

# **UNIVERSITI PUTRA MALAYSIA**

EFFECTS OF STRONTIUM SUBSTITUTION ON STRUCTURAL, ELECTRICAL AND MAGNETIC PROPERTIES OF POLYCRYSTALLINE AND NANOCRYSTALLINE La0.67(Ca1-xSrx)0.33MnO3

**CHANG SEN CHOUNG** 

FS 2014 59



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Thesis Submitted to the School of Graduate Studies, Universiti Putra Malaysia, in Fulfillment of the Requirements for the Degree of Master of Science

April 2014

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Abstract of thesis presented to the Senate of Universiti Putra Malaysia in fulfillment of the requirement for the degree of Master of Science

# EFFECTS OF STRONTIUM SUBSTITUTION ON STRUCTURAL, ELECTRICAL AND MAGNETIC PROPERTIES OF POLYCRYSTALLINE AND NANOCRYSTALLINE La<sub>0.67</sub>(Ca<sub>1-x</sub>Sr<sub>x</sub>)<sub>0.33</sub>MnO<sub>3</sub>

By

### CHANG SEN CHOUNG

#### **April 2014**

# Chairman: Abdul Halim Shaari, PhD Faculty: Science

This research is aimed at studying the influences of Sr ions substitution on the structural, electrical and magnetic properties of polycrystalline and nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ . The differences between polycrystalline and nanocrystalline samples were discussed.

The polycrystalline and the nanocrystalline La<sub>0.67</sub>(Ca<sub>1-x</sub>Sr<sub>x</sub>)<sub>0.33</sub>MnO<sub>3</sub> for  $0 \le x \le 1.0$  were synthesized via the solid-state reaction and the sol gel based polymerizable complex method respectively. The X-Ray Diffraction (XRD) measurements were carried out to determine the crystal structure properties. The XRD spectrums revealed that the structural transition from orthorhombic structure to rhombohedra structure took place when the Ca ions were gradually substituted by the Sr ions in both polycrystalline and nanocrystalline samples. The lattice parameters, Mn-O bond length and Mn-O-Mn bond angle were obtained by the Rietveld refinement method. The microstructures for both polycrystalline and nanocrystalline samples were obtained from the Scanning Electron Microscope (SEM). The grain sizes were found in the ranges of 2.83 µm - 8.78 µm and 35.72 nm - 45.38 nm for polycrystalline samples and nanocrystalline samples respectively.

The temperature dependences of resistivity and magnetoresistance (MR) were measured by the four point probe method at variable magnetic field range of 0 T - 1 T for both polycrystalline and nanocrystalline samples. The metal-insulator transition temperature  $(T_p)$  for polycrystalline samples increased with the substitution of Sr ions. However, the nanocrystalline samples with high surface to volume ratio showed that its  $T_p$  varied with the grain size. The intrinsic MR around the  $T_p$  and the extrinsic MR at  $T \leq T_p$  were observed in polycrystalline samples. The substitution of Sr ions shifted the intrinsic MR of polycrystalline samples towards higher temperature but lowering its magnitude whereas the intrinsic MR of nanocrystalline samples were suppressed and left behind the extrinsic MR at low temperature. The electrical transport properties for both polycrystalline and nanocrystalline samples were explained by the double exchange



interaction within the grain and the spin-polarized tunneling mechanism across the grain boundaries. The highest MR values were found to be -26.79% for polycrystalline x = 0.0 at 244 K and -23.37% for nanocrystalline x = 0.0 at 80 K. At room temperature, the highest MR for polycrystalline samples and nanocrystalline samples were found to be - 8.45% at x = 0.4 and -4.19% at x = 1.0 respectively.

The field dependences of magnetization for both polycrystalline and nanocrystalline samples were carried out by the Vibrating Sample Magnetometer (VSM). The magnetization increased with the Sr ion substitution for the polycrystalline and nanocrystalline samples due to the increase of the Mn-O-Mn bond angle. The nanocrystalline samples have lower magnetization than that of the polycrystalline samples due to the loss of long-range ferromagnetic ordering in the nanocrystalline samples.



Abstrak tesis yang dikemukakan kapade Senat Universiti Putra Malaysia sebagai memenuhi keperluan untuk ijazah Sarjana Sains

# KESAN PENGGANTIAN STRONTIUM PADA SIFAT STRUKTUR, ELEKTRIK DAN MAGNET BAGI POLIHABLUR DAN NANOHABLUR La<sub>0.67</sub>(Ca<sub>1-</sub> <sub>x</sub>Sr<sub>x</sub>)<sub>0.33</sub>MnO<sub>3</sub>

Oleh

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#### April 2014

# Pengerusi: Abdul Halim Shaari, PhD Fakulti: Sains

Tujuan penyelidikan ini adalah untuk mengaji kesan penggantian ion Sr pada sifat struktur, magnet and elektrik bagi polihablur and nanohablur  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ . Perbezaan antara polihablur dan nanohablur bagi sampel  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  telah dibincang dalam tesis ini.

