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PHYSICAL AND MECHANICAL PROPERTIES OF NANOCOPPER PARTICLE-REINFORCED ALUMINA MATRIX COMPOSITES

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Thesis Submitted to the School of Graduate Studies, Universiti Putra Malaysia, in Fulfillment of the Requirements for the Degree of Doctor of Philosophy

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DEDICATION

To the spirit of my dear father (Sabah Ali Al-Mayali)

To my mother For her unconditional love and support

> To my siblings and family For making my life complete

To my wife (Intisar), daughters (Noor and Tabark), and sons (Ali and Hussain) For their love and care

> To all my very wonderful friends For making my life full of joy and happiness

> > Thank you all.

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PHYSICAL AND MECHANICAL PROPERTIES OF NANOCOPPER PARTICLE-REINFORCED ALUMINA MATRIX COMPOSITES

By

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September 2017

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Over the past century, there has been a dramatic increase in fabrication and synthesizing of porous ceramics. However, only a few of them used waste material to fabricate alumina porous ceramics and reinforced it using nano-copper (Cu) particles. The motivation behind these efforts are the increasing raw materials cost and decreasing natural resources consumption which requires the use of byproducts and wastes as raw material for different industrial processes. This is a step towards environmental protection, sustainable development, and also to produce porous alumina ceramics with good porosity and mechanical properties. Thus, in this study, porous alumina ceramics were fabricated using graphite waste, natural active yeast, and rice husk ash as poreforming agents and source of silica (SiO₂). Series of porous alumina ceramics was prepared using powder metallurgy technique. The physical and mechanical properties of porous alumina ceramics with and without nano-copper (Cu) particles were measured by differential thermal analysis (DTA), energy-dispersive X-ray spectroscopy (EDX), linear shrinkage, average density (green and sintered) data measurement, and Universal Testing Machine (UTM). The average densities for both green and sintered samples decrease with increasing pore forming agent ratio for porous alumina ceramics with and without nano-copper (Cu) particles. While the linear shrinkage increases with the increase of pore forming agent ratio with and without nano-copper (Cu) particles. Besides, the structural properties of porous alumina ceramics with and without nanocopper (Cu) particles, ceramic phases, morphology, and porosity were examined using X-ray diffraction (XRD) and field-emission scanning electron microscopy (FESEM). The effects of the pore-forming agent ratios on the mechanical properties, the porosity and the microstructure with and without nano-copper (Cu) particles have been investigated in this study. The results showed that through increasing the pore-forming agent ratio for graphite waste, natural active yeast, and rice husk ash, the porosity increased from 37.3 to 61.1%, 30.2 to 63.8% and 42.9 to 49.0%, respectively. The hardness also decreased from 172.6 to 38.1 HV₁ and from 160.6 to 15.0 HV₁ for porous alumina ceramics using graphite waste and yeast as pore-forming agents, respectively.

However, the hardness of the porous alumina ceramics with rice husk ash as a poreforming agent increased at 30 wt.% (150.9 HV₁) and 50 wt.% (158.9 HV₁). The tensile strength for porous alumina ceramics using graphite waste and natural active yeast as pore-forming agents decreased from 24.9 to 14.3 MPa and from 26.2 to 5.4 MPa, respectively. The compressive strength decreased from 112.3 to 34.3 MPa and from 19.5 to 1.8 MPa, respectively. The flexural strength decreased from 71.28 MPa to 30.42 MPa and from 72.56 MPa to 20.72 MPa, respectively. However, for porous alumina ceramics using rice husk ash, the tensile strength increased at 30 wt.% (24.1 MPa) and 50 wt.% (21.9 MPa). The compressive strength also increased at 30 wt.% (69.7 MP) and at 50% (60.1 MPa). The flexural strength increased at 30 wt.% (93.38 MPa) and 50 wt.% (92.38 MPa). The variation in mechanical properties was also attributed to the formation of ceramic phases such as mullite, cristobalite, corundum, and sillimanite other than the formation porosity. It is also found that with increasing porosity, the mechanical properties decrease. This is a good agreement with Rice's formula. While by adding nano-copper (Cu) particles all mechanical properties improved with increasing Cu ratio which attributed to decrease porosity and formation ceramic phases such as tenorite (CuO).

Abstrak tesis yang dikemukakan kepada Senat Universiti Putra Malaysia sebagai memenuhi keperluan untuk Ijazah Doktor Falsafah

SIFAT FIZIKAL DAN MEKANIKAL ZARAH TEMBAGA NANO BERTETULANG KOMPOSIT MATRIKS ALUMINA

Oleh

MOHAMMED SABAH ALI

September 2017

Pengerusi Fakulti : Profesor Madya Azmah Hanim Mohamed Ariff, PhD : Kejuruteraan

Sejak ber abad yang lalu, terdapat peningkatan dramatik dalam fabrikasi dan sintesis seramik berliang menggunakan bahan-bahan buangan. Walau bagaimanapun, hanya sebahagian sahaja menggunakan bahan buangan untuk menghasilkan alumina seramik berliang dan diperkukuh menggunakan zarah nano tembaga (Cu). Motivasi di sebalik usaha ini adalah kerana kurangnya penggunaan sumber asli dan kos bahan mentah yang semakin meningkat yang memerlukan penggunaan hasil sampingan dan sisa sebagai bahan mentah untuk proses industri yang berbeza. Ini adalah satu langkah ke arah perlindungan alam sekitar dan pembangunan lestari serata untuk menghasilkan seramik berliang alumina dengan keliangan yang sesuai dan sifat-sifat mekanikal yang baik. Oleh itu, dalam kajian ini, alumina seramik berliang telah direka menggunakan sisa grafit, yis aktif semula jadi dan abu sekam padi sebagai ejen pembentuk liang dan sumber silika (SiO₂). Beberapa seramik berliang alumina telah disediakan dengan menggunakan teknik metalurgi serbuk. Sifat-sifat fizikal dan mekanikal seramik alumina berliang samada dengan dan tanpa zarah nano-tembaga (Cu) diukur melalui analisis terma (DTA), tenaga-serakan X-ray spektroskopi (EDX), pengecutan linear, ketumpatan purata (hijau dan tersinter) pengukuran data dan mesin ujian sejagat (UTM). Ketumpatan purata bagi kedua-dua sampel hijau dan tersinter menurun dengan peningkatan nisbah ejen pembentuk liang untuk seramik alumina berliang dengan dan tanpa zarah nanotembaga (Cu). Manakala, pengecutan linear meningkat dengan peningkatan nisbah ejen pembentuk liang dengan dan tanpa zarah nano-tembaga (Cu). Di samping itu, sifat-sifat struktur alumina seramik berliang dengan atan tanpa zarah nano-tembaga (Cu), fasa seramik, morfologi dan keliangan telah diperiksa menggunakan X-ray pembelauan (XRD) mikroskop elektron pengimbas (FESEM). Kesan nisbah ejen pembentuk liang ke atas sifat mekanik, keliangan dan mikrostruktur dengan dan tanpa zarah nanotembaga (Cu) telah disiasat dalam kajian ini. Hasil kajian menunjukkan bahawa dengan meningkatkan nisbah ejen pembentuk liang bagi sisa grafit, yis aktif semulajadi dan abu sekam padi, keliangan meningkat setiap satu daripada 37.3 ke 61.1%, 30.2 ke 63.8% dan 42.9 ke 49.0%. Kekerasan juga menurun 172.6 ke 38.1 HV₁ dan 160.6 ke 15.0 HV₁

