of thesis are fully-owned by Universiti Putra Malaysia, as according to the Universiti Putra Malaysia (Research) Rules 2012;
- written permission must be obtained from supervisor and the office of Deputy Vice-Chancellor (Research and Innovation) before thesis is published (in the form of written, printed or in electronic form) including books, journals, modules, proceedings, popular writings, seminar papers, manuscripts, posters, reports, lecture notes, learning modules or any other materials as stated in the Universiti Putra Malaysia (Research) Rules 2012;
- there is no plagiarism or data falsification/fabrication in the thesis, and scholarly integrity is upheld as according to the Universiti Putra Malaysia (Graduate Studies) Rules 2003 (Revision 2012-2013) and the Universiti Putra Malaysia (Research) Rules 2012. The thesis has undergone plagiarism detection software.

Signature: _____ Date: _____

Name and Matric No.: Low Zhi Huang GS40429

Declaration by Members of Supervisory Committee

This is to confirm that:

- the research conducted and the writing of this thesis was under our supervision;
- supervision responsibilities as stated in the Universiti Putra Malaysia (Graduate Studies) Rules 2003 (Revision 2012-2013) are adhered to.

| Signature: Name of Chairman of Supervisory | |
|--|--------------------------------|
| Committee: | Assoc. Prof. Dr. Chen Soo Kien |
| Signature: Name of Member of Supervisory Committee: | Dr. Ismayadi Ismail |
| Signature: Name of Member of Supervisory Committee: | Dr. Josephine Liew Ying Chyi |
| Signature: Name of Member of Supervisory Committee: | Dr. Yap Wing Fen |
| Signature: Name of Member of Supervisory Committee: | Dr. Tan Kim Song |

TABLE OF CONTENTS

| | | | Page |
|-------------|----------------------------|------------------------|----------------|
| ABSTRACT | | | i |
| ABSTRAK | | | iii |
| ACKNOWLI | DGEMENTS | | v |
| APPROVAL | | | vi |
| DECLARAT | [ON | | vii |
| LIST OF TA | BLES | | xv |
| LIST OF FIG | URES | | xviii |
| LIST OF AB | REVIATIONS AND SYM | BOLS | XXV |
| 2101 01 112 | | | |
| CHAPTER | | | |
| | | | |
| 1 INT | RODUCTION | | 1 |
| 1.1 | Background of the study | | 1 1 |
| 1.2 | Relationship between cha | racteristics and me | echanically 2 |
| | induced microstructural re | sponse of ferrites | |
| 1.3 | Problem statement | | 3 |
| 1.4 | Objectives and hypotheses | of the study | 4 |
| 1.5 | Scope of study | | 5 |
| 1.6 | Thesis outline | | 5 |
| | | | |
| | | | |
| 2 LIT | ERATURE REVIEW | | 6 |
| 2.1 | Introduction | | 6 |
| 2.2 | Some aspects on the r | elationship betwee | en chosen 6 |
| | ceramic synthesis techniqu | e and ferrite behavi | ours |
| 2.3 | Hard and soft ferrite na | anoparticles prepar | ration and 7 |
| | processing | | |
| 2.4 | Bottom-up (BU) approach | synthesis | 8 |
| | 2.4.1 Sol-gel process | | 9 |
| | 2.4.2 Chemical copred | cipitation process | 10 |
| | 2.4.3 Hydrothermal pr | ocess | 11 |
| | 2.4.4 Mechanochemic | al activation-based | synthesis 12 |
| 2.5 | Top-down (TD) approach | synthesis | 13 |
| | 2.5.1 Mechanochemis | try | 14 |
| | 2.5.2 Mechanochemic | al synthesis of | complex 14 |
| | ceramic oxides | | |
| | 2.5.3 Process variab | oles or factors | affecting 16 |
| | mechanochemic | al synthesis | |
| | 2.5.3.1 Typ | es of ball mills | 16 |
| | 2.5.3.2 Via | s or milling contain | ner 17 |
| | 2.5.3.3 Mil | ing medium | 17 |
| | 2.5.3.4 Mill | ing speed and energ | gy 18 |
| | 2.5.3.5 Mill | ing time | 19 |
| | 2.5.3.6 Exte | ent of filling the via | l and Ball- 20 |
| | to-P | owder Ratio (BPR) | |

| | | | 2.5.3.7 | Milling environme | atmosphere ent | and | 21 |
|---|-----|----------|---------------|------------------------------|--|---------|----|
| | 2.6 | Microst | ructural asp | ects of ferrite | S | | 22 |
| | | 2.6.1 | Size depe | ndant behavi | ours | | 22 |
| | | 2.6.2 | Defects a | nd porosity | | | 23 |
| | | 2.6.3 | Boundary | region | | | 24 |
| 3 | THE | ORY | | | | | 26 |
| | 3.1 | Introdu | ction | | | | 26 |
| | 3.2 | The orig | gin of magne | etism | | | 26 |
| | 3.3 | Magnet | ization | | | | 27 |
| | | 3.3.1 | Diamagn | etism and par | amagnetism | | 28 |
| | | 3.3.2 | Ferromag | netism and a | ntiferromagnetisr | n | 29 |
| | | 3.3.3 | Ferrimag | netism | | | 30 |
| | | 3.3.4 | Superpara | a <mark>m</mark> agnetism | | | 30 |
| | 3.4 | Classifi | cation of fer | rites | | | 31 |
| | | 3.4.1 | Hexaferri | tes: BaFe ₁₂ O | 19 | | 32 |
| | | 3.4.2 | Spinel fer | rrites: Ni _{0.5} Zn | $h_{0.5}$ Fe ₂ O ₄ | | 35 |
| | 3.5 | Magnet | ic behaviour | s of ferrites | | | 36 |
| | | 3.5.1 | Intrinsic p | properties | | | 36 |
| | | | 2.5.1.1 | Magnetic | moments | | 36 |
| | | | 2.5.1.2 | Exchange | interaction | | 36 |
| | | | 2.5.3.3 | Saturation | magnetization (<i>I</i> | (M_s) | 37 |
| | | | 2.5.3.4 | Magnetic | anisotropy | | 37 |
| | | 3.5.2 | Extrinsic | properties | | | 38 |
| | | | 2.5.2.1 | Domains a | and domain walls | | 38 |
| | | | 2.5.2.2 | Magnetic | hysteresis | | 40 |
| | 3.6 | Hard an | d soft magn | etic Materials | 8 | | 41 |
| | | 3.6.1 | Hard Mag | gnetic materia | als | | 42 |
| | | 3.6.2 | Soft mag | netic material | ls | | 42 |
| | 3.7 | Sinterin | g | | | | 43 |
| | | 3.7.1 | Types of | sintering | | | 43 |
| | | 3.7.2 | Driving f | orce of sinter | ing | | 44 |
| | | 3.7.3 | Three sta | ges of sintering | ng | | 44 |
| | | 3.7.4 | Mechanis | ms of sinteri | ng | | 45 |
| | | 3.7.5 | Grain gro | wth and coar | sening | | 46 |
| | 3.8 | Mechan | ism of mech | nanochemical | process | | 47 |
| 4 | MET | HODOL | OGY | | | | 49 |
| | 4.1 | Introdu | ction | | | | 49 |
| | 4.2 | Sample | selection | | | | 49 |
| | 4.3 | Researc | h design | | | | 49 |

| | · · · · · · | | | | |
|-----|-------------|------------------------|------------------------------------|--|----|
| 4.3 | Researc | h design | | | 49 |
| 4.4 | Raw che | Raw chemical materials | | | |
| 4.5 | Sample | preparation and | d experimen | t procedure | 50 |
| | 4.5.1 | Top-down | approach | mechanochemical | 51 |
| | | process of | BaFe ₁₂ O ₁₉ | and Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ | |

xi

systems

| | 4.5.2 | Mechanocl of BaFe ₁₂ C | hemical activation-based synthesis D ₁₀ and Ni ₀ ₅ Zn ₀ ₅ Fe ₂ O ₄ systems | 52 |
|--------|----------------------|--------------------------------------|--|----------|
| 4.6 | Characte | rizations | 19 | 55 |
| | 4.6.1 | Physical ar | nd structural properties | 55 |
| | | 4.6.1.1 | X-ray Diffractometry (XRD) | 55 |
| | | 4.6.1.2 | High Resolution Transmission Electron Microscopy (HRTEM) | 58 |
| | | 4.6.1.3 | FieldEmissionScanningElectronMicroscopy(FESEM) | 59 |
| | 4.6.2 | Magnetic p 4.6.2.1 | oroperties measurements Vibrating Sample Magnetometer | 60 60 |
| | | | (VSM) | |
| | | 4.6.2.2 | Electron Spin Resonance (ESR) | 61 |
| | 4.6.3 | NIR-UV-V | vis spectrophotometer | 61 |
| | 4.6.4 | Vector Net | work Analyser (VNA) | 63 |
| | 4.6.5 | Error estin | nation | 64 |
| 5 RESU | LTS ANI | DISCUSSI | ION | 66 |
| 51 | Introduc | tion | | 66 |
| 5.2 | Top-dov | vn approach | synthesis of barium hexaferrite. | 66 |
| 0.2 | BaFe ₁₂ O | and nickel | zinc ferrite. Ni $_{0.5}$ Zn $_{0.5}$ Fe $_{2}$ O $_{4}$ | 00 |
| | 521 | The bulk n | naterial | 66 |
| | 5.2.2 | Phase eva | luation of mechanically induced | 69 |
| | 0.12.12 | BaFe ₁₂ O ₁₀ | and Ni _{0.5} Zn _{0.5} Fe ₂ O_4 nanoparticles | 07 |
| | 5.2.3 | Crystal str | ucture evaluation of mechanically | 70 |
| | | induced | BaFe ₁₂ O ₁₀ and Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ | |
| | | nanopartic | les | |
| | 5.2.4 | Morpholog | cical evaluation of mechanically | 81 |
| | | induced nanopartic | $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ les | |
| | | 5.2.4.1 | Particle size evolution of BaFe ₁₂ O ₁₉ nanoparticles | 82 |
| | | 5.2.4.2 | Particle size evolution of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 85 |
| | | 5.2.4.3 | Different stages of mechanochemical process based on particle size evolution | 88 |
| | 5.2.5 | Mechanica mechanoch | l induced defects during nemical process | 90 |
| | | 5.2.5.1 | Defects during early stage of mechanochemical process | 90 |
| | | 5.2.5.2 | Defects during intermediate stage of mechanochemical process | 92 |
| | | 5.2.5.3 | Defects during final stage of mechanochemical Process | 94 |