Polihablur dan nanohablur bagi La<sub>0.67</sub>(Ca<sub>1-x</sub>Sr<sub>x</sub>)<sub>0.33</sub>MnO<sub>3</sub> dengan  $0 \le x \le 1.0$  telah disintesis melalui kaedah tindak balas keadaan pepejal and kaedah "sol-gel" melalui pempolimeran kompleks masing-masing. Pengukuran belauan sinar-x (XRD) telah dilaksanakan untuk menentukan sifat struktur hablur. Spektrum XRD menunjukkan bahawa perubahan struktur daripada struktur orthrombik kepada struktur rombohedra berlaku apabila ion Ca diganti dengan ion Sr secara beransur-ansur bagi sampel polihablur dan nanohablur. Parameter kekisi, panjang ikatan Mn-O dan sudut ikatan Mn-O-Mn diperolehi daripada kaedah Rietveld. Mikrostruktur bagi sampel polihablur dan nanohablur diperolehi daripada mikroskop imbasan elektron (SEM). Saiz butiran didapati dalam lingkungan 2.83  $\mu$ m - 8.78  $\mu$ m dan 35.72 nm - 45.38 nm bagi sampel polihablur dan sampel nanohablur masing-masing.

Kerintangan elektrik dan magnetorintangan (MR) melawan suhu bagi sampel polihablur dan nanohablur telah diukur dengan menggunakan kaedah penduga empat titik pada medan magnet dalam lingkungan 0 T - 1 T. Suhu peralihan logam-penebat  $(T_p)$  bagi sampel polihablur didapati meningkat dengan menambahkan penggantian ion Sr. Walau bagaimanapun, sampel nanohablur yang mempunyai nisbah permukaan terhadap isipadu yang tinggi menunjukkan bahawa  $T_p$  berubah dengan saiz butiran. MR intrisik pada sekeliling  $T_p$  dan MR ekstrinsik pada  $T \leq T_p$  telah diperhati dalam sampel polihablur. Penggantian ion Sr menyebabkan MR intrinsik bagi sampel polihablur beralih ke suhu tinggi tetapi magnitudnya menurun. Manakala, MR intrinsik bagi sampel nanohablur didapati ditindaskan dan hanya MR ekstrinsik yang dikekalkan pada suhu rendah. Sifat pergerakan elektrik bagi sampel polihablur dan nanohablur boleh dijelaskan dengan tindakbalas tukarganti gandaan dalam butiran dan mekanisme penerowongan spin-



terkutub melintasi sempadan butiran. Nilai MR yang tertinggi adalah -26.79% untuk polihablur x = 0.0 pada suhu 244 K dan -23.37% untuk nanohablur x = 0.0 pada suhu 80 K. Pada suhu bilik, nilai MR yang tertinggi untuk sampel polihablur dan sampel nanohablur adalah -8.45% pada x = 0.4 dan -4.19% pada x = 1.0 masing-masing.

Pemagnetan terhadap medan magnet telah dijalankan dengan menggunakan magnetometer getaran sampel (VSM). Pemagnetan bagi sampel polihablur dan nanohablur didapati meningkat dengan penambahkan penggantian ion Sr kerana sudut ikatan Mn-O-Mn meningkat. Nilai pemagnetan bagi sampel nanohablur didapati lebih rendah daripada nilai pemagnetan bagi sampel polihablur disebabkan oleh kehilangan susunan feromagnet berjulat panjang dalam sampel nanohablur.



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I certify that a Thesis Examination Committee has met on 28 April 2014 to conduct the final examination of Chang Sen Choung on his thesis entitled "Effects of Strontium Substitution on Structural, Electrical and Magnetic Properties of Polycrystalline and Nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ " in accordance with the Universities and University Colleges Act 1971 and the Constitution of the Universiti Putra Malaysia [P.U.(A) 106] 15 March 1998. The Committee recommends that the student be awarded the Master of Science.