untuk seramik alumina berliang menggunakan sisa grafit dan yis sebagai ejen pembentuk liang. Walau bagaimanapun, kekerasan seramik alumina berliang dengan abu sekam padi sebagai ejen pembentuk liang meningkat pada 30 wt.% (150.9 HV₁) dan 50 wt.% (158.9 HV₁). Kekuatan tegangan untuk seramik alumina berliang menggunakan sisa grafit dan vis aktif semulajadi sebagai agen pembentuk liang menurun daripada 24.9 ke 14.3 MPa dan 26.2 ke 5.4 MPa. Kekuatan mampatan menurun daripada 112.3 ke 34.3 MPa dan 19.5 ke 1.8 MPa. Kekuatan lenturan menurun daripada 71.28 MPa kepada 30.42 MPa dan dari 72.56 MPa kepada 20.72 MPa, secara respektif. Walau bagaimanapun, untuk seramik alumina berliang menggunakan abu sekam padi, kekuatan tegangan meningkat pada 30 wt.% (24.1 MPa) dan 50 wt.% (21.9 MPa). Kekuatan mampatan juga meningkat pada 30 wt.% (69.7 MP) dan pada 50% (60.1 MPa). Kekuatan lenturan meningkat pada 30 wt.% (93.38 MPa) dan 50 wt.% (92.38 MPa). Perubahan dalam sifat-sifat mekanikal juga disebabkan oleh pembentukan fasa seramik seperti mullite, cristobalite, aluminum oksida dan sillimanite selain daripada pembentukan keliangan. Kajian mendapati dengan peningkatan keliangan, sifat-sifat mekanikal berkurangan. Ini adalah bersamaan dengan formula Rice. Walau bagaimanapun selepas menambah zarah nano-tembaga (Cu), semua sifat-sifat mekanikal meningkat dengan peningkatan nisbah Cu yang dikaitkan dengan mengurangkan bilangan keliangan dan pembentukan fasa seramik seperti tenorite (CuO).

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May 2017

I certify that a Thesis Examination Committee has met on 12 September 2017 to conduct the final examination of Mohammed Sabah Ali on his thesis entitled "Physical and Mechanical Properties of Nanocopper Particle-Reinforced Alumina Matrix Composites" in accordance with the Universities and University Colleges Act 1971 and the Constitution of the Universiti Putra Malaysia [P.U.(A) 106] 15 March 1998. The Committee recommends that the student be awarded the Doctor of Philosophy.

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LIST OF ABBREVIATIONS

| ACP | Ammonium hexachloroplatinate |
|----------|---|
| ASTM | American Society for Testing and Materials |
| DTA | Differential thermal analysis |
| DTS | Diametric tensile strength |
| EDX | Energy-dispersive X-ray |
| FESEM | Field-emission scanning electron microscopy |
| НАР | Hydroxyapatite |
| HRD | Hardness |
| JCPDS | Joint Committee on Powder Diffraction Standards |
| КР | Kenaf powder |
| PMMA | Polymethylmethacrylate |
| Pos.[2θ] | Position [20] |
| PVB | Polyvinyl butyral |
| R&D | Research and development |
| RBAO | Reaction bonding of aluminum oxide |
| RHA | Rice husk ash |
| RPC | Reticulated porous ceramic |
| S. D | Stander deviation |
| TEM | Transmission electron microscopy |
| TGA | Thermogravimetric Analysis |
| UTM | Universal Testing Machine |
| XRD | X-ray diffraction |
| | |

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CHAPTER 1

INTRODUCTION

Overview

This study investigates the effect of different pore-forming agents (graphite waste from the primary battery, natural active yeast, and rice husk ash) on the physical, microstructural and mechanical properties of porous alumina ceramics with and without nano-copper particles (Cu). The physical properties included porosity, density (green and sintered), and linear shrinkage. The microstructural properties involved morphology, pore shape, and grains while the mechanical properties included the hardness, compressive strength, tensile strength and flexural strength. This study involved using sacrificial and pressureless techniques to improve the mechanical properties of porous alumina ceramics using waste materials and sugar as a binder.

This chapter highlights the research background, problem statement, research hypothesis, research objective, the scope of this study and contributions to knowledge.

1.1 Research background

The solid materials that have been obtained from the burning of clays are known the ceramics, which derived from the Greek word keramos. Also, the ceramics can be defined as materials, which often include crystalline structure, inorganic and non-metallic materials. The ceramic materials involve of both nonmetallic and metallic elements such as Si₃N₄, ZrO₂, CaO, SiO₂, and Al₂O₃. In other words, based on the modern definition, ceramics materials are either amorphous or crystalline solid materials comprising only covalent, ionic or ionocovalent chemical bonds between nonmetallic and metallic elements. Firing and calcining are the important processes used in the preparation the ceramic and raw materials. Burning or firing is the final heat treatment conducted in the furnace on the green ceramic material to develop a strong chemical bond and produce other required chemical, mechanical and physical properties. Calcining involves the heat treatment of raw materials before used to produce the final ceramic materials. The point of calcination is to produce changes in volume and remove the combined constituents which will volatile chemically (Cardarelli, 2008).