| 5.2.6 | Three | proposed | mechan | nisms | of | 96 |
|--------------------|--|---|--|--|---|--|
| 5 7 7 | mechanoch Magnotic | emical proc | ess r ovelu | untion | of | 08 |
| 5.2.1 | mechanical | ly induced | nanonarticl | les | 01 | 90 |
| | 5 2 7 1 | Magnetic | naram | neters | of | 98 |
| | 5.2.7.1 | mechanica | illy induce | d BaFe | Ω_{10} | 90 |
| | | nanopartic | les | a bar o ₁ | 2019 | |
| | 5.2.7.2 | Magnetic | param | ieters | of | 102 |
| | | mechanica | ılly | indu | uced | |
| | | $Ni_{0.5}Zn_{0.5}F$ | e ₂ O ₄ nanop | particles | | |
| 5.2.8 | Electron | Spin Re | sonance | (ESR) | of | 105 |
| | mechanical | ly induc | ed BaFe | $e_{12}O_{19}$ | and | |
| 520 | $N1_{0.5}Zn_{0.5}Fe$ | 20_4 nanopa | armachility | t of Vk | and | 100 |
| 5.2.9 | froquoncios | nagnetic p | | at A-C | | 109 |
| | BaFeroOro | and Nio $-7n$ | Echamican | y mut nonarticl | es | |
| 5 2 10 | Ontical bar | ndgan evali | $0.51 \circ 204$ flat | mechanic | ally | 112 |
| 5.2.10 | induced nat | oparticles | Junion of 1 | meename | ally | 112 |
| | 5 2 10 1 | The offect | e of structu | ral disor | dore | 112 |
| | 5.2.10.1 | and size of | n ontical ha | andgan v | alue | 112 |
| | | of BaFero | D_{10} nanonal | rticles | arue | |
| | 5.2.10.2 | The effect | s of structu | ral disor | ders | 115 |
| | | and size of | n optical ba | andgap v | alue | - |
| | | of Ni _{0.5} Zn ₀ | $_{0.5}$ Fe ₂ O ₄ name | noparticl | es | |
| Mechano | chemical ac | tivation-bas | ed synthes | sis of bar | ium | 117 |
| hexaferri | te. BaFe ₁₂ | D_{10} and | nickel zi | inc Fer | rite. | 117 |
| $Ni_{0.5}Zn_{0.4}$ | Fe_2O_4 | -1, | | | , | |
| 531 | Dhase and | crustal s | tructure | voluction | of | 117 |
| 5.5.1 | mechanoch | emical | acti | varianon-h | ased | 11/ |
| | synthesized | nanopartic | les | vation of | 1500 | |
| | 5.3.1.1 | Sintarad B | | | | |
| | | Sincred D | $aFe_{12}O_{19}p$ | owder | | 117 |
| | 5.3.1.2 | Sintered D Sintered N | $aFe_{12}O_{19} p_{10.5}Fe_{2}$ | owder $_{2}O_{4}$ powd | ler | 117 121 |
| 5.3.2 | 5.3.1.2 Morpholog | Sintered D Sintered N ical | $aFe_{12}O_{19} p$ $i_{0.5}Zn_{0.5}Fe_2$ evaluation | owder 2O4 powd n | ler of | 117 121 126 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch | Sintered N Sintered N ical emical | aFe ₁₂ O ₁₉ p li _{0.5} Zn _{0.5} Fe ₂ evaluation activ | owder 2O4 powd n vation-ba | ler of 1sed | 117 121 126 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized | Sintered D Sintered N ical emical nanopartic | $aFe_{12}O_{19} p_{10,5}Fe_{2}$ evaluation activities | owder ₂ O ₄ powd n vation-ba | ler of ased | 117 121 126 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 | Sintered D Sintered N ical emical nanopartic Particle size | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_{2}$ evaluation activities ze evolution | owder 2O4 powc n vation-ba n of sinte | ler of ased ered | 117 121 126 126 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 | Sintered B Sintered N ical emical nanopartic Particle siz $BaFe_{12}O_{19}$ | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_{2}$ evaluation activities ze evolution nanopartic | owder 2O4 powc n vation-ba n of sinte tles | ler of ased ered | 117 121 126 126 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz | $aFe_{12}O_{19} p_{10,5}Fe_{2}$ evaluation activities ze evolution nanopartic ze evolution | owder ₂ O ₄ powc n vation-ba n of sinte n of sinte | ler of ased ered ered | 117 121 126 126 131 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charge | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F | $aFe_{12}O_{19} p_{10}$ $i_{10,5}Zn_{0,5}Fe_{2}$ evaluation activities les ze evolution nanopartic ze evolution $ie_{2}O_{4}$ nanop | owder ₂ O ₄ powc n vation-ba n of sinte cles n of sinte particles three st | ler of ased ered ered | 117 121 126 126 131 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_{2}$ evaluatio activites ze evolution nanopartic ze evolution $e_{2}O_{4}$ nanop particles at | owder ₂ O ₄ powc n vation-ba n of sinte les n of sinte particles t three sta | ler of ased ered ered ages | 117 121 126 126 131 134 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering 5.3.3.1 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_2$ evaluatio activites ze evolution nanopartic ze evolution $e_{2}O_4$ nanop particles at | owder 2O4 powc n vation-ba n of sinte eles n of sinte particles t three sta | ler of ased ered ered ages of | 117 121 126 126 131 134 134 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering 5.3.3.1 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles sintering | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_2$ evaluatio activities ze evolution nanopartic ze evolution $^{3}e_2O_4$ nanop particles at at early | owder ₂ O ₄ powc n vation-ba n of sinte cles n of sinte particles t three sta stage | ler of ased ered ered ages of | 117 121 126 126 131 134 134 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering 5.3.3.1 5.3.3.2 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles sintering Particles a | $aFe_{12}O_{19} p_{10}$ $i_{10,5}Zn_{0,5}Fe_{2}$ evaluation activities ze evolution ranoparticize evolution $i^{2}e_{2}O_{4}$ nanop particles at at early at intermed | owder 2O4 powe n vation-ba n of sinte cles n of sinte particles t three sta stage iate stag | ler of ased ered ered ages of e of | 117 121 126 126 131 134 134 134 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering 5.3.3.1 5.3.3.2 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles sintering Particles a sintering | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_{2}$ evaluatio active les ze evolution nanopartic ze evolution $Ge_{2}O_{4}$ nanop particles at at early at intermed | owder 2O4 powe n vation-ba n of sinte les n of sinte particles t three sta stage iate stag | ler of ased ered ered ages of e of | 117 121 126 126 131 134 134 134 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Characo of sintering 5.3.3.1 5.3.3.2 5.3.3.3 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles sintering Particles a sintering Particles | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_2$ evaluatio acti- les ze evolution nanopartic ze evolution Fe_2O_4 nanop particles at at early it intermed at final | owder 2O4 powe n vation-ba n of sinte les n of sinte particles t three sta stage iate stag stage | ler of ased ered ages of e of of | 117 121 126 126 131 134 134 134 136 138 |
| 5.3.2 | 5.3.1.2 Morpholog mechanoch synthesized 5.3.2.1 5.3.2.2 The Charac of sintering 5.3.3.1 5.3.3.2 5.3.3.2 | Sintered B Sintered N ical emical nanopartic Particle siz BaFe ₁₂ O ₁₉ Particle siz Ni _{0.5} Zn _{0.5} F teristics of Particles sintering Particles a sintering Particles sintering | $aFe_{12}O_{19} p_{10,5}Zn_{0,5}Fe_2$ evaluatio activities ze evolution nanopartic ze evolution e_2O_4 nanop particles at at early it intermed at final | owder 2O4 powe n vation-ba n of sinte les n of sinte particles t three sta stage iate stag stage | ler of ased ered ages of e of of | 117 121 126 131 134 134 136 138 |

5.3

| 5.3. | 5 Magnetic behaviour evaluation of bottom-up approach synthesized nanoparticles | 141 |
|------------|--|-----|
| | 5.3.5.1 Magnetic parameters of sintered BaFe ₁₂ O ₁₉ nanoparticles | 141 |
| | 5.3.5.2 Magnetic parameters of sintered $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 144 |
| 5.3. | 6 Electron Spin Resonance (ESR) of mechanochemical activation-based synthesized BaFe ₁₂ O ₁₉ and Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ nanoparticles | 146 |
| 5.3. | 7 Optical bandgap evaluation of mechanochemical activation-based synthesized $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 148 |
| CONCLUS | IONS AND SUGGESTIONS | 151 |
| 6.1 Intr | oduction | 151 |
| 6.2 Con | clusion | 151 |
| 6.2. | 1 Top-down approach mechanochemical process | 151 |
| 6.2. | 2 Mechanochemical activation-based synthesis | 153 |
| 6.3 Sug | gestions | 154 |
| ERENCES | | 155 |
| ENDICES | | 165 |
| DATA OF ST | UDENT | 168 |