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# **TABLE OF CONTENTS**

	Page
ABSTRACT	ii
ABSTRAK	iv
ACKNOWLEDGEMENTS	vi
APPROVAL	vii
DECLARATION	ix
LIST OF TABLES	xiv
LIST OF FIGURES	xvi
LIST OF ABBREVIATIONS AND SYMBOLS	xx

# CHAPTER

0

1	INT	RODUCTION	1
	1.1	Introduction	1
	1.2	CMR Manganites	2
	1.3	Potential Application	3
		1.3.1 Magnetoresistive Read Head Sensor in Magnetic	3
		Recording Devices	
		1.3.2 Magnetoresistvie Random Access Memory	3
		(MRAM)	
		1.3.3 Speed Control Sensor in Anti-Lock Brake	3
		System (ABS)	
	1.4	Problem Statement	3
	1.5	Scope and Objectives of the Research	4
	1.6	Overview of the Thesis	5
2	LIT	ERATURE REVIEW	6
	2.1	History	6
	2.2	Lanthanum Based Manganites System	7
		2.2.1 Lanthanum Calcium Manganite	7
		2.2.2 Lanthanum Strontium Manganite	9
		2.2.3 Lanthanum Calcium Strontium Manganite	10
	2.3	Polycrystalline Manganites	11
	2.4	Nanocrystalline Manganites	12
	2.5	Synthesis of Doped Manganites by Solid State Reaction	13
	2.6	Synthesis of Doped Manganites by Polymerizable	13
		Complex Sol-Gel Method	
	2.7	Effect of Ionic Size Mismatch in CMR Manganite	14
		Porperties	
3	TH	EORY	17
	3.1	Perovskite Structure	17
	3.2	Tolerance Factor	17

3.3 Crystal Field Splitting and Jahn-Teller Distortion 18

	3.3.1 Crystal Field Splitting	19
	3.3.2 Jahn-Teller Distortion	19
3.4	Double Exchange	21
3.5	Electron-Phonon Coupling	23
		20
4 <b>ME</b>	THODOLOGY	25
4.1	Introduction	25
4.2	Solid State Reaction	25
	4.2.1 Weighing of Raw Materials	26
	4.2.2 Mixing and Milling	27
	4.2.3 Calcination	27
	4.2.4 Grinding and Sieving	27
	4.2.5 Pelletizing	27
	4.2.6 Sintering	27
4.3	Sol-Gel Based Polymerizable Complex Method	28
	4.3.1 Weighing of Raw Materials	29
	4.3.2 Dissolving	29
	4.3.3 Mixing	29
	4.3.4 Chelating	29
	4.3.5 Polymerization	30
	4.3.6 Calcination	30
	4.3.7 Moulding	30
	4.3.8 Sintering	30
4.4	Sample Characterization	30
	4.4.1 X-Ray Diffraction (XRD)	30
	4.4.2 Scanning Electron Microscope (SEM) and	31
	Energy-Dispersive X-Ray Spectroscopy (EDX)	
	4.4.3 Four Point Probe Resistivity Measurement	32
	4.4.4 Vibrating Sample Magnetometer (VSM)	33
5 BES	SULTS AND DISCUSSION	35
5.1	Polycrystalline La <sub>0.67</sub> (Ca <sub>1.x</sub> Sr <sub>x</sub> ) <sub>0.33</sub> MnO <sub>3</sub> Compounds	35
	5.1.1 Structural Properties	35
	5.1.2 Electrical Properties and Magnetoresistance	42
	5.1.3 Magnetic Properties	47
5.2	Nanocrystalline La <sub>0.67</sub> (Ca <sub>1-x</sub> Sr <sub>x</sub> ) <sub>0.33</sub> MnO <sub>3</sub> Compounds	49
	5.2.1 Structural Properties	49
	5.2.2 Electrical Properties and Magnetoresistance	56
	5.2.3 Magnetic Properties	62
5.3	Comparison of Polycrystalline and Nanocrystalline	63
	$La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ Compounds	
	5.3.1 Structural Properties	63
	5.3.2 Electrical Properties and Magnetoresistance	63
	5.3.3 Magnetic Properties	66

6	<b>CONCLUSION AND FUTURE RESEARCH</b>	68
	6.1 Conclusion	68
	6.2 Future Research	69
REFERE	NCES	71
APPEND	ICES	76
BIODAT	A OF STUDENT	93
LIST OF	PUBLICATIONS	94