Based on the industrial applications of ceramic materials, ceramics are classified to major categories such as cements, refractories, glasses, abrasives, and advanced porous ceramics. Today, one of the important industrial applications of ceramic materials is the advance porous ceramics due to their benefit in the scientific and industrial fields, which focus on the relationship between properties and microstructure, developments of processing and discovering new application. The unique properties of tailored porous ceramic, such as its excellent strain and damage tolerance, good thermal shock resistance, wear resistance, high corrosion and its lightweight, render advanced ceramic as potential components (Jean, 2014; Zhang *et al.*, 2012) of filtering materials for separation membranes, lightweight structural materials (Tang, 2004), catalyst supports, thermal insulation, bioreactors, gas filters for high temperature, (Dessai, 2013; Dong *et al.*, 2017; Yu, 2011) medical ultrasonic imaging and underwater sonar detectors. Therefore, these advantages, make advanced porous ceramic more distinctive compared to other materials such as polymeric and metallic materials in certain applications (Rahaman, 2006).

Ceramics with designed porosity is one of the most versatile materials for thermal insulation, filters, bio-scaffold for tissue engineering, absorption and as catalysts (Konrad *et al.*, 2014). The past decade has seen the rapid development of porous ceramic, several efforts have been devoted by the researchers on inventing porous ceramic processing technologies, that lead to a significant improvement in porous ceramic structure and properties (Hammel *et al.*, 2014; Ohji and Fukushima, 2012).

Macroporous ceramics with designed porosity have a wide application including 1filtration in high temperature 2- diesel filters 3- thermal insulation 4- bone implants and others. In addition, replica, sacrificial templates, and direct foaming methods have been discovered by several scientists for manufacturing macroporous ceramics (Ahmad *et al*, 2014) as shown in Figure 1.1.

Generally, porous ceramics can be classified into three grades according to its pore diameter: 1) micro-pore ceramics in the range of d < 2 nm, 2) meso-pore ceramics in the range of 50 nm > d >2 nm, and 3) macro-pore ceramics in the range of d > 50 nm. (Ohji and Fukushima, 2012; Studart *et al.*, 2006). For example, meso- and macro-pore ceramics are desired in sensors and catalysis to supply a high surface area and to improve the accessibility of liquids and gases to reactive areas. Small pores in the range of 50-100 nm are desired to provide physical cues that promote differentiation, proliferation, the migration of cells and finally quick healing. Large pores >300 - 400 µm with hierarchical structures are desired in regenerative medicine for implanted scaffold vascularization (Studart *et al.*, 2011).

Unfortunately, the mechanical properties of porous ceramics decreased when the porosity area increased and the fracture toughness of ceramic is also low.



Figure 1.1: Classification of porous ceramics according to pore size, applications, and fabrication methods (Ohji and Fukushima, 2012)

1.2 Problem statement

The motivation behind these efforts are the increasing raw materials cost and decreasing natural resources consumption which requires the use of byproducts and wastes as raw material for different industrial processes. This is also a step towards environmental protection and sustainable development. Because of the large amounts of agricultural and industrial waste in the world this days, the present research would like to use graphite waste from primary battery as industrial waste, natural active yeast as microorganism's materials and rice husk ash as pore-forming agent to produce macroporous ceramic materials reinforced with ductile nano-metals particles (nano-copper).

In spite of the growth in macroporous ceramics with designed porosity and their wide applications including filtration in high temperature, diesel filters, thermal insulation, bone implants, absorptions, and catalyst. The main disadvantage of porous ceramic with designed porosity that is the decreasing mechanical properties when the porosity increase. In filters, the mechanical properties must be strong enough to withstand the pressure during operating time and must have thermal and chemical properties that is important for it to function sustainably especially in hot gas and molten metal filtration (Hammel et al., 2014; Konrad et al., 2014; Ohji and Fukushima, 2012). Therefore, in the case of the filtration of hot gas and molten metal, the fluctuation of temperature during the process will leave the materials liable to thermal shock. During service, the mechanical properties of the filter must be high enough to bear the operation pressure, and also the filter properties must not deteriorate with the temperature increase. In addition, the range of temperature (260-900°C) in the filtration process is considered in the filtration of hot gas and these filters may face pressures of up to 8 MPa. Because filtration occurs under these conditions, it is important that the filters of ceramics have sufficient mechanical strength and thermal shock resistance (Hammel, 2014). Therefore, in this study, nano-copper particles have been used as a reinforcement factor to improve the mechanical properties of porous alumina samples. The conditions for the porous alumina ceramics include a reinforced phase when sintering at high temperatures using a new process that requires the addition of Cu metal in nanoscale directly through a combination of the sacrificial technique and pressure-less sintering methods which is a cost-effective procedure.

1.3 Research hypothesis

This study is carried out with three main hypotheses as follows.

1- Depending on the thermal properties of pore-forming agents, it can produce alumina porous ceramics with different level of porosity and mechanical properties through sintering at high temperature.

2-The presence of porosity with different levels leads to decrease in the mechanical properties of alumina porous ceramics however the presence of ceramic phases such as silica (SiO₂) plays a significant role in improving mechanical properties despite the presence of porosity.

3-Addition of nano-metal particles in porous alumina ceramics would affect strongly the mechanical properties by decreasing the porosity, toughening mechanism, and formation of ceramic phases.

1.4 Research objectives

In the present research work, porous alumina ceramics with and without nano-copper particles (Cu) have been prepared using pressureless and sacrificial techniques. All porous alumina ceramics were characterized for the physical and mechanical properties.