169

REF APP BIODATA OF STUDENT LIST OF PUBLICATIONS

LIST OF TABLES

| Table | | Page |
|-------|---|------|
| 2.1 | Table of summary of previous studies on the development from conventional solid state route to mechanochemical activation-based synthesis | 12 |
| 2.2 | Table of summary of previous studies on top-down approach mechanochemical process | 14 |
| 3.1 | Magnetic Quantities and Units | 28 |
| 3.2 | Classification of ferrites according to variation in molar ratio of Fe_2O_3 to modifier oxide. (Modified from Louh et al., 2004) | 32 |
| 3.3 | Cation sublattices in BaFe ₁₂ O ₁₉ | 33 |
| 3.4 | Formula of anisotropy field for different easy axes | 38 |
| 3.5 | Summary of diffusion mechanisms for sintering (Rahaman, 2007) | 46 |
| 4.1 | Milling variables for top-down synthesis approach samples | 51 |
| 4.2 | Milling variables for the preparation of nanoparticles for mechanochemical activation-based synthesis | 53 |
| 4.3 | Error estimation for characterization measurements | 65 |
| 5.1 | Lattice constants, theoretical density, experimental density, and percentage of porosity of the starting material of top-down approach synthesis of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 68 |
| 5.2 | Rietveld refinement factors, lattice constants, and unit cell volume of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ at increasing milling times (Errors are shown in parentheses) | 75 |
| 5.3 | Site occupancies of the cations obtained by Rietveld refinement of $BaFe_{12}O_{19}$ | 77 |
| 5.4 | Site occupancies of cations and anions of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ as determined from Rietveld analysis | 77 |
| 5.5 | Bond angle information from Rietveld refinement of | 78 |

| | $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | |
|------|---|-----|
| 5.6 | Bond length information from Rietveld refinement of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 79 |
| 5.7 | Comparison of average crystallite size calculated using Scherrer's method and average particle size measured from FESEM images for $BaFe_{12}O_{19}$ nanoparticles | 82 |
| 5.8 | Comparison of average crystallite size calculated using Scherrer's method and average particle size measured from FESEM images for Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ nanoparticles | 85 |
| 5.9 | Average crystallite size, magnetization at 10 kOe (M_{10} $_{kOe}$), coersive field (H_c), and remanent magnetization (M_r) derived from the hysteresis loops measured for BaFe ₁₂ O ₁₉ milled at different milling time (t_m) | 101 |
| 5.10 | Average crystallite size, magnetization at 10 kOe (M_{10} _{kOe}), coersive field (H_c), and remanent magnetization (M_r) derived from the hysteresis loops measured for Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ milled at different milling time (t_m) | 104 |
| 5.11 | Parameters that extracted from ESR spectra for BaFe ₁₂ O ₁₉ Nanoparticles | 106 |
| 5.12 | Parameters that extracted from ESR spectra for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ Nanoparticles | 115 |
| 5.13 | Variation of direct optical bandgap values of hard and soft ferrite at different milling time | 120 |
| 5.14 | Rietveld refinement factors, and lattice constants of $BaFe_{12}O_{19}$ at increasing sintering temperature (Errors are shown in parentheses) | 179 |
| 5.15 | Structural Information of $BaFe_{12}O_{19}$ extracted from Rietveld refinement at increasing sintering temperature (Errors are shown in parentheses) | 120 |
| 5.16 | Rietveld refinement factors, and lattice constants of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ at increasing sintering temperature (Errors are shown in parentheses) | 125 |
| 5.17 | Structural Information of Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ extracted from Rietveld refinement at increasing sintering | 125 |
| 5.18 | Average crystallite size calculated using Scherrer's method and average particle size measured by FESEM | 126 |
| | | |

| | technique for mechanochemical activation-based synthesis of $BaFe_{12}O_{19}$ nanoparticles | |
|------|---|-----|
| 5.19 | Average crystallite size calculated using Scherrer's method and average particle size measured by FESEM microscopy technique for mechanochemical activation-based synthesis of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 131 |
| 5.20 | Average crystallite size, magnetization at 10 kOe (M_{10} $_{kOe}$), coersive field (H_c), and remanent magnetization (M_r) derived from the hysteresis loops measured for BaFe ₁₂ O ₁₉ sintered at different sintering temperature | 142 |
| 5.21 | Average crystallite size, magnetization at 10 kOe (M_{10} koe), coersive field (H_c), and remanent magnetization (M_r) derived from the hysteresis loops measured for Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ sintered at different sintering temperature | 145 |
| 5.22 | Parameters that extracted from ESR spectra for $BaFe_{12}O_{19}$ nanoparticles sintered at different sintering temperature | 146 |
| 5.23 | Parameters that extracted from ESR spectra for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at different sintering temperature | 147 |

LIST OF FIGURES

| Figure | | Page |
|--------|--|------|
| 2.1. | Schematic representation of Bottom-up Approach Synthesis | 9 |
| 2.2. | Schematic representation of top-down approach synthesis for the preparation of nanostructured materials | 13 |
| 2.3 | Nucleation and growth mechanisms of direct mechanochemical synthesis (Sopicka-Lizar, 2010) | 15 |
| 2.4 | Free Energy diagram for phases involved in mechanical alloying (MA) and mechanical disordering (MD) process (El-Eskandarany, 2015d) | 22 |
| 3.1 | Schematic of magnetic moment alignment for: (a) diamagnetic material, (b) paramagnetic material with and without an external applied field | 29 |
| 3.2 | Superparamagnetic particles with and without an external applied field | 31 |
| 3.3 | Unit cell of $BaFe_{12}O_{19}$ showing polyhedra coordination of Fe^{3+} ions (Valenzuela, 1994) | 34 |
| 3.4 | The spinel structure. A unit cell can be divided into octants; tetrahedral cations A and octahedral cations B, are shown in two octants with unit cell edge a (Valenzuela, 1994) | 35 |
| 3.5 | Schematic representations of reduction of magnetostatic energy by subdivision of magnetic domains. The dashed lines represent the domain walls | 39 |
| 3.6 | Schematic representation of typical hysteresis loop, the darken regions in the bubbles represent domains with parallel spin orientation. Initial magnetization curve showing parameters like initial permeability μ i, and critical field Hcr (adapted from Bertotti, 1998; Valenzuela, 1994) | 41 |
| 3.7 | Comparison of hard and soft hysteresis loops | 42 |
| 3.8 | Typical procedure of sintering process | 43 |

 \bigcirc

| 2 | 3.9 | The basic phenomena of sintering that involve changes of specific surface energy and interfacial surface area (Kang, 2005) | 44 |
|---|------|--|----|
| 3 | 3.10 | Six matter transport mechanisms of sintering (Rahaman, 2007) | 46 |
| 2 | 3.11 | Three stages of mechanochemical process in terms of dispersity; (a) Rittinger stage, (b) aggregation stage, and (c) agglomeration stage (Balaz, 2008) | 48 |
| 2 | 4.1 | Flow chart for the starting bulk material preparation, experiment, and characterization for top-down approach mechanochemical process synthesized samples | 52 |
| 2 | 4.2 | Flow chart for the starting material preparation, experiment, and characterization for mechanochemical activation-based synthesis | 54 |
| 2 | 4.3 | Sintering profile of mechanochemical activation- based synthesis | 55 |
| 2 | 4.4 | Schematic diagram of XRD (Cullity & Stock, 2001) | 56 |
| 2 | 4.5 | Schematic diagram of HRTEM (modified from Yang, 2008) | 58 |
| 2 | 4.6 | Schematic diagram of the structure of FESEM (Yang, 2008) | 60 |
| 2 | 4.7 | Schematic representation of model for the Kubelka- Munk analysis (Hecht, 1976) | 62 |
| 2 | 4.8 | The set-up of VNA measurements | 64 |
| | 4.9 | Scattering parameter (S parameters) description of two-pot device | 64 |
| | 5.1 | XRD pattern for the starting material of top-down approach synthesis of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 67 |
| | | | |

xix

| 5.2 | (a) FESEM micrograph of the starting material of $BaFe_{12}O_{19}$; (b) Grain size distribution curve and histogram of the starting material top-down approach synthesis of $BaFe_{12}O_{19}$ nanoparticles; (c) FESEM micrograph of the starting material of $Ni_{0.5}Zn_{0.5}Fe_2O_4$; (d) Grain size distribution curve and histogram of the starting bulk material of top-down approach synthesis of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 69 |
|------|--|----|
| 5.3 | X-ray Diffraction patterns of (a) $BaFe_{12}O_{19}$ and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ after different milling times: 1, 2, 4, 8, 12, 16, 20 hours | 71 |
| 5.4 | Rietveld refined XRD patterns for $BaFe_{12}O_{19}$ samples milled for (a) 1 hour, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, (g) 20 hours. The bottom graphs show the difference in the plots between the experimental and refined data | 73 |
| 5.5 | Rietveld refined XRD patterns for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ samples milled for (a) 1 hour, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, (g) 20 hours. The bottom graphs show the difference in the plots between the experimental and refined data | 74 |
| 5.6 | Graphs of (a) lattice constants, a and c, against milling time, (b) unit cell volume against milling time for $BaFe_{12}O_{19}$, and (c) lattice parameter and unit cell volume against milling time for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ | 76 |
| 5.7 | $\begin{array}{c} Crystallite \ size \ and \ lattice \ strain \ as \ function \ of \\ milling \ time \ for \ (a) \ BaFe_{12}O_{19} \ and \ (b) \\ Ni_{0.5}Zn_{0.5}Fe_2O_4 \end{array}$ | 81 |
| 5.8 | FESEM images for $BaFe_{12}O_{19}$ nanoparticles milled for (a) 1 hour, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, and (g) 20 hours | 83 |
| 5.9 | Particle size distribution for $BaFe_{12}O_{19}$ nanoparticles milled for (a) 1 hour, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, and (g) 20 hours | 84 |
| 5.10 | FESEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles milled for (a) 1 hours, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, and (g) 20 hours | 86 |
| 5.11 | Particle size distribution for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles milled for (a) 1 hour, (b) 2 hours, (c) 4 hours, (d) 8 hours, (e) 12 hours, (f) 16 hours, and (g) 20 hours | 87 |