# LIST OF TABLES

Table		Page
4.1	Stoichiometric amounts of the raw materials for $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ synthesized by solid state reaction	26
4.2	Stoichiometric amounts of the raw materials for $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ synthesized by sol-gel based polymerizable complex method	29
5.1	Structural properties of polycrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds obtained from XRD Rietveld refinement and VPSEM	37
5.2	Elemental ratio for polycrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds at different value of x	41
5.3	Magnetization, coercivity and retentivity of polycrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds	48
5.4	Structural properties of nanocrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds obtained from XRD Rietveld refinement and FESEM	51
5.5	Elemental ratio for nanocrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds at different value of x	52
5.6	Magnetization, coercivity and retentivity of nanocrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds	62
B.1	The crystallite sizes for nanocrystalline $La_{0.67}Ca_{0.33}MnO_3$ with different Ca/Mi ratio	78
C.1	The crystallite size for nanocrystalline La <sub>0.67</sub> Ca <sub>0.33</sub> MnO <sub>3</sub> calcined at different temperature	79
E.1	Weight percent of elements for polycrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.0$	81
E.2	Weight percent of elements for polycrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.2$	82
E.3	Weight percent of elements for polycrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.4$	83
E.4	Weight percent of elements for polycrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.6$	84

E.5	Weight $_x Sr_x)_{0.33} N$	percent InO <sub>3</sub> com	of poun	elements d at $x = 0.8$	for	polycrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	85
E.6	Weight $_x Sr_x)_{0.33} N$	percent InO3 com	of poun	elements d at $x=1.0$	for	polycrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	86
F.1	Weight $_x Sr_x)_{0.33} N$	percent /InO3 comj	of poun	elements d at $x = 0.0$	for	nanocrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	87
F.2	Weight $_x Sr_x)_{0.33} N$	percent /InO3 comj	of poun	elements d at $x = 0.2$	for	nanocrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	88
F.3	Weight $_x Sr_x)_{0.33} N$	percent InO <sub>3</sub> com	of poun	elements d at $x = 0.4$	for	nanocrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	89
F.4	Weight $_x Sr_x)_{0.33} N$	percent /InO3 comj	of poun	elements d at $x = 0.6$	for	nanocrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	90
F.5	Weight $_x Sr_x)_{0.33} N$	percent InO <sub>3</sub> com	of poun	elements d at $x = 0.8$	for	nanocrystalline	La <sub>0.67</sub> (Ca <sub>1-</sub>	91
F 6	Weight	nercent	of	elemente	for	nanocrystalline	Las ca(Ca)	92

F.6 Weight percent of elements for nanocrystalline  $La_{0.67}(Ca_{1.} 92 xSr_x)_{0.33}MnO_3$  compound at x = 1.0

C

# **LIST OF FIGURES**

Figure		Page
2.1	Possible magnetic structures of Mn oxides in the perovskite structure	7
2.2	Phase diagram of the magnetic and electrical properties of La <sub>1</sub> . $_x$ Ca <sub>x</sub> MnO <sub>3</sub> in ranges of 0< x<1	8
2.3	Phase diagram of the electrical and magnetic properties of $La_1$ . $_xSr_xMnO_3$	10
2.4	Phase diagram of temperature as a function of tolerance factor for $A_{0.7}A'_{0.3}MnO_3$ system, where A is a trivalent rare earth cation and A' is a divalent alkali earth cation	15
3.1	Ideal cubic perovskite structure in different views	17
3.2	Distorted perovskites structures of manganites: orthorhombic (left) and rhombohedral (right).	18
3.3	Tilting of MnO <sub>6</sub> octahedra in perovskite distortion	18
3.4	Five-fold degeneracy of <i>d</i> -orbitals	20
3.5	Effect of ligands approaching to a metal ion on the energies of $e_g$ -orbitals and $t_{2g}$ orbitals	20
3.6	Energy levels of 3d-orbitals in Mn <sup>3+</sup> ions. The degeneracies of 3d-orbitals are lifted by crystal field splitting and Jahn-Teller distortion	21
3.7	Schematic of Zener's double exchange interaction	22
3.8	Schematic of Jahn-Teller polaran in an ionic crystal material	23
4.1	Main procedures of the solid state reaction	26
4.2	Main procedures of the sol-gel based polymerizable complex method	28
4.3	Schematic diagram of VSM	33
5.1	XRD patterns scanned at room temperature for polycrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ compounds with different composition <i>x</i>	36