The research objectives are;

- To investigate the pore formation in alumina matrix with graphite waste, natural active yeast, and rice husk ash (RHA) and its effect on the physical properties.
- To determine the relationship between different pore modifier wt. % from 10 to 50% on the pore formation and the relationship to the mechanical properties.
- To investigate the physical properties of alumina matrix with different pore modifier reinforced with copper particles between 3-12 wt.%.
- To investigate mechanical properties of alumina matrix with different pore modifier reinforced with copper particles between 3-12 wt.%.

1.5 Scope of the study

In order to reach the objective of the study, the scope of the study are as follows.

- 1- Porous alumina ceramics have been prepared using different pore agents (graphite waste, natural active yeast, and rice husk ash) based on the ratios 10 wt.%, 20 wt.%, 30 wt.%, 40 wt.%, and 50 wt.% of pore agent using the sacrificial and pressureless sintering techniques.
- 2- A reinforced porous alumina ceramics have been prepared using Cu metal in nanoscale particles as reinforcement phase through the ratios of 3 wt. %, 6 wt. %, 9 wt. % and 12 wt. % of (Cu) metal for selected ratios of all pore agent.
- 3- The chemical phases and chemical composition of pore agents and alumina powder have been determined using XRD, TEM, and EDX in order to discover the chemical phases and chemical composition of pore agent and material matrix.
- 4- Identifying the first sintering temperature of green ceramics to remove the pore agent according to the weight loss by conducting the TGA and DTA of pore agent materials.
- 5- Mechanical properties of porous and reinforced porous alumina ceramics have been measured using UTM-machine.
- 6- Pore size distribution, physical and structural properties of porous and reinforced porous alumina ceramics have been analyzed using FESEM, XRD, Archimedes method, and linear shrinkage.

1.6 Importance of the study and limitation

- 1- Contribution of knowledge to the materials engineering field in the possibility of using new material as a pore former and improve the technique to strengthen and produce the macroporous ceramic with porosity designed by using ductile nano metal particles.
- 2- To manufacture porous ceramic composite by using industrial and agricultural waste.
- 3- To produce porous ceramic composites with high mechanical properties by adding the nano metal particle.
- 4- To produce macroporous ceramic materials that can be used in potentials application, for example, metal filters, hot gas filters, membranes, and bioceramics. In addition, one of the importance limitation of producing of porous ceramics using sacrificial fugitives is low interconnectivity among the pores.

1.7 Outline of thesis

The thesis arrangement is designed as follows.

Chapter 1 explains an introduction of porous and reinforced porous alumina ceramics, the problem statement, the objective, the scopes and also the importance of this research study. The theory, features and previous works including the past and current work that has been carried out by other researchers of porous ceramics are explained in **Chapter 2**.

The methodology and characterization of the porous and reinforced porous alumina using graphite waste, natural active yeast and rice husk ash as pore-forming agent are explained **in Chapter 3**.

The results regarding the effect of the addition of different pore agent (graphite waste, natural active yeast, and rice husk ash) to alumina matrix, the effect of the addition of Cu metal in nanoscale, on the physical and mechanical properties of porous alumina ceramics are analyzed and discussed in **Chapter 4**. Finally, the conclusion and suggestion for future works are showed in **Chapter 5**.

REFERENCES

- Ahmad, R., Ha, J.-H., and Song, I.-H. (2014). Effect of valeric acid on the agglomeration of zirconia particles and effects of the sintering temperature on the strut wall thickness of particle-stabilized foam. *Journal of the European Ceramic Society*, 34 (5), 1303-1310
- Ali, A. M., Abdullah, N. S., Ratnam, M. M., and Ahmad, Z. A. (2016). Linear Shrinkage of the ZTA Ceramic Cutting Inserts. *Procedia Chemistry*, *19*, 879-883
- Ali, E. S., and Ahmad, S. (2012). Bionanocomposite hybrid polyurethane foam reinforced with empty fruit bunch and nano clay. *Composites Part B: Engineering*, 43(7), 2813-2816
- Alman, D. E., and Hawk, J. A. (2001). Abrasive wear behavior of a brittle matrix (MoSi 2) composite reinforced with a ductile phase (Nb). *Wear*, 251(1), 890-900
- Antsiferov, V. P., SE. (2007). Enhancing strength of high-porous cordierite ceramics by mechanochemical activation of the charge. *Russian Journal of Non-Ferrous Metals*, 48 (6), 456-460
- Aramaki, S., and Roy, R. (1962). Revised Phase Diagram for the System Al₂O₃—SiO₂. Journal of the American Ceramic Society, 45(5), 229-242.
- ASTM, C1327-03, (2005). Vickers Indentation Hardness of Advanced Ceramics. ASTM-International, West Conshohocken, PA 19428-2959, https://www.astm.org, United States.
- ASTM, C1161-02c, (2005). Standard Test Method for Flexural Strength of Advanced Ceramics at Ambient Temperature". *ASTM-International, West Conshohocken, PA 19428-2959*, https://www.astm.org, United States.
- ASTM, C1424-10, (2005). standard test method for monotonic compressive strength of advanced ceramics at ambient temperature. *ASTM International, West Conshohocken, PA19428-2959*, https://www.astm.org, United States.
- ASTM, C20-00 (2005) Standard test methods for apparent porosity, water absorption, apparent specific gravity, and bulk density of burned refractory brick and shapes by boiling water. *ASTM International, West Conshohocken, PA 19428-*2959, https://www.astm.org, United States.
- ASTM. D3967, (2005) Standard Test Method for Splitting Tensile Strength of Intact Rock Core Specimens. *ASTM International, West Conshohocken, PA 19428-*2959, https://www.astm.org, United States.
- Manoj Kumar, B.V, J.-H. E., Young-Wook Kim, In-Sub Han And Sang-Kuk Woo. (118 (2010) 13-18). Effect of aluminum source on flexural strength of mullite-

bonded porous silicon carbide ceramics. *Journal of the Ceramic Society of Japan*(1)