XX

| 5.12 | Three stages during mechanochemical process for (a) $BaFe_{12}O_{19}$ and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ | 89 |
|------|---|-----|
| 5.13 | High Resolution TEM images for $BaFe_{12}O_{19}$ samples milled for 2 hours (early stage) | 91 |
| 5.14 | Schematic illustration of grain boundary orientation generating fringes in HRTEM images | 92 |
| 5.15 | High Resolution TEM images for Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ samples during early stage of mechanochemical process | 92 |
| 5.16 | High Resolution TEM images for BaFe ₁₂ O ₁₉ samples milled for 12 hours (intermediate stage) | 93 |
| 5.17 | High Resolution TEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ samples milled for 8 hours (intermediate stage) | 94 |
| 5.18 | High Resolution TEM images for $BaFe_{12}O_{19}$ samples milled for 20 hours (final stage) | 95 |
| 5.19 | High Resolution TEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ samples milled for 16 hours (final stage) | 96 |
| 5.20 | Relative percentage of crystalline and amorphous content of (a) hard ferrite and (b) soft ferrite at different milling times | 97 |
| 5.21 | Schematic drawings of the evolutional stages of mechanochemical process | 98 |
| 5.22 | M-H hysteresis loops for three stages of mechanochemical mechanism for BaFe ₁₂ O ₁₉ | 100 |
| 5.23 | The variation of magnetic parameters with particles size of $BaFe_{12}O_{19}$ nanoparticles | 101 |
| 5.24 | M-H hysteresis loops for Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ nanoparticles at different milling time | 103 |
| 5.25 | The variation of magnetic parameters with particle size of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 104 |
| 5.26 | Schematic representation of transition from single to multi domain particle (adapted from Hadjipanayis, 1999) | 105 |
| | | |

| | 5.27 | The definition of $A+$, $A-$, and ΔHpp (From ESR spectrum of milled 2 hours $BaFe_{12}O_{19}$ nanoparticles) | 105 |
|--|------|--|-----|
| | 5.28 | ESR spectra of (a) $BaFe_{12}O_{19}$ and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles milled at different milling time | 108 |
| | 5.29 | Plot of real part permeability against frequency for $BaFe_{12}O_{19}$ | 110 |
| | 5.30 | Plot of imaginary part permeability against frequency of $BaFe_{12}O_{19}$ | 111 |
| | 5.31 | Plot of real and imaginary part permeability against frequency for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 111 |
| | 5.32 | Absorption spectra of $BaFe_{12}O_{19}$ nanoparticles milled at different milling time | 113 |
| | 5.33 | The plot of crystallite size and optical bandgap value against milling time of BaFe ₁₂ O ₁₉ nanoparticles | 114 |
| | 5.34 | Schematic diagram of band structure of a core-shell nanoparticle (Adapted from Naldoni et al., 2012) | 114 |
| | 5.35 | Absorption spectra of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles milled at different milling time | 116 |
| | 5.36 | The plot of crystallite size and optical bandgap value against milling time of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles | 116 |
| | 5.37 | X-ray Diffraction patterns of BaFe ₁₂ O ₁₉ sintered from 600 °C to 1200 °C | 118 |
| | 5.38 | Rietveld refined XRD patterns for $BaFe_{12}O_{19}$ samples sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C. The bottom graphs show the difference in the plots between the experimental and refined data | 119 |
| | 5.39 | Crystallite size and lattice strain of $BaFe_{12}O_{19}$ as function of sintering temperature | 121 |
| | 5.40 | X-ray Diffraction patterns of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ sintered from 600 °C to 1200 °C | 122 |
| | 5.41 | Rietveld refined XRD patterns for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ samples sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C. The bottom graphs show the difference in the plots between the experimental and refined data | 124 |

| 5.42 | Crystallite size and lattice strain of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ as function of sintering temperature | 126 |
|------|---|-----|
| 5.43 | FESEM micrographs for $BaFe_{12}O_{19}$ nanoparticles sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C | 128 |
| 5.44 | Particle size distribution of $BaFe_{12}O_{19}$ nanoparticles sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C | 129 |
| 5.45 | Plot of log D versus the reciprocal of absolute temperature $(1/T)$ of BaFe ₁₂ O ₁₉ | 131 |
| 5.46 | FESEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C | 132 |
| 5.47 | Particle size distribution of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered for (a) 600 °C, (b) 700 °C, (c) 800 °C, (d) 900 °C, (e) 1000 °C, (f) 1100 °C, (g) 1200 °C | 133 |
| 5.48 | Plot of log D versus the reciprocal of absolute temperature $(1/T)$ of Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ | 134 |
| 5.49 | High Resolution TEM images for $BaFe_{12}O_{19}$ nanoparticles sintered at (a) 600 °C, (b) 700 °C, and (c) 800 °C (Initial stage of sintering) | 135 |
| 5.50 | High Resolution TEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at (a) 600 °C, and (b) 800 °C (Initial stage of sintering) | 136 |
| 5.51 | High Resolution TEM images for $BaFe_{12}O_{19}$ nanoparticles sintered at (a) 900 °C, (b) 900 °C (another feature), (c) 1000 °C (Intermediate stage of sintering) | 137 |
| 5.52 | High Resolution TEM images for $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at (a) 900 °C, and (b) 1100 °C (Intermediate stage of sintering) | 137 |
| 5.53 | High Resolution TEM images for particles sintered at 1200 °C for (a) $BaFe_{12}O_{19}$, and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ (final stage of sintering) | 138 |
| 5.54 | Schematic representation of bulk and nanoparticles, and the definition of R, radius of a particle, and r, radius of the core of a particle | 139 |

| 5.55 | Relative percentage of crystalline and amorphous content of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ at different sintering temperature | 140 |
|------|--|-----|
| 5.56 | Three stages of sintering | 141 |
| 5.57 | Magnetic parameters of mechanochemical activation- based synthesis of $BaFe_{12}O_{19}$: (a) Hysteresis loops at different sintering temperature, (b) Plot of M_{10kOe} versus sintering temperature, (c) Plot of coercivity versus sintering temperature | 142 |
| 5.58 | Coercivity of $BaFe_{12}O_{19}$ samples as a function of average particle size | 143 |
| 5.59 | Magnetic parameters of mechanochemical activation- based synthesis of Ni _{0.5} Zn _{0.5} Fe ₂ O ₄ : (a) Hysteresis loops at different sintering temperature, (b) Plot of M_{10kG} versus sintering temperature, (c) Plot of coercivity versus sintering temperature | 144 |
| 5.60 | Coercivity of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ samples as a function of average particle size | 145 |
| 5.61 | ESR spectra of (a) $BaFe_{12}O_{19}$, and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at different temperature | 148 |
| 5.62 | Absorption spectra of (a) $BaFe_{12}O_{19}$, and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at different sintering temperature | 149 |
| 5.63 | The plot of crystallite size and optical bandgap value against sintering temperature of (a) $BaFe_{12}O_{19}$, and (b) $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles sintered at different temperature | 150 |
| | - | |

LIST OF ABBREVIATIONS AND SYMBOLS

| Barium hexaferrite |
|--------------------------------------|
| Nickel zinc ferrite |
| Bottom-up approach |
| Top-down approach |
| Zinc ferrite |
| Cobalt ferrite |
| Barium carbonate |
| Nickel ferrite |
| Nickel zinc ferrite |
| Strontium ferrite |
| Figure |
| Zinc oxide |
| Iron oxide |
| Nickel oxide |
| Ball-to-powder weight ratio |
| Other similar things |
| Magnetocrystalline anisotropy energy |
| Saturation magnetization |
| Porosity |
| Bohr magneton |
| Density |
| Magnetic susceptibility |
| Magnetostriction |
| Anisotropy energy |
| |

| E_m | Magnetostatic energy |
|----------------|--|
| H_c | Coercivity |
| k_B | Boltzmann constant |
| γ | Specific surface energy |
| XRD | X-ray Diffraction |
| HRTEM | High Resolution Transmission Electron Microscope |
| VSM | Vibrating sample magnetometer |
| VNA | Vector network analyser |
| ESR | Electron spin resonance |
| FESEM | Field Emission Scanning Electron Microscope |
| NIR-UV-vis | Near infrared-ultraviolet- visible |
| h k l | Miller indices |
| λ | Wavelength |
| ~ | Approximately |
| mins | Minutes |
| В | Magnetic flux density |
| DUT | Device under test |
| μ _r | Complex permeability |
| px-ray | X-ray density |
| D | Crystallite size |
| 20 | 2 theta degree |
| a.u. | Arbitrary unit |
| | Goodness of fit |
| $M_{10 \ kOe}$ | Magnetization at 10 kOe |
| t_m | Milling time |
| | |

| W | Volume fraction |
|--------------------|-------------------------------|
| V _{shell} | Volume of shell |
| V _{core} | Volume of core |
| ΔH_{pp} | Peak-to-peak line width |
| R | Asymmetry parameter |
| H_r | Resonance field |
| h | Plank constant |
| k _B | Boltzmann constant |
| μ' | Real part of the permeability |
| μ'' | Loss factor |
| α | Absorbance |
| Eg | Energy bandgap |
| v | Frequency |
| E _a | Activation energy |
| D_c | Critical size |



CHAPTER 1

INTRODUCTION

1.1 Background of the study

Concerns with regard to the hazardous effects of current technologies on the environment and climate change are forcing industrialized and developing countries to seek for solutions and green technologies for a sustainable future. A more efficient utilisation of resources and energy implies a reduction in both waste and the environmental impact of human activities (Holdren, 2008). Therefore, rethinking the known current methodologies while maintaining or even improving productivity has become the major role of advancing fundamental and applied sciences. Mechanochemical procedure or mechanochemistry technology has attracted considerable interest because: (1) it is easy and simple to apply, (2) it allows fast chemical reactions under controlled conditions, and (3) it is a highly productive and relatively efficient methodology for the production of nanoscale materials. Therefore, mechanochemical procedure can be used as either a top-down approach synthesis technique to convert used bulk materials into raw nanopowder, or as a mechanochemical activation-based synthesis, which referring to the utilisation of mechanochemical process to increase the reactivity of the starting powders before heat treatment.