Enlarged view of the diffraction pattern in the ranges of 32.25° - 33.5° 5.2 36 for polycrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds with different composition x5.3 VPSEM images for polycrystalline  $La_{0.67}(Ca_{1-x}Sr_{x})_{0.33}MnO_{3}$ 40 compound with different value of x 5.4 Temperature dependence of resistivity at different applied magnetic 45 field for polycrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds and the corresponding MR at 1 T 5.5 Field dependence of MR for polycrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ 46 compounds at room temperature (300 K). Inset presents the curve of MR at 1 T as a function of x value at room temperature 5.6 Field dependence of MR for polycrystalline La<sub>0.67</sub>(Ca<sub>0.6</sub>Sr<sub>0.4</sub>)<sub>0.33</sub>MnO<sub>3</sub> 47 at different temperature 5.7 Field dependence of magnetization at room temperature for 48 polycrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds 5.8 XRD patterns scanned at room temperature for nanocrystalline 50  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds with different composition x 5.9 Enlarged observation of the diffraction pattern in the ranges of 32.25° 50 - 33.5° for nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds with different composition 5.10 The FESEM images for nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ 55 compound with different value of x5.11 Temperature dependence of resistivity at different applied magnetic 59 field for nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds and the corresponding MR at 1 T 5.12 Field dependence of MR for nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ 60 compounds at room temperature (300 K). Inset presents the curve of MR at 1 T as a function of x value at room temperature 5.13 Field dependence of MR for nanocrystalline La<sub>0.67</sub>Sr<sub>0.33</sub>MnO<sub>3</sub> at 61 different temperature 5.14 Field dependence of magnetization at room temperature for 62 nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  compounds

5.15	Variation of $T_C$ and $T_p$ for polycrystalline La <sub>0.67</sub> (Ca <sub>1-x</sub> Sr <sub>x</sub> ) <sub>0.33</sub> MnO <sub>3</sub> and $T_p$ for nanocrystalline La <sub>0.67</sub> (Ca <sub>1-x</sub> Sr <sub>x</sub> ) <sub>0.33</sub> MnO <sub>3</sub> as a function of <i>x</i> value	64
5.16	Temperature dependence of MR for polycrystalline and nanocrystalline $La_{0.67}Ca_{0.33}MnO_3$	65
5.17	Magnetization at 1 T as a function of x value for polycrystalline and nanocrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ at room temperature. Inset shows the hysteresis curves for polycrystalline and nanocrystalline $La_{0.67}Sr_{0.33}MnO_3$ at room temperature	66
5.18	Variation of coercivity and retentivity as a function of x value for polycrystalline and nanocrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ at room temperature	67
A.1	XRD patterns of the polycrystalline La <sub>0.67</sub> Ca <sub>0.33</sub> MnO <sub>3</sub> with different sintering temperature	76
A.2	SEM images of polycrystalline $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$ sintered at (a) 1000 °C, (b) 1100 °C, (c) 1200 °C and (d) 1300 °C	77
B.1	XRD patterns for nanocrystalline La <sub>0.67</sub> Ca <sub>0.33</sub> MnO <sub>3</sub> with different Ca/Mi ratio. The samples were calcined at 600 °C	78
C.1	XRD patterns for nanocrystalline $La_{0.67}Ca_{0.33}MnO_3$ with different calcination temperature	79
D.1	Measurement of grain sizes for La <sub>0.67</sub> Sr <sub>0.33</sub> MnO <sub>3</sub> compound by using VIS-PRP software	80
E.1	Scanning of EDX in spot and area for polycrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.0$	81
E.2	Scanning of EDX in spot and area for polycrystalline La <sub>0.67</sub> (Ca <sub>1-x</sub> Sr <sub>x</sub> ) <sub>0.33</sub> MnO <sub>3</sub> compound at $x = 0.2$	82
E.3	Scanning of EDX in spot and area for polycrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.4$	83
E.4	Scanning of EDX in spot and area for polycrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.6$	84
E.5	Scanning of EDX in spot and area for polycrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.8$	85

E.6	Scanning of EDX in spot and area for polycrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 1.0$	86
F.1	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.0$	87
F.2	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.2$	88
F.3	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 0.4$	89
F.4	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.6$	90
F.5	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.}xSr_x)_{0.33}MnO_3$ compound at $x = 0.8$	91
F.6	Scanning of EDX in spot and area for nanocrystalline $La_{0.67}(Ca_{1.x}Sr_x)_{0.33}MnO_3$ compound at $x = 1.0$	92