- Bai, J. (2010). Fabrication and properties of porous mullite ceramics from calcined carbonaceous kaolin and α-Al 2 O 3. *Ceramics International*, *36* (2), 673-678s.
- Bakhtiari, F., and Darezereshki, E. (2011). One-step synthesis of tenorite (CuO) nanoparticles from Cu 4 (SO 4)(OH) 6 by direct thermal-decomposition method. *Materials Letters*, 65(2), 171-174
- Barham, N. L., Kaplan, W. D., and Rittel, D. (2014). Static and dynamic mechanical properties of alumina reinforced with sub-micron Ni particles. *Materials Science and Engineering: A*, 597, 1-9
- Bartolomé, J. F., Gutiérrez-González, C. F., and Torrecillas, R. (2008). Mechanical properties of alumina–zirconia–Nb micro–nano-hybrid composites. *Composites Science and Technology*, 68 (6), 1392-1398
- Benhammou, A., El Hafiane, Y., Abourriche, A., Abouliatim, Y., Nibou, L., Yaacoubi, A., (2014). Effects of oil shale addition and sintering cycle on the microstructure and mechanical properties of porous cordierite-ceramic. *Ceramics International*, 40 (7), 8937-8944
- Benhammou, A. E. H., Y Abourriche, A Abouliatim, Y Nibou, L Yaacoubi, A Tessier-Doyen, N Smith, A Tanouti, B. (2014). Effects of oil shale addition and sintering cycle on the microstructure and mechanical properties of porous cordierite-ceramic. *Ceramics International*, 40 (7), 8937-8944
- Boch, P., and Niepce, J.-C. (2010). *Ceramic Materials: Processes, Properties, and Applications* (Vol. 98): John Wiley & Sons.
- Bradt, R. C. (2008). The Sillimanite Minerals: Andalusite, Kyanite, and Sillimanite *Ceramic and Glass Materials* (pp. 41-48): Springer.
- Cao, J., Dong, X., Li, L., Dong, Y., and Hampshire, S. (2014). Recycling of waste fly ash for production of porous mullite ceramic membrane supports with increased porosity. *Journal of the European Ceramic Society*, 34(13), 3181-3194
- Cao, J., Lu, J., Jiang, L., andWang, Z. (2016). Sinterability, microstructure and compressive strength of porous glass-ceramics from metallurgical silicon slag and waste glass. *Ceramics International*, 42(8), 10079-10084
- Cardarelli, F. (2008). Ceramics, refractories, and glasses. *Materials Handbook: A Concise Desktop Reference*, 593-689
- Carter, C. B., and Norton, M. G. (2007). *Ceramic materials: science and engineering:* Springer Science & Business Media.

- Chae, S.-H. K., Young-Wook Song, In-Hyuek Kim, Hai-Doo Bae, Ji-Soo. (2009). Effects of Template Size and Content on Porosity and Strength of Macroporous Zirconia Ceramics. *Journal of The Korean Ceramic Society*, 46(1), 35-40
- Chen, R., and Tuan, W. (1999). Pressureless sintering of Al₂O₃/Ni nanocomposites. Journal of the European Ceramic Society, 19(4), 463-468
- Chen, R., andTuan, W. (2001). Toughening alumina with silver and zirconia inclusions. Journal of the European Ceramic Society, 21(16), 2887-2893
- Chi, W., Jiang, D., Huang, Z., and Tan, S. (2004). Sintering behavior of porous SiC ceramics. *Ceramics International*, 30 (6), 869-874
- Chi, W. J., Dongliang Huang, Zhengren Tan, Shouhong. (2004). Sintering behavior of porous SiC ceramics. *Ceramics International*, 30(6), 869-874
- Chmielewski, M., andPietrzak, K. (2007). Processing, microstructure and mechanical properties of Al 2 O 3–Cr nanocomposites. *Journal of the European Ceramic Society*, 27(2), 1273-1279
- Choi, Y.-H. K., Young-Wook Woo, Sang-Kuk Han, In-Sub. (47, No. 6 (2010) 509~514,). Effect of Template Content on Microstructure and Flexural Strength of Porous Mullite-Bonded Silicon Carbide Ceramics. *Journal of The Korean Ceramic Society*
- Chou, W., and Tuan, W. (1995). Toughening and strengthening of alumina with silver inclusions. *Journal of the European Ceramic Society*, 15(4), 291-295
- Choy, M.-T., Tang, C.-Y., Chen, L., Law, W.-C., Tsui, C.-P., and Lu, W. W. (2015). Microwave assisted-in situ synthesis of porous titanium/calcium phosphate composites and their in vitro apatite-forming capability. *Composites Part B: Engineering*, 83, 50-57
- Chung, D. D. (2010). *Composite materials: science and applications:* Springer Science & Business Media.
- Clegg, R. E., and Paterson, G. (2004). *Ductile particle toughening of hydroxyapatite ceramics using platinum particles*. Structural Integrity and Fracture International Conference (SIF'04), Brisbane, pp. 47-53.
- Cook, S. G., Little, J. A., and King, J. E. (1995). Etching and microstructure of engineering ceramics. *Materials Characterization*, 34 (1), 1-8
- David Linden, a. T. B. R., and Thomas B, R. (2002). Hand book of batteries *Book,McGraw-Hill, chapter* 8 (Third edition)
- Deng, Z.-Y., Fukasawa, T., Ando, M., Zhang, G.-J., and Ohji, T. (2001). Bulk alumina support with high tolerant strain and its reinforcing mechanisms. Acta materialia, 49 (11), 1939-1946