Ceramic composite powders can be synthesized by two techniques, which are known as the bottom-up and top-down approaches. The bottom-up approach synthesis method involves the construction of nanostructures in the material from small to large sizes. Examples of bottom-up approach synthesis methods include the sol-gel, melt spinningmelt quenching (MQ), chemical vapour deposition (CVD), and physical vapour deposition (PVD) methods. Typically, researchers would investigate variations in the properties of materials with a specific bottom-up synthesis technique with controlled parameters to obtain a controlled microstructure and fully dense polycrystalline material. On the other hand, the top-down approach synthesis method utilizes mechanical, chemical, or other forms of energy to break down macro-structured materials into smaller components (Sopicka-Lizer, 2010). The emergence of "Green Technology" recently has highlighted the importance of top-down approach synthesis method studies, which have been neglected by researchers in the past as the issues of conservation and preservation were not as urgent as at present. Nanomagnetic materials have attracted considerable interest from many researchers due to their novel properties when particles size decreases to nanoscale regime. These properties are different from those in bulk form. The correlation between particle size (1-100 nm) and critical magnetic parameters has led to a new study field called nanomagnetism. Nanostructured materials possess a high surface area to volume ratio, which allows the dominance of quantum confinement effects and a larger influence of surface atoms compared to those in the interior (Kumar, 2013).

Ferrites are a family of materials, in which the main constituent of the material is mixed metal oxides, typically iron oxide which contains Fe³⁺ ions. Ferrites are categorized into two groups based on how ease these materials are to be magnetized or demagnetized. Materials that can be permanently magnetized and require strong applied magnetic field to demagnetize are known as hard ferrites. Hard ferrites are widely used in speakers, recording devices, magneto-optical sensors. On the other hand, materials that are easily and temporarily magnetised under magnetic field are known as soft ferrites. In contrast with hard ferrites, soft ferrites are used in communication and electronic devices like transformers and inductors, and recently, in the field of biomedicine. Ferrites can also be categorized according to their structures into three groups: (1) spinel ferrites, (2) garnet ferrites, and lastly (3) hexagonal ferrites, which are known as magnetoplumbite ferrites. In terms of magnetic characteristics, spinel ferrites and garnet ferrites are soft ferrites and hexagonal ferrites are hard ferrites. Both hard and soft ferrites play important roles in various current technologies. Hard ferrites, particularly BaFe12O19, barium hexaferrite, exhibits some salient properties. It has strong magnetocrystalline anisotropy and an easy magnetization at the *c*-axis. Besides, barium hexaferrite has a high coercivity, high saturation magnetization, excellent chemical stability, and is resistant to corrosion. Due to its unique characteristics, barium hexaferrite is one of the most important magnetic materials with great scientific and technological roles (Shafie et al., 2014). On the other hand, nickel zinc ferrite, which is the most popular composition of soft ferrites, is one of the most abundant magnetic materials found in electrical devices and telecommunication devices. Characteristics like high resistivity, low eddy current loses, low coercivity, low cost, and easily altered magnetic behaviours due to its compositional sensitive nature, has made nickel zinc ferrite an important material in high frequency applications such as microwave devices, transformers, antennas, and inductors (Ibrahim et al., 2014; Ismail et al., 2011). In this study, BaFe₁₂O₁₉ and a well-known composition $Ni_{0.5}Zn_{0.5}Fe_2O_4$ are chosen as the specimens.

1.2 Relationship between Characteristics and Mechanically Induced Microstructural Response of Ferrites

The response of the structure and size of hexaferrites and spinel ferrites to mechanical energy through high energy ball milling and its impacts on their optical, physical, and magnetic properties are interesting because mechanical energy makes them differ from their bulk counterparts. Investigations in the field of mechanically induced materials, especially iron-based ceramics, have been considerably developed recently (Rodziah et al., 2012; Shafie et al., 2014). Šepelák et al. (2014), and Waje et al. (2010) recognised mechanochemical process was capable of producing nano-sized powder, and the changes in the microstructural properties were the main factor responsible for the changes in the investigated properties. Ferrites exhibit complicated disordering phenomena under mechanical impacts. Šepelák et al. (2014) found several disordering phenomena such as redistribution of cations over non-equivalent cation sublattices, the formation of canted spin arrangements, the changes of polyhedra geometry, and formation of cation with unsaturated oxygen coordination. Heat treatment like sintering provides a recovery path or recrystallization process: to enable the excited unstable metastable state transforms to the low energy crystalline state (Idza et al., 2012). The reformation of crystalline phase in ferrites changed the behaviours or characteristics of

mechanical alloyed induced metastable materials in terms of microstructural properties, caused an increase in crystalline volume, thus had a direct relationship with the investigated physical, chemical, and magnetic properties (Low et al., 2015). On the other hand, for top-down approach, mechanically induced microstructural defects and disordering structures remained in ferrite specimens (Šepelák et al., 2014). Studies on the mechanically induced response of both hard and soft ferrites, are essential not only for fundamental understanding of science, but also due to the industrial and technological importance of these materials in telecommunication, microwave, memory storage, ferrofluids, and even biomedical applications. To strengthen the fundamental science knowledge on the evolutional relationship between characteristics and microstructural response of ferrites, this study undertakes the response of fine nanoparticles made up of mechanical alloyed Ba-hexaferrite and NiZn-spinel ferrite to vary in the sintering temperature (mechanochemical activation-based synthesis), and further investigate the correlations by comparing with the response of single phase Bahexaferrite and NiZn-spinel ferrite to mechanical action through mechanochemical process (top-down approach mechanochemical process). For both approaches, their physical, optical, and magnetic properties are investigated. Many researchers involving, but not restricted to the following (Rodziah et al., 2012; Šepelák et al., 2014; Shafie et al., 2014; Waje et al., 2010) have performed investigations on nano-sized ferrites. However, studies of systematic mechanochemical activation-based synthesis and the production of nanoparticles via mechanochemical process are very scarce in literature.

1.3 Problem statement

Mechanochemical process is one of the promising candidates of 'green processes' which can be used to develop methods which minimise damage to the environment. However, extensive research had been merely carried out on sample synthesized by mechanochemical procedure by neglecting the parallel evolution of microstructure and material properties at various intermediate controlling process factors. Therefore, much of the essential information of process mechanisms has been neglected, thus reducing the capability of attaining good fundamental scientific knowledge which lies behind the parallel evolution of the microstructural-material properties. Characteristics or behaviours of hard and soft ferrites are directly related to the microstructural properties, which are strongly dependant on the preparation route, therefore, the evolutional relationship between microstructural properties with controlling process factors has to be investigated. This study will carefully track the fundamental evolution of hard and soft ferrites are listed as:

- 1. What are the characteristics of materials synthesized by mechanochemical activation-based synthesis and top-down approach mechanochemical process?
- 2. What is the relationship of evolving microstructure properties with optical and magnetic properties of material?
- 3. What would be the structural properties of bulk materials after the top-down approach mechanochemical process?

4. What would be the unique characteristics possessed by materials synthesized by the top-down approach that would make the recycling of used materials possible?

1.4 Objectives and hypotheses of the study

The main objective of this study is to investigate and compare the parallel evolution of microstructural, optical bandgap, physical and magnetic properties of two different approaches of mechanochemical procedure: (1) top-down approach mechanochemical process (at different milling time) and (2) mechanochemical activation-based synthesis (at different sintering temperature). The achieved goals from this research work can be utilised to develop fundamental knowledge on mechanochemical procedure. Furthermore, the understanding on the parallel evolution of the microstructure and various properties of the materials will help to develop a general theoretical model for future studies. In this research, the work-step objectives are presented as below:

- 1. To prepare $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles using top-down approach by breaking bulk materials via mechanochemical process.
- 2. To study the microstructure-optical and magnetic properties of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles as a consequence of milling time.
- 3. To prepare BaFe₁₂O₁₉ and Ni_{0.5}Zn_{0.5}Fe₂O₄ using mechanochemical activationbased synthesis by pre-treating the starting powders with mechanochemical process before sintering.
- 4. To study the microstructure-optical and magnetic properties of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$ nanoparticles as a consequence of sintering temperature.