C

# LIST OF ABBREVIATIONS AND SYMBOLS

ABS	Anti-Lock Brake System
AF	Antiferromagnetic
CA	Citric Acid
CAF	Canted Antiferromagnetic
CMR	Colossal Magnetoresistance
CNI	Spin-Canted Insulator
СО	Charge/Orbital Ordering
DTA	Differential Thermal Analysis
EDX	Energy-Dispersive X-Ray Spectroscopy
EG	Ethylene Glycol
FESEM	Field Emission Scanning Electron Microscope
FI or FMI	Ferromagnetic Insulator
FM or FMM	Ferromagnetic Metal
GMR	Giant Magnetoresistance
LFMR	Low Field Magnetoresistance
MI	Metal Ion
MR	Magnetoresistance
MRAM	Magnetoresistive Random Access Memory
PI or PMI	Paramagnetic Insulator
PM	Paramagnetic Metal
SEM	Scanning Electron Microscope
TGA	Thermogravimetric Analysis
VPSEM	Variable Pressure Scanning Electron Microscope
VSM	Vibrating Sample Magnetometer
XRD	X-Ray Diffraction
$\langle r_{A} \rangle$	Average A-site ionic radius
0	Resistivity
G G	Correction Factor
V	Voltage
V	Unit Cell Volume
$T_{C}$	Ferromagnetic-Paramagnetic Transition Temperature or Curie
- C	Temperature
$T_n$	Metal-Insulator Transition Temperature
$T_N$	Nell Temperature
I	Current
n	Integer Number
λ	wavelength of X-Ray
$d_{hkl}$	Interplaner Distance of (hkl) Planes
θ	Angle between Incident Ray and (hkl) Plane
t	Thickness of a Slice
_	Thickness Factor
$T_2$	THICKNESS FACIOI
$T_2$ $F(t,c)$	Additional Correction Factor depending on both thickness and
$T_2 F(t,c)$	Additional Correction Factor depending on both thickness and contour

emf	Electromotive Force
N	Number of Wire turns
S	Time
В	Magnetic Field
A	Coil Turn Area
9	Angle between B Field and the Direction Normal to the Coil
	Surface
$r_{\rm A}$	Ionic Radius of A-site Ion
r <sub>B</sub>	Ionic Radius of B-site Ion
r <sub>O</sub>	Ionic Radius of Oxygen Ion
r <sub>Mn</sub>	Ionic Radius of Manganese Ion
d <sub>A-O</sub>	Distance between A-Site Ion and Oxygen Ion
d <sub>Mn-O</sub>	Distance between Manganese Ion and Oxygen Ion
$t_e$	Effective Transfer Integral
$t_0$	Normal Transfer Integral
$\theta_s$	Angle between the Two Spin Direction
f	Goldschmidt Tolerance Factor
Κ	Constant Depending on the Grain Shape
$\beta_{Size}$	Full Width at Half Maxima of XRD Peak
$R_H$	Resistance or Resistivity in the Present of External Magnetic Field
$R_0$	Resistance or Resistivity in the Absence of External Magnetic Field

C

### **CHAPTER 1**

# **INTRODUCTION**

# **1.1 Introduction**

The present high technology era strongly relies on the development of the smart and smaller magnetic material for various applications particularly in the computer industries. The discovery of the magnetoresistance (MR) effect which is defined as the ability of resistivity changed upon external magnetic field, have boosted up the development of magnetic sensor and read head sensor in data storage and electronic devices by the application of MR. One of the current technological challenges is to produce a smaller device with high sensitivity. Thus, studying the MR effect and looking for the highest MR operated at room temperature are required for next-generation devices.

During recent decades, the manganites systems have received tremendous attention in consequence of the discovered extraordinary large negative MR near the metal-insulator transition temperature  $(T_p)$  and the ferro-paramagentic transitions temperature  $(T_c)$ , so-called 'Colossal' Magnetoresistance (CMR). The MR effect of the CMR materials is in the order of magnitude larger than a typical giant magnetoresistance (GMR) that is currently used in the magnetic device. Thus, the CMR manganites with enormous MR have greater sensitivity to the magnetic field, making it possible to be applied in various technologies such as magnetic sensor and the read head sensor for hard disk drive. Researchers believe that the CMR manganites are promising to be the new generation of the magnetic sensor in the hard disk drive.

A lot of researches have devoted much effort in the study of CMR properties in manganites systems, in order to have a deeper understanding of the physical origin of the CMR properties. Moreover, researchers are looking for the optimization of the CMR properties at room temperature and lower magnetic field in order to achieve their applicability to hand on devices. The ferromagnetic coupling between Mn<sup>3+</sup> and Mn<sup>4+</sup> plays an important role in governing the CMR properties. The mechanism is explained by the double exchange interaction where the  $e_g$  electrons from Mn<sup>3+</sup> can hop to Mn<sup>4+</sup> when their spins are parallel. The hopping amplitude of  $e_g$  electron via double exchange is strongly dependent on the Mn-O-Mn bond angle and Mn-O bond length in the perovskite structure (Coey et al., 1999). Radaelli et al. (1997) showed that the magnetic and electrical properties are sensitive to the average A-site ionic radius that directly influences the internal atomic structure and the electronic band width. Hence, it is possible to optimize the CMR properties by manipulating the atomic structure of perovskite manganites. Besides that, researchers have found an extrinsic MR effect at a moderately low magnetic field (<0.1T), which is contributed by the grain boundary (Hwang et al., 1996). This brings out the necessity to study the microstructure dependence of the MR properties of manganites as well.