- Dessai, R. R. D., JAE Sen, D Mazumder, S. (2013). Effects of pressure and temperature on pore structure of ceramic synthesized from rice husk: A small angle neutron scattering investigation. *Journal of Alloys and Compounds*, 564, 125-129
- Ding S huqiang , Z., Zeng Yu-Ping, Jiang Dongliang. (2007). Fabrication of mullitebonded porous silicon carbide ceramics by in situ reaction bonding. *Journal of the European Ceramic Society*, 27(4), 2095-2102
- Ding, S., Zhu, S., Zeng, Y.-P., and Jiang, D. (2007). Fabrication of mullite-bonded porous silicon carbide ceramics by in situ reaction bonding. Journal of the European Ceramic Society, 27(4), 2095-2102
- Dittmann, J. W., Norbert, and Franks, G. (2014). Micro Structural Investigations and Mechanical Properties of Macro Porous Ceramic Materials from Capillary Suspensions. *Journal of the American Ceramic Society*, 97(12), 3787-3792
- Donald, W. O. (2008). Graphite. Minerals Yearbook, Metals and Minerals, By Geological Survey (US), 1
- Dong, X., Wang, M., Guo, A., Zhang, Y., Ren, S., Sui, G. (2017). Synthesis and properties of porous alumina ceramics with inter-locked plate-like structure through the tert-butyl alcohol-based gel-casting method. *Journal of Alloys and Compounds, 694*, 1045-1053
- Dong, Y., Zhou, J.-e., Lin, B., Wang, Y., Wang, S., Miao, L.(2009). Reaction-sintered porous mineral-based mullite ceramic membrane supports made from recycled materials. *Journal of hazardous materials*, 172 (1), 180-186
- Dong, Y. W., Chang-An Zhou, Jun Hong, Zhanglian. (2012). A novel way to fabricate highly porous fibrous YSZ ceramics with improved thermal and mechanical properties. *Journal of the European Ceramic Society*, 32(10), 2213-2218
- Dong, Z., andChen, W. (2013). Synthesis and hardness evaluation of porous M (Cr, Co) 7 C 3–Co composites. *Materials Science and Engineering: A*, 576, 52-60
- Elsen, S. R., and Ramesh, T. (2015). Optimization to develop multiple response hardness and compressive strength of zirconia reinforced alumina by using RSM and GRA. *International Journal of Refractory Metals and Hard Materials*, 52, 159-164
- Eom, J.-H., Kim, Y.-W., Song, I.-H., and Kim, H.-D. (2007). Microstructure and properties of porous silicon carbide ceramics fabricated by carbothermal reduction and subsequent sintering process. *Materials Science and Engineering: A*, 464(1), 129-134
- Eom, J.-H., Kim, Young-Wook, Raju, Santosh. (2013). Processing and properties of macroporous silicon carbide ceramics: A review. *Journal of Asian Ceramic Societies*, 1(3), 220-242

- Eom, J.-H. K., Young-Wook. (2009). Effect of additive composition on microstructure and strength of porous silicon carbide ceramics. *Journal of Materials Science*, 44(16), 4482-4486
- Eom, J.-H. K., Young-Wook Song, In-Hyuck Kim, Hai-Doo. (2008). Processing and properties of polysiloxane-derived porous silicon carbide ceramics using hollow microspheres as templates. *Journal of the European Ceramic Society*, 28(5), 1029-1035
- Fairhurst, C. (1964). On the validity of the 'Brazilian'test for brittle materials. Paper presented at the International Journal of Rock Mechanics and Mining Sciences & Geomechanics Abstracts, 1(4), 535-546.
- Falamaki, C., Aghaei, Alireza, Ardestani, Navid Razavi. (2001). RBAO membranes/catalyst supports with enhanced permeability. *Journal of the European Ceramic Society*, 21(12), 2267-2274
- Feng, Y., Wang, K., Yao, J., Webley, P. A., Smart, S., and Wang, H. (2013). Effect of the addition of polyvinylpyrrolidone as a pore-former on microstructure and mechanical strength of porous alumina ceramics. *Ceramics International*, 39(7), 7551-7556
- Fukushima, M. (2013). Microstructural control of macroporous silicon carbide. *Journal* of the Ceramic Society of Japan, 121(1410), 162-168
- Fung, Y.-L. E. W., Huanting. (2013). Investigation of reinforcement of porous alumina by nickel aluminate spinel for its use as ceramic membrane. *Journal of Membrane Science*, 444, 252-258
- Gan, M., and Wang, J. (2012). Applications of image processing technique in porous material characterization. In *Advanced Image Acquisition, Processing Techniques and Applications I.* InTech. Shanghai, China.
- German, R. M., Suri, P., and Park, S. J. (2009). Review: liquid phase sintering. *Journal* of Materials Science, 44(1), 1-39
- Görhan, G., andŞimşek, O. (2013). Porous clay bricks manufactured with rice husks. *Construction and Building Materials*, 40, 390-396
- Grigoriev, O., Karoteev, A., Maiboroda, E., Berezhinsky, I., Serdega, B., Ostrovoi, D. Y., (2006). Structure, nonlinear stress-strain state and strength of ceramic multilayered composites. *Composites Part B: Engineering*, 37(6), 530-541
- Gu, M., Huang, C., Zou, B., and Liu, B. (2006a). Effect of (Ni, Mo) and TiN on the microstructure and mechanical properties of TiB2 ceramic tool materials. *Materials Science and Engineering: A*, 433(1-2), 39-44

- Gu, M., Huang, C., Zou, B., and Liu, B. (2006b). Effect of (Ni, Mo) and TiN on the microstructure and mechanical properties of TiB₂ ceramic tool materials. *Materials Science and Engineering: A*, 433(1), 39-44
- Hammel, E., Ighodaro, O.-R., and Okoli, O. (2014). Processing and properties of advanced porous ceramics: An application based review. *Ceramics International*, 40 (10), 15351-15370
- Han, M., Yin, X., Cheng, L., Ren, S., and Li, Z. (2017). Effect of core-shell microspheres as pore-forming agent on the properties of porous alumina ceramics. *Materials* & Design, 113, 384-390
- Hu Liangfa, Benitez, R., Basu, S., Karaman, I., and Radovic, M. (2012). Processing and characterization of porous Ti 2 AlC with controlled porosity and pore size. Acta Materialia, 60(18), 6266-6277
- Hu LiangFa and Wang, C.-A. (2010). Effect of sintering temperature on compressive strength of porous yttria-stabilized zirconia ceramics. *Ceramics International*, 36(5), 1697-1701
- Hua, K., Shui, A., Xu, L., Zhao, K., Zhou, Q., and Xi, X. (2016a). Fabrication and characterization of anorthite-mullite-corundum porous ceramics from construction waste. *Ceramics International*
- Hua, K., Shui, A., Xu, L., Zhao, K., Zhou, Q., andXi, X. (2016b). Fabrication and characterization of anorthite-mullite-corundum porous ceramics from construction waste. *Ceramics International*, 42 (5), 6080-6087
- Hussain, F., Hojjati, M., Okamoto, M., andGorga, R. E. (2006). Review article: polymermatrix nanocomposites, processing, manufacturing, and application: an overview. *Journal of Composite Materials*, 40 (17), 1511-1575
- Ighodaro, O. L., Okoli, O. I., Zhang, M., and Wang, B. (2012). Ceramic Preforms with 2D Regular Channels for Fabrication of Metal/Ceramic-Reinforced Composites. *International Journal of Applied Ceramic Technology*, 9(2), 421-430
- Jancar, J. (2008). Review of the role of the interphase in the control of composite performance on micro-and nano-length scales. *Journal of Materials Science*, 43(20), 6747-6757
- Jean, G. S., Valérie Demuynck, Maryse Cambier, Francis Gonon, Maurice. (2014). Macroporous ceramics: Novel route using partial sintering of alumina-powder agglomerates obtained by spray-drying. *Ceramics International*, 40(7), 10197-10203
- Ji, Y., andYeomans, J. (2002). Processing and mechanical properties of Al 2 O 3–5 vol.% Cr nanocomposites. *Journal of the European Ceramic Society*, 22(12), 1927-1936