Thus, according the above main objectives, this study is hypothesized as follows:

- 1. Top-down approach mechanochemical process would induce defects or amorphous phases in the $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$, while effectively decrease the size of the ferrites.
- 2. The existence of amorphous and crystalline mixture state with nanocrystalline microstructure would deteriorate the optical and magnetic properties.
- 3. Mechanochemical activation based synthesis would increase the reactivity of the starting powders, thus reduce the sintering temperature for the formation of single phase ferrites.
- 4. The parallel evolving of microstructural properties with elevating sintering temperature of polycrystalline BaFe₁₂O₁₉ and Ni_{0.5}Zn_{0.5}Fe₂O₄ would affect the optical and magnetic properties of the materials.

1.5 Scope of the Study

This study will focus on synthesizing barium hexaferrite ($BeFe_{12}O_{19}$) and nickel zinc ferrite ($Ni_{0.5}Zn_{0.5}Fe_2O_4$) powders via two different mechanochemical procedures in their nominal composition (Costa et al., 2003; Pullar, 2012). These ferrites with their nominal compositions have increasing degree of interest due to their importance in commercial and technology. Key mechanisms associated with both preparation routes have been studied. In particular, a considerable investigation has been carried out to understand the microstructural, magnetic, and optical properties of $BaFe_{12}O_{19}$ and $Ni_{0.5}Zn_{0.5}Fe_2O_4$.

1.6 Thesis outline

Chapter One comprehensively describes the general introduction of hard and soft ferrites, mechanochemistry, the difference between top-down and bottom-up approaches, correlation between microstructural properties and various properties, and some research questions. In Chapter Two, related literature reviews about previous studies on synthesis techniques, mechanochemical process or high energy ball milling with its optimization of variable parameters, and the effects of particle size and microstructural properties are summarised. Chapter Three describes hard and soft ferrites, mechanical alloying mechanisms, magnetism, particle size related properties in terms of fundamental theories. Chapter Four focuses on employed methodologies for sample preparation and equipment and measurement involved in the characterisation of the study. Chapter Five presents obtained data and results of current research work, followed by precise discussion. Chapter Six concludes the research findings. A research summary was made, followed by future research recommendations. At last, Chapter Six was attached with references, appendix, and a list of publications.

REFERENCES

- Abbas, S. I., John, H. T., & Fraih, A. J. (2017). Preparation of Nano Crystalline Zinc Ferrite as Material for Micro Waves Absorption by Sol-Gel Methods. *Indian Journal of Science and Technology*, 10(21), 1–6.
- Ashima, Sanghi, S., Agarwal, A., & Reetu. (2012). Rietveld refinement, electrical properties and magnetic characteristics of Ca-Sr substituted barium hexaferrites. *Journal of Alloys and Compounds*, *513*, 436–444.
- Balaz, P. (2008). *Mechanochemistry in Nanoscience and Minerals Engineering*. Berlin: Springer.
- Baláž, P., Achimovičová, M., Baláž, M., Billik, P., Cherkezova-Zheleva, Z., Criado, J. M., Delugo, D., Dutkova, E., Gaffet, E., Gotor, F. J., Kumar, R., Mitor, I., Rojac, T., Senna, M., Streletskii, A., & Wieczorek-Ciurowa, K. (2013). Hallmarks of mechanochemistry: from nanoparticles to technology. *Chemical Society Reviews*, 42(18), 7571.
- Baro, M. D., Kolobov, Y. R., Ovid'ko, I. A., Schaefer, H. E., Straumal, B. B., Valiev, R. Z., Alexandrov, I. V., Ivanov, M., Reimann, K., Reizis, A. B., Surinach, S., & Zhilyaev, A. P. (2001). Diffusion and Related Phenomena in Bulk Nanostructured Materials. *Reviews on Advanced Materials Science*, 2, 1–43.
- Baykal, A., Auwal, I. A., Güner, S., & Sözeri, H. (2017). Magnetic and optical properties of Zn²⁺ ion substituted barium hexaferrites. *Journal of Magnetism and Magnetic Materials*, 430, 29–35.
- Benito, G., Morales, M. P., Requena, J., Raposo, V., Vázquez, M., & Moya, J. S. (2001). Barium hexaferrite monodispersed nanoparticles prepared by the ceramic method. *Journal of Magnetism and Magnetic Materials*, 234(1), 65–72.
- Bera, J., & Roy, P. K. (2005). Effect of grain size on electromagnetic properties of Ni_{0.7}Zn_{0.3}Fe₂O₄ ferrite. *Physica B: Condensed Matter*, 363(1–4), 128–132.
- Bertotti, G. (1998). Hysteresis in Magnetism: For Physicists, Materials Scientists, and Engineers. United Kingdom: Academic Press.
- Cividanes, L. S., Campos, T. M. B., Rodrigues, L. A., Brunelli, D. D., & Thim, G. P. (2010). Review of mullite synthesis routes by sol-gel method. *Journal of Sol-Gel Science and Technology*, 55(1), 111–125.
- Costa, A. C. F. M., Tortella, E., Morelli, M. R., & Kiminami, R. H. G. A. (2003). Synthesis, microstructure and magnetic properties of Ni – Zn ferrites, 256, 174– 182.

- Cullity, B. D., & Stock, S. R. (2001). *Elements of X-ray diffraction. Prentice Hall* (3rd ed.). United Kingdom: Pearson.
- Da Silva, K. L., Menzel, D., Feldhoff, A., Kübel, C., Bruns, M., Paesano, A., & Šepelák, V. (2011). Mechanosynthesized BiFeO₃ Nanoparticles with Highly Reactive Surface and Enhanced Magnetization. J. Phys. Chem. C, 115(15), 7209– 7217.
- Dho, J., Lee, E. K., Park, J. Y., & Hur, N. H. (2005). Effects of the grain boundary on the coercivity of barium ferrite BaFe₁₂O₁₉. *Journal of Magnetism and Magnetic Materials*, 285(1–2), 164–168.
- Díaz-Pardo, R., & Valenzuela, R. (2015). Characterization of Magnetic Phases in Nanostructured Ferrites by Electron Spin Resonance. In *Advanced Electromagnetic Waves* (pp. 210–237).
- Dippong, T., Cadar, O., Levei, E. A., Bibicu, I., Diamandescu, L., Leostean, C., Lazar, M., Borodi, G., & Barbu Tudoran, L. (2017). Structure and magnetic properties of CoFe₂O₄/SiO₂ nanocomposites obtained by sol-gel and post annealing pathways. *Ceramics International*, 43(2), 2113–2122.
- Dixit, G., Pal Singh, J., Srivastava, R. C., & Agrawal, H. M. (2012). Magnetic resonance study of Ce and Gd doped NiFe₂O₄ nanoparticles. *Journal of Magnetism and Magnetic Materials*, 324(4), 479–483.
- El-Eskandarany, M. S. (2015a). 1 Introduction. In *Mechanical Alloying* (pp. 1–12).
- El-Eskandarany, M. S. (2015b). 6 Mechanically induced solid state reduction. In *Mechanical Alloying* (pp. 132–151).
- El-Eskandarany, M. S. (2015c). 8 Reactive ball milling for fabrication of metal nitride nanocrystalline powders. In *Mechanical Alloying* (pp. 182–201).
- El-Eskandarany, M. S. (2015d). *Mechanical alloying: nanotechnology, materials science and powder metallurgy.*
- Fox, M. (2010). *Optical Properties of Solids* (2nd ed.). United States: Clarendon Press Oxford.
- Frenkel, J., & Doefman, J. (1930). Spontaneous and Induced Magnetisation in Ferromagnetic Bodies. *Nature*, 126(3173), 274–275.
- Fultz, B., & Howe, J. M. (2008). *Transmission Electron Microscopy and Diffractometry of Materials, 3rd Edition* (3rd ed.). New York: Springer.
- Gaikward, A., Navale, S., Samuel, V., Murugan, A., & Ravi, V. (2006). A coprecipitation technique to prepare $BiNbO_4$, $MgTiO_3$ and $Mg_4Ta_2O_9$ powders. *Materials Research Bulletin*, 41(2), 347–353.

- Gleiter, H. (1989). Nanocrystalline Materials. Progress in Materials Science, 33, 223– 315.
- Goldman, A. (2006). *Modern Ferrite Technology (2nd ed.)*. Pittsburgh, PA, USA: Springer.
- Gonzalez, G., D'Angelo, L., Ochoa, J., Lara, B., & Rodriguez, E. (2002). The Influence of Milling Intensity on Mechanical Alloying. *Material Science Forum*, 388, 159– 164.
- Gul, I. H., Ahmed, W., & Maqsood, A. (2008). Electrical and magnetic characterization of nanocrystalline Ni-Zn ferrite synthesis by co-precipitation route. *Journal of Magnetism and Magnetic Materials*, 320(3–4), 270–275.
- Hadjipanayis, G. C. (1999). Nanophase hard magnets. Journal of Magnetism and Magnetic Materials, 200(1-3), 373–391.
- Hajalilou, A., Hashim, M., Ebrahimi-kahrizsangi, H., Kamari, H., & Sarami, N. (2014). Synthesis and structural characterization of nano-sized nickel ferrite obtained by mechanochemical process. *Ceramics International*, 40(4), 5881–5887.
- Harringa, J. L., Cook, B. A., & Beaudry, B. J. (1992). Effects of vial shape on the rate of mechanical alloying in Si₈₀Ge₂₀. *Journal of Materials Science*, 27, 801–804.
- Hecht, H. G. (1976). The Interpretation of Diffuse Reflectance Spectra. J. Res. NBS A Phys. Ch., 80(4), 567–583.
- Hench, L. L. (1998). 2 Sol-Gel Kinetics. In Sol-Gel Silica (pp. 8-23).
- Holdren, J. P. (2008). Science and Technology for Sustainable Well-Being. Science, 319, 424–434.
- Huang, J. Y. (1999). HRTEM and EELS Studies of Defects Structure and Amorphouslike Graphite by Ball Milling, 47(6), 1801–1808.
- Ibrahim, I. R., Hashim, M., Nazlan, R., Ismail, I., Wan Ab Rahman, W. N., Abdullah, N. H., Idris, F. M., Shafie, M. S. E., & Muhamad Zulkimi, M. M. (2014). Grouping trends of magnetic permeability components in their parallel evolution with microstructure in Ni_{0.3}Zn_{0.7}Fe₂O₄. *Journal of Magnetism and Magnetic Materials*, 355, 265–275.
- Idza, I. R., Hashim, M., Rodziah, N., Ismayadi, I., & Norailiana, A. R. (2012). Influence of evolving microstructure on magnetic-hysteresis characteristics in polycrystalline nickel – zinc ferrite, Ni_{0.3}Zn_{0.7}Fe₂O₄. *Materials Research Bulletin*, 47(6), 1345–1352.