### **1.2 CMR Manganites**

The general formula of manganites compound is  $R_{1-x}A_xMnO_3$  where R is a trivalent rare earth ion and A is a divalent alkaline earth ion. Without hole doping (x = 0), the parent manganites compound is an antiferromagnetic insulator. By the partial substitution of

trivalent rare earth ions with divalent alkali earth ions, this leads to the co-existing of mix-valence  $Mn^{3+}$  and  $Mn^{4+}$  in the manganites (Jonker and Santen, 1950). Around x = 0.33, the mix-valence manganites exhibit a transition from a high-temperature paramagnetic insulator to a low-temperature ferromagnetic metal. At this transition point, an enormous change in resistivity upon external applied magnetic field is observed and attributing to the CMR effect (Jin et al., 1994; Helmolt et al., 1993).

MR is the relative change in the electrical resistance or resistivity of a material upon an external magnetic field. The MR effect is defined as the percentage of the fractional change to the zero field resistance,

$$MR = \frac{\Delta R}{R_0} \times 100\% = \left[\frac{R_H - R_0}{R_0}\right] \times 100\%,$$
(1.1)

where  $R_0$  and  $R_H$  are the resistance or resistivity in the absence and the presence of external magnetic field respectively. Recent investigations have found that mix-valence manganites tend towards 100% of MR value. Thus, alternative definition of MR effect is expressed in MR ratio,

MR ratio = 
$$\left[\frac{R_H - R_0}{R_H}\right] \times 100\%$$
. (1.2)

This definition gives a better way to show how many orders of resistance magnitude can be decreased by an applied magnetic field. For example, Jin et al. (1994) observed a large negative MR as large as 99.92% in  $La0_{.67}Ca_{0.33}MnO_3$  thin film at temperature 77 K and 6 T magnetic field. Alternatively, expression in term of MR ratio in this case is 127000%, which is a truly "colossal" MR factor (Dagatto et al., 2001; Raveau et al., 1998). Thus, the term "colossal" was coined because of its thousand-fold of MR ratio observed in manganites oxide.

The CMR is due to the suppression of spin fluctuations by aligning the spin parallel upon an external magnetic field which favors the double exchange interaction, consequently enhancing the mobility of charge carries. CMR effect is only observed in the appropriate doping of parent compound where the mix-valance state is present ( $Mn^{3+}$  and  $Mn^{4+}$  coexists in the manganites oxide). In this mix-valence state, the itinerant electrons can hop to the neighboring  $Mn^{4+}$  ion via the double exchange mechanism. Generally, the highest  $T_C$  was found in doped manganites at doping level x = 0.33. The most significant MR effect is observed at the vicinity of  $T_C$ .

### **1.3 Potential Application**

The CMR manganites material has a large potential for the application based on their various physical and chemical properties. For examples: magnetoresistive read head sensor in the magnetic recording devices, magnetoresistive random access memory and speed control sensor in the anti-lock brake system.

# 1.3.1 Magnetoresistive Read Head Sensor in Magnetic Recording Devices

CMR manganite materials with large MR is a very good magnetic sensor that can be used as read head sensor for the magnetic recording device. In hard disk, the read head detects the change in the direction of magnetization (represents the binary data bits) emanating from the magnetic media and then changes its resistivity correspondingly. The MR effect in CMR materials are in the order of magnitude larger than the typical GMR that is currently used in magnetic device. Hence, this material has a greater sensitivity to the magnetic field and making it possible to detect smaller recorded data bit (White et al., 1994).

## 1.3.2 Magnetoresistive Random Access Memory (MRAM)

MRAM is a non-volatile random-access memory that can retain information for a long period of time even in the absence of electrical power. A data bit is stored in a spin valve which is composed of two ferromagnetic plates separated by a thin insulating layer. Both the ferromagnetic plates hold a magnetic field where one of the two plates is permanently magnetized in a particular orientation and the other plate changes its magnetic field according to the external field to store memory. Both data bits 1 and 0 are represented by the parallel and antiparallel moment in the spin valve and can be determined by measuring the resistance of the spin valve. Comparing with the conventional RAM chip technologies, MRAM records data in the form of magnetic moment instead of electric charge. This makes MRAM has an advantage in retaining their data over time and not necessary to refresh their contents. Hence, MRAM is expected to have much lower power consumption compared to the conventional RAM (Zhuang et al., 2002; Katti, 2000).