- Junkes, J. A. D., Benjamin Gutbrod, Björn Hotza, Dachamir Greil, Peter Travitzky, Nahum. (2013). Influence of coatings on microstructure and mechanical properties of preceramic paper-derived porous alumina substrates. *Journal of Materials Processing Technology*, 213(2), 308-313
- Kafkaslıoğlu, B., andTür, Y. K. (2016). Pressureless sintering of Al₂O₃/Ni nanocomposites produced by heterogeneous precipitation method with varying nickel contents. *International Journal of Refractory Metals and Hard Materials*, *57*, 139-144
- Kawai, C. (2001). Effect of grain size distribution on the strength of porous Si3N4 ceramics composed of elongated β -Si3N4 grains. *Journal of Materials Science*, 36(23), 5713-5717
- Kayal, N., Dey, A., andChakrabarti, O. (2012). Synthesis of mullite bonded porous SiC ceramics by a liquid precursor infiltration method: Effect of sintering temperature on material and mechanical properties. *Materials Science and Engineering: A*, 556, 789-795
- Kennedy, G. P., Lim, K.-Y., Kim, Y.-W., Song, I.-H., and Kim, H.-D. (2011). Effect of SiC particle size on flexural strength of porous self-bonded SiC ceramics. *Metals and Materials International*, 17 (4), 599-605
- Kennedy, L. J., Vijaya, J. J., and Sekaran, G. (2005). Electrical conductivity study of porous carbon composite derived from rice husk. *Materials chemistry and physics*, *91*(2), 471-476
- Kim, J.-H. E. a. Y.-W. (2008). Effect of template size on microstructure and strength of porous silicon carbide ceramics *Journal of the Ceramic Society of Japan* 116(10), 1159-1163
- Konrad, C. H., Völkl, R., andGlatzel, U. (2014). A novel method for the preparation of porous zirconia ceramics with multimodal pore size distribution. *Journal of the European Ceramic Society*, 34(5), 1311-1319
- Kumar, A., Mohanta, K., Kumar, D., and Parkash, O. (2014). Green properties of drypressed alumina compacts fabricated using sucrose as binder. *Ceramics International*, 40 (4), 6271-6277
- Kumar, B. M., Eom, J.-H., Kim, Y.-W., Song, I.-H., and Kim, H.-D. (2011). Effect of aluminum hydroxide content on porosity and strength of porous mullite-bonded silicon carbide ceramics. *Journal of the Ceramic Society of Japan*, 119 (1389), 367-370
- Lalande, J., Scheppokat, S., Janssen, R., and Claussen, N. (2002). Toughening of alumina/zirconia ceramic composites with silver particles. *Journal of the European Ceramic Society*, 22(13), 2165-2171

- Lange, F. F. (1989). Thermodynamics of densification: II, grain growth in porous compacts and relation to densification. *Journal of the American Ceramic Society*, 72(5), 735-741
- Le Huec, J. S., Tclement, Dfaber, J Le Rebeller, A. (1995). Influence of porosity on the mechanical resistance of hydroxyapatite ceramics under compressive stress. *Biomaterials*, 16 (2), 113-118
- L. F. Hu and C.-A. Wang (2010), "Effect of sintering temperature on compressive strength of porous yttria-stabilized zirconia ceramics," *Ceramics International, vol. 36, no. 5, pp.* 1697–1701,
- Lemes-Rachadel, P., Birol, H., Oliveira, A., and Hotza, D. (2012). Development of alternative glass ceramic seal for a planar solid oxide fuel cell. *Advances in Materials Science and Engineering*, 2012
- Li, G.-J., Huang, X.-X., and Guo, J.-K. (2003). Fabrication, microstructure and mechanical properties of Al 2 O 3/Ni nanocomposites by a chemical method. *Materials Research Bulletin*, 38(11), 1591-1600
- Li, G. F., Yiqun Zheng, Yuan Wu, Yuping. (2010). Preparation and properties of high toughness RBAO macroporous membrane support. *Ceramics International*, 36 (7), 2025-2031
- Li, S., Wang, C.-A., and Zhou, J. (2013). Effect of starch addition on microstructure and properties of highly porous alumina ceramics. *Ceramics International*, 39(8), 8833-8839
- Lieberthal, M., andKaplan, W. D. (2001). Processing and properties of Al₂O₃ nanocomposites reinforced with sub-micron Ni and NiAl₂O₄. *Materials Science and Engineering: A, 302*(1), 83-91
- Lin, Z., Li, M., and Zhou, Y. (2007). TEM investigations on layered ternary ceramics. *J. Mater. Sci. Technol*, 23(2), 145-165
- Liu, D.-M. (1997). Influence of porosity and pore size on the compressive strength of porous hydroxyapatite ceramic. *Ceramics International*, 23(2), 135-139
- Liu, D.-M., and Tuan, W. (1997). Microstructure and its influence on thermal and electrical conductivity of ZrO₂–Ag composites. *Materials Chemistry and Physics*, 48(3), 258-262
- Liu, S., Zeng, Y. P., and Jiang, D. (2009). Effects of Preheat-Treated Aluminosilicate Addition on the Phase Development, Microstructure, and Mechanical Properties of Mullitized Porous OBSC Ceramics. *International Journal of Applied Ceramic Technology*, 6(5), 617-625
- Liu, Y., Zhou, J., and Shen, T. (2013). Effect of nano-metal particles on the fracture toughness of metal–ceramic composite. *Materials & Design*, 45, 67-71