- Igarashi, H., & Okazaki, K. (1977). Effects of Porosity and Grain Size on the Magnetic Properties of NiZn Ferrite. *Journal of the American Ceramic Society*, 60(1–2), 51–54.
- Ismail, I., Hashim, M., Amin, K., Alias, R., & Hassan, J. (2011). Journal of Magnetism and Magnetic Materials Milling time and BPR dependence on permeability and losses of. *Journal of Magnetism and Magnetic Materials*, 323(11), 1470–1476.
- Ismail, I., Hashim, M., Matori, K. A., Alias, R., & Hassan, J. (2012). Dependence of magnetic properties and microstructure of mechanically alloyed Ni_{0.5}Zn_{0.5}Fe₂O₄ on soaking time. *Journal of Magnetism and Magnetic Materials*, 324(16), 1–8.
- Ismail, I., Hashim, M., Khamirul, A. M., & Alias, R. (2009). The Effect of Milling Time on Ni_{0.5}Zn_{0.5}Fe₂O₄ Compositional Evolution and Particle Size Distribution. *American Journal of Applied Sciences*, 6(8), 1553–1558.
- Ismail, I., Hashim, M., Matori, K. A., Alias, R., & Hassan, J. (2012). The transition from paramagnetic to ferromagnetic states as influenced by evolving microstructure of Ni_{0.5}Zn_{0.5}Fe₂O₄. *Journal of Superconductivity and Novel Magnetism*, 25(1), 71–77.
- Jacobo, S. E., Domingo-Pascual, C., Rodriguez-Clemnte, R., & Blesa, M. A. (1997). Synthesis of ultrafine particles of barium ferrite by chemical coprecipitation. *Journal of Materials Science*, 32(4), 1025–1028.
- Jarcho, M., Bolen, C. H., Thomas, M. B., Bobick, J., Kay, J. F., & Doremus, R. H. (1976). Hydroxylapatite Synthesis and Characterization in Sense Polycristalline Forms. *J. Mater. Sci.*, *11*, 2027–2035.
- Job, A., Siqueira, A. De, Danna, C. S., Bellucci, F., Cabrera, F. C., & Silva, L. E. K. (2014). Utilization of Composites and Nanocomposites Based on Natural Rubber and Ceramic Nanoparticles as Control Agents for Leishmania braziliensis. In D. Claborn (Ed.), *Leishmaniasis*. Rijeka: InTech.
- Jović, N. G., Masadeh, A. S., Kremenović, A. S., Antić, B. V, Blanuša, J L Cvjetičanin, N. D., & Božin, E. S. (2009). Effects of Thermal Annealing on Structural and Magnetic Properties of Lithium Ferrite Nanoparticles. J. Phys. Chem. C, 113(48), 20559–20567.
- Jović, N., Prekajski, M., Kremenović, A., Jančar, B., Kahlenberg, V., & Antić, B. (2012). Influence of size/crystallinity effects on the cation ordering and magnetism of α-lithium ferrite nanoparticles. *Journal of Applied Physics*, 111(3).
- Kabezya, K. M., & Motjotji, H. (2015). Material Science & Engineering The Effect of Ball Size Diameter on Milling Performance. *Journal of Material Science and Engineering*, 4(1), 4–6.

- Kang, S. J. L. (2005). Sintering Densification, Grain Growth, and Microstructure. Elsevier Butterworth-Heinemann Linacre House. London, United Kingdom: Elsevier.
- Karmakar, M., Mondal, B., Pal, M., & Mukherjee, K. (2014). Acetone and ethanol sensing of barium hexaferrite particles: A case study considering the possibilities of non-conventional hexaferrite sensor. *Sensors and Actuators, B: Chemical*, 190, 627–633.
- Kaur, M., Yadav, K. L., & Uniyal, P. (2015a). Investigations on multiferroic, optical and photocatalytic properties of lanthanum doped bismuth ferrite nanoparticles. *Advanced Materials Letters*, 6(10), 895–901.
- Kaur, T., Kumar, S., Bhat, B. H., Want, B., & Srivastava, A. K. (2015b). Effect on dielectric, magnetic, optical and structural properties of Nd–Co substituted barium hexaferrite nanoparticles. *Applied Physics A*, 119(4), 1531–1540.
- Khadar, M., Biju, V., & Inoue, A. (2003). Effect of finite size on the magnetization behavior of nanostructured nickel oxide. *Materials Research Bulletin*, *38*, 1341–1349.
- Kingery, W. D. (1974). Plausible Concepts Necessary and Sufficient for Interpretation of Ceramic Grain-Boundary Phenomena: I, Grain -Boundary characteristics, Structure, and Electrostatic Potential. *Journal of the American Ceramic Society*, 57(1), 1–8.
- Klassen, T., Herr, U., & Averback, R. S. (1997). Ball milling of systems with positive heat of mixing: Effect of temperature in Ag-Cu. *Acta Materialia*, 45(7), 2921–2930.
- Klein, L. C. (1996). *Processing of Nanostructured Sol-gel Materials. In Nanomaterials : Synthesis , Properties and Applications* (pp. 145-161). New York: Taylor and Francis.
- Klug, H. P., & Alexander, L. E. (1974). X-Ray Diffraction Procedures: For Polycrystalline and Amorphous Materials. New York: John Wiley & Sons.
- Koch, C. C., Cavin, O. B., Mckamey, C. G., & Scarbrough, J. O. (1983). Preparation of "amorphous" Ni₆₀Nb₄₀ by mechanical alloying Preparation of "amorphous" Ni₆₀Nb₄₀ by mechanical alloying. *Applied Physics Letters*, *1017*, 1–4.
- Komarneni, S., Arrigo, M. C. D., Leonelli, C., Pellacani, G. C., & Katsuki, H. (1998). Microwave-Hydrothermal Synthesis of Nanophase Ferrites. *Journal of the American Ceramic Society*, 81(11), 3041–3043.
- Kumar, C. S. S. R. (2013). Transmission Electron Microscopy Characterization of Nanomaterials. (C. S. S. R. Kumar, Ed.) (1st ed.). New York: Springer.

- Kwon, Y., Gerasimov, K., Lomovsky, O., & Pavlov, S. (2003). Steady state products in the Fe–Ge system produced by mechanical alloying. *Journal of Alloys and Compounds*, *353*(1–2), 194–199.
- Lee, W. D., & Rainforth, W. M. (1994). *Ceramic Microstructures: Property control by processing*. London, United Kingdom: Chapman & Hall.
- Li, X., & Wang, G. (2009). Low-temperature synthesis and growth of superparamagnetic Zn_{0.5}Ni_{0.5}Fe₂O₄ nanosized particles. *Journal of Magnetism* and Magnetic Materials, 321(9), 1276–1279.
- Liang, Y. Y., Bao, S. J., & Li, H. L. (2006). A series of spinel phase cathode materials prepared by a simple hydrothermal process for rechargeable lithium batteries. *Journal of Solid State Chemistry*, *179*(7), 2133–2140.
- Louh, R., Reynolds III, T. G., & Buchanan, R. C. (2004). "Ferrite Ceramics" in Ceramic Materials for Electronics (3rd ed.). New York: Marcel Dekker.
- Low, Z. H., Chen, S. K., Ismail, I., Tan, K. S., & Liew, J. Y. C. (2017). Structural transformations of mechanically induced top-down approach BaFe₁₂O₁₉ nanoparticles synthesized from high crystallinity bulk materials. *Journal of Magnetism and Magnetic Materials*, 429(2016), 192–202.
- Low, Z. H., Hashim, M., Ismail, I., Kanagesan, S., Ezzad Shafie, M. S., Idris, F. M., & Ibrahim, I. R. (2015). Development of Magnetic B-H Hysteresis Loops Through Stages of Microstructure Evolution of Bulk BaFe₁₂O₁₉. Journal of Superconductivity and Novel Magnetism, 28(10), 3075–3086.
- Lutterotti, L., Campostrini, R., Gialanella, S., & Di Maggio, R. (2000). Microstructural Characterisation of Amorphous and Nanocrystalline Structures Through Diffraction Methods. *Materials Science Forum*, 343–346.
- Mahmud, S. T., Akther Hossain, A. K. M., Abdul Hakim, A. K. M., Seki, M., Kawai, T., & Tabata, H. (2006). Influence of microstructure on the complex permeability of spinel type Ni-Zn ferrite. *Journal of Magnetism and Magnetic Materials*, 305(1), 269–274.
- Mastryukov, V. S., Palafox, M. A., & Boggs, J. E. (1994). Inverse bond length/bond angle relationships. Part 6. An ab initio survey of behavioral types. *Journal of Molecular Structure: THEOCHEM*, 300(3), 261–267.

Moulson, A. J., & Herbert, J. M. (2003). Electroceramics (2nd ed.). Wiley.