### 1.3.3 Speed Control Sensor in Anti-Lock Brake System (ABS)

The CMR materials can be used as a speed control sensor to detect the rotation and speed of the steel disc attached to the automobiles wheel and then transferring the information to a computer for monitoring the rotational speed of each wheel. If the system detects a wheel rotating significantly slower than the others, it reduces the braking pressure at the detected wheel. Conversely, if the system detects a wheel rotating faster than the others, it increases the braking pressure at the detected wheel. This condition of releasing and increasing on the braking pressure allows the wheels on the automobile keep friction contact with the road surface while braking. Hence, it is able to avoid the automobiles from uncontrolled skidding and to decrease the automobiles braking distance (Aly et al., 2011).

# **1.4 Problem Statement**

The CMR effect in the manganites is much higher than the current employing magnetic multi-layer system, i.e. GMR. Hence, it is prospected to increase the sensitivity of the magnetic sensor and to reduce the operation power required. However, the conventional magnetic sensor requires material which is able to operate at room temperature and low magnetic field. Commercialization of the CMR is discouraged because of

- Significant CMR in manganites appears only in a high magnetic field of several Tesla range, which is considered too large for the potential use in the magnetic recording (Dagotto et al., 2001) and
- The CMR effect is rather low at room temperature.
  - Significant CMR effect is confined to a quite narrow temperature range around the  $T_C$ . The CMR effect is relatively low beyond  $T_C$  (Tokura and Tomioka, 1999).
  - > CMR effect appears significant only when its  $T_C$  temperature is at low temperature. High  $T_C$  temperature will sacrifice the CMR effect (Dagotto et al., 2001).

Hence, enhancing the CMR properties of the manganite materials for utilizing in room temperature and low magnetic field will be the final goal for researchers.

### 1.5 Scope and Objectives of the Research

This project mainly studies the structural, electrical, magnetic and CMR properties on  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  manganites compounds with the x value ranging from 0 to 1.0. Samples with two different crystalline forms were synthesized i.e. polycrystalline bulk and nanocrytalline manganites. The polycrystalline bulk manganites were synthesized via the solid state reaction which is the most common method in synthesizing various kind of ceramic. Nanocrystalline manganites were synthesized via the sol-gel based polymerizable complex method in order to confine the crystallite size down to nanoscale. A systematic characterization was carried out upon the polycrystalline bulk and the nanocrystalline samples. Structural properties of these samples were investigated by X-Ray Diffraction (XRD), Scanning Electron Microscope (SEM) and Energy-Dispersive X-Ray Spectroscopy (EDX). While the MR effect in the manganite samples was measured by employing the four point probe method with applied magnetic field range of 0 T to 1 T. The magnetization of the samples was investigated by Vibrating Sample Magnetometer (VSM) at room temperature and applied magnetic field ranges from 0 kG to 10 kG. Lastly, the experiments data obtained were analyzed and the features of both polycrystalline and nanocrystalline manganites samples were studied and discussed. Hence, the objectives of this research are:

- I. To characterize polycrystalline and nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  samples synthesized by the solid state reaction and sol-gel based polymerizable complex method respectively.
- II. To study the effects of Strontium ions substitution at Calcium site by investigating and comparing the structural, electrical and magnetic properties of the polycrystalline and nanocrystalline samples.
- III. To find the optimum composition of both polycrystalline and nanocrystalline  $La_{0.67}(Ca_{1-x}Sr_x)_{0.33}MnO_3$  for obtaining the highest MR effect at room temperature.

## **1.6 Overview of the Thesis**

In the first chapter, an introduction concerning about this project is presented. Problem statements regarding to the research of CMR materials are mentioned. The scope of project and the objectives are presented. Lastly, the overview of this thesis is given.

In chapter 2, the histories of the CMR material will be presented. A review on the existing literatures related to the physical properties of CMR manganites is also given.

In chapter 3, some theories related to the CMR material will be presented. It is important to understand the theories behind the CMR phenomenon so that the experimental results can be well interpreted.

In chapter 4, the implemented methodology, the synthesis method adopted in this project will be discussed in details. Systematic characterizations are carried out. The equipments involved in this project will be introduced and the experimental settings will also be provided in chapter 4.

In chapter 5, the experimental data will be presented in the form of graphs, pictures or tables to give a clear and facilitated observation. The structural, magnetic and electrical properties of the CMR manganite samples will be discussed with respect to the analyzed graph, picture and table.

Finally, in chapter 6, the conclusion of this project will be presented. The future research will also be suggested in this chapter.

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