Najafabadi, A. H., Ghasemi, A., & Mozaffarinia, R. (2016). Development of novel magnetic-dielectric ceramics for enhancement of reflection loss in X band. *Ceramics International*, 42(12), 13625–13634.

- Naldoni, A., Allieta, M., Santangelo, S., Marelli, M., Fabbri, F., Cappelli, S., Bianchi, C. L., Psaro, R., & Dal Santo, V. (2012). Effect of nature and location of defects on bandgap narrowing in black TiO2nanoparticles. *Journal of the American Chemical Society*, 134(18), 7600–7603.
- Nasipuri, D. (1991). Molecular geometry and chemical bonding. In Stereochemistry of organic compounds: Principles, and applications. New Delhi, India: New Delhi: New Age International Limited.
- Novák, P., & Rusz, J. (2005). Exchange interactions in barium hexaferrite. *Physical Review B Condensed Matter and Materials Physics*, 71(18), 1–6.
- Phuoc, T. X., & Chen, R.-H. (2012). Modeling the effect of particle size on the activation energy and ignition temperature of metallic nanoparticles. *Combustion and Flame*, 159(1), 416–419.
- Pullar, R. C. (2012). Hexagonal ferrites: A review of the synthesis, properties and applications of hexaferrite ceramics. *Progress in Materials Science*, 57(7), 1191– 1334.
- Rahaman, M. N. (2007). Sintering of Ceramics. CRC Press. Boca Raton: CRC Press.
- Rikukawa, H. (1982). Relationship Between Mlcrostructures and Magnetic Properties of Ferrites Containing Closed Pores. *IEEE Transactions on Magnetics*, 18(6), 1535–1537.
- Ring, T. A. (1996). Fundamentals of Ceramic Powder Processing and Synthesis. Academic Press.
- Roca, A. G., Marco, J. F., Morales, P., & Serna, C. J. (2007). Effect of Nature and Particle Size on Properties of Uniform Magnetite and Maghemite Nanoparticles. *Journal of Physical Chemistry C*, 111(50), 18577–18584.
- Rodziah, N., Hashim, M., Idza, I. R., Ismayadi, I., Hapishah, A. N., & Khamirul, M. A. (2012). Applied surface science dependence of developing magnetic hysteresis characteristics on stages of evolving microstructure in polycrystalline yttrium iron garnet. *Applied Surface Science*, 258(7), 2679–2685.
- Ross, R. B. (2013). Materials Specification Handbook Metallic Materials Specification Handbook Fourth Edition. United Kingdom: Springer Science & Business Media.
- Sadhana, K., Praveena, K., Matteppanavar, S., & Angadi, B. (2012). Structural and magnetic properties of nanocrystalline BaFe₁₂O₁₉ synthesized by microwave-hydrothermal method. *Applied Nanoscience*, *2*(3), 247–252.

- Saravanan, R., Kannan, Y. B., Srinivasan, N., & Ismail, I. (2017). Study of Various Site Interactions Using Maximum Entropy Method on Mechanically Alloyed Ni_{0.5}Zn_{0.5}Fe₂O₄ Nanoferrite Particles Sintered from 1100 to 1400 °C. *Journal of Superconductivity and Novel Magnetism*, 30(2).
- Šepelák, V., Bergmann, I., Feldhoff, A., Heitjans, P., Krumeich, F., Menzel, D., Litterst, F.J., Campbell, S. J. & Becker, K. D. (2007). Nanocrystalline nickel ferrite, NiFe₂O₄: Mechanosynthesis, nonequilibrium cation distribution, canted spin arrangement, and magnetic behavior. *Journal of Physical Chemistry C*, 111(13), 5026–5033.
- Šepelák, V., Myndyk, M., Witte, R., Röder, J., Menzel, D., Schuster, R. H., hahn, H., Heitjans, P., & Becker, K.D. (2014). The mechanically induced structural disorder in barium hexaferrite, BaFe₁₂O₁₉, and its impact on magnetism. *Faraday Discuss.*, 170, 121–135.
- Sepelak, V., Tkacova, K., & Boldyrev, V. V. (1996). Crystal structure refinement of the mechanically activated spinel-ferrite. *Materials Science Forum*, 783, 228–231.
- Šepelák, V., Wißmann, S., & Becker, K. (1999). Magnetism of nanostructured mechanically activated and mechanosynthesized spinel ferrites. *Journal of Magnetism and Magnetic Materials*, 203(1–3), 135–137.
- Shafie, M. S. E., Hashim, M., Ismail, I., Kanagesan, S., Fadzidah, M. I., Idza, I. R., Hajalilou, A., & Sabbaghizadeh, R. (2014). Magnetic M–H loops family characteristics in the microstructure evolution of BaFe₁₂O₁₉. *Journal of Materials Science: Materilas in Electronics*, 25(9), 3787–3794.
- Shirley, W. A., Hoffmann, R., & Mastryukov, V. S. (1995). An Approach to Understanding Bond Length/Bond Angle Relationships. *The Journal of Physical Chemistry*, 99(12), 4025–4033.
- Siegel, R. W. (1993). Synthesis and Prosessing of Nanostructured Materials. In M. Nastasi, D. M. Parkin, & H. Gleiter, *Mechanical Properties and Deformation Behavior of Materials Having Ultra-Fine Microstructures* (pp. 509–538). Dordrecht: Springer.
- Singh, J. P., Dixit, G., Srivastava, R. C., Kumar, H., Agrawal, H. M., & Chand, P. (2013). Magnetic resonance in superparamagnetic zinc ferrite. *Bulletin of Materials Science*, 36(4), 751–754.
- Singh, J. P., Srivastava, R. C., Agrawal, H. M., Chand, P., & Kumar, R. (2011). Observation of size dependent attributes on the magnetic resonance of irradiated zinc ferrite nanoparticles. *Current Applied Physics*, 11(3), 532–537.
- Singh, R. K., Upadhyay, C., Layek, S., & Yadav, A. (2010). Cation distribution of Ni_{0.5}Zn_{0.5}Fe₂O₄ nanoparticles. *MultiCraft International Journal of Engineering*, *Science and Technology*, 2(8), 104–109.

- Singh, S., Singh, M., Kotnala, R. K., & Verma, K. C. (2014). Nanostructure, Magnetic and Dielectric Properties. *Indian Journal of Pure & Applied Physics*, 52(August), 550–555.
- Sivakumar, N., Narayanasamy, A., Ponpandian, N., Greneche, J.-M., Shinoda, K., Jeyadevan, B., & Tohji, K. (2006). Effect of mechanical milling on the electrical and magnetic properties of nanostructured Ni_{0.5}Zn_{0.5}Fe₂O₄. *Journal of Physics D: Applied Physics*, 39(21), 4688–4694.
- Skomski, R., & Sellmyer, D. J. (2006). Intrinsic and Extrinsic Properties of Advanced Magnetic Materials. *ChemInform*, 37(47), 1.
- Soni, P. R. (2001). *Mechanical alloying*. Cambridge: Cambridge International Science Publishing.
- Sopicka-Lizer, M. (2010). *High-energy ball milling Mechanochemical processing of nanopowders*. Boca Raton, Florida: CRC Press.
- Suryanarayana, C. (2004). *Mechanical Alloying and Milling*. New York: Marcel Dekker.
- Suryanarayana, C., Chen, G. H., & Froes, F. H. (1992). Milling Maps for Phase Identification During Mechanical Alloying. *Scripta Metallurgica et Materialia*, 26(c), 1727–1732.
- Vaezi, M. R., Ghassemi, S. H. M. S., & Shokuhfar, A. (2012). Effect of different sizes of balls on crystalline size, strain, and atomic diffusion on Cu-Fe nanocrystals produced by mechanical alloying. *Journal of Theorrtical and Apllied Physics*, 6(29), 1–7.

Valenzuela, R. (1994). Magnetic Ceramics. New York: Cambridge University Press.

- Viezbicke, B. D., Patel, S., Davis, B. E., & Birnie, D. P. (2015). Evaluation of the Tauc method for optical absorption edge determination: ZnO thin films as a model system. *Physica Status Solidi* (B), 252(8), 1700–1710.
- Waje, S. B., Hashim, M., Yusoff, W. D. W., & Abbas, Z. (2010). Applied Surface Science X-ray diffraction studies on crystallite size evolution of CoFe₂O₄ nanoparticles prepared using mechanical alloying and sintering. *Applied Surface Science*, 256, 3122–3127.
- Wang, J., Xue, J. M., Wan, D. M., & Gan, B. K. (2000). Mechanically activating nucleation and growth of complex perovskites. J. Solid State Chem, 154(2), 321– 328.
- Wang, K. Y., Shen, T. D., Wang, J. T., & Quan, M. X. (1993). Characteristics of the mechanically-alloyed Ni₆₀Ti₄₀ amorphous powders during mechanical milling in different atmospheres. *Journal of Materials Science*, 28, 6474–6478.

- Winterer, M. (2002). Nanocrystalline Ceramics Synthesis and Structure. Berlin: Springer.
- Yang, L. (2008). Materials Characterisation: Introduction to Microscopic and Spectroscopic Methods. Materials Characterization. Singapore: John Wiley & Sons(Asia) Pte Ltd.
- Zahi, S., Daud, A. R., & Hashim, M. (2007). A comparative study of nickel-zinc ferrites by sol-gel route and solid-state reaction. *Materials Chemistry and Physics*, 106(2–3), 452–456.
- Zhong, W., Ding, W., Zhang, N., Hong, J., Yan, Q., & Du, Y. (1997). Key step in synthesis of ultrafine BaFe₁₂O₁₉ by sol-gel technique Wei. *Journal of Magnetism and Magnetic Materials*, 168, 196–202.