- Lu, J., Gao, L., Sun, J., Gui, L., andGuo, J. (2000). Effect of nickel content on the sintering behavior, mechanical and dielectric properties of Al 2 O 3/Ni composites from coated powders. *Materials Science and Engineering: A*, 293(1), 223-228
- Luo, J.-J., and Daniel, I. M. (2003). Characterization and modeling of mechanical behavior of polymer/clay nanocomposites. *Composites Science and Technology*, 63(11), 1607-1616
- Lyckfeldt, O., and Ferreira, J. (1998). Processing of porous ceramics by 'starch consolidation'. *Journal of the European Ceramic Society*, 18 (2), 131-140
- Malaiskiene, J., Skripkiunas, G., Vaiciene, M., and Kizinievic, O. (2016). The influence of mullite wool waste on the properties of concrete and ceramics. *Construction and Building Materials*, *110*, 8-16
- Mao, J. Z., FL Liao, GC Zhou, Ym huang, HP Wang, Cywu, SH. (2014). Effect of granulated sugar as pore former on the microstructure and mechanical properties of the vitrified bond cubic boron nitride grinding wheels. *Materials & Design*, 60, 328-333
- Mashhadi, M., Taheri-Nassaj, E., Mashhadi, M., andSglavo, V. M. (2011). Pressureless sintering of B 4 C–TiB 2 composites with Al additions. *Ceramics International*, 37(8), 3229-3235
- Mashhadi, M., Taheri-Nassaj, E., Sglavo, V. M., Sarpoolaky, H., and Ehsani, N. (2009). Effect of Al addition on pressureless sintering of B 4 C. *Ceramics International*, 35(2), 831-837
- Matori, K. A., Haslinawati, M., Wahab, Z., Sidek, H., Ban, T., and Ghani, W. (2009). Producing amorphous white silica from rice husk. *MASAUM Journal of Basic* and Applied Sciences, 1(3), 512
- Meille, S., Lombardi, M., Chevalier, J., and Montanaro, L. (2012). Mechanical properties of porous ceramics in compression: On the transition between elastic, brittle, and cellular behavior. *Journal of the European Ceramic Society*, 32 (15), 3959-3967
- Menchavez, Ruben L, Intong, andLori-Ann S. (2010). Red clay-based porous ceramic with pores created by yeast-based foaming technique. *Journal of materials science*, 45 (23), 6511-6520
- Mohanta, K., Kumar, A., Parkash, O., and Kumar, D. (2014). Processing and properties of low cost macroporous alumina ceramics with tailored porosity and pore size fabricated using rice husk and sucrose. *Journal of the European Ceramic Society*, *34* (10), 2401-2412

- Moya, J. S., Díaz, M., Gutiérrez-González, C. F., Diaz, L. A., Torrecillas, R., andBartolomé, J. F. (2008). Mullite-refractory metal (Mo, Nb) composites. *Journal of the European Ceramic Society*, 28 (2), 479-491
- Mubarak Ali, D., Arunkumar, J., Pooja, P., Subramanian, G., Thajuddin, N., and Alharbi, N. S. (2015). Synthesis and characterization of biocompatibility of tenorite nanoparticles and potential property against biofilm formation. *Saudi Pharmaceutical Journal*, 23(4), 421-428
- Negahdari, Z., Willert-Porada, M., and Pfeiffer, C. (2010). Mechanical properties of dense to porous alumina/lanthanum hexaaluminate composite ceramics. *Materials Science and Engineering: A*, 527(12), 3005-3009
- Nie, Z., and Lin, Y. (2015). Fabrication of Porous Alumina Ceramics With Corn Starch In An Easy and Low-Cost Way. *Ceramics–Silikáty*, 59 (4), 348-352
- Novais, R. M. S., MP Labrincha, JA. (2014). Ceramic tiles with controlled porosity and low thermal conductivity by using pore-forming agents. *Ceramics International*, 40(8), 11637-11648
- Oh, S.-T., Lee, Jai-Sung, Sekino, Tohru, Niihara, Koichi. (2001). Fabrication of Cu dispersed Al 2 O 3 nanocomposites using Al₂O₃/CuO and Al₂O₃/Cu-nitrate mixtures. *Scripta Materialia*, 44(8), 2117-2120
- Oh, S.-T., Sando, M., Sekino, T., and Niihara, K. (1998). Processing and properties of copper dispersed alumina matrix nanocomposites. *Nanostructured Materials*, 10(2), 267-272
- OH, U. C., Chung, Y. S., Kim, D. Y., and Yoon, D. N. (1988). Effect of Grain Growth on Pore Coalescence During the Liquid-Phase Sintering of MgO-CaMgSiO₄ Systems. *Journal of the American Ceramic Society*, 71(10), 854-857
- Ohji, T., and Fukushima, M. (2012). Macro-porous ceramics: processing and properties. *International Materials Reviews*, 57(2), 115-131
- Oliver, W. C., and Pharr, G. M. (2004). Measurement of hardness and elastic modulus by instrumented indentation: Advances in understanding and refinements to methodology. *Journal of Materials Research*, *19*(01), 3-20
- Park, Y., Yang, T., Yoon, S., Stevens, R., and Park, H. (2007). Mullite whisker derived from coal fly ash. *Materials Science and Engineering: A*, 454, 518-522
- Piazza, D., Capiani, C, Galassi, C, (2005). Piezoceramic material with anisotropic graded porosity. *Journal of the European Ceramic Society*, 25(12), 3075-3078
- Prabhakaran, K., Melkeri, A., Gokhale, N., andSharma, S. (2007). Preparation of macroporous alumina ceramics using wheat particles as gelling and pore forming agent. *Ceramics International*, 33(1), 77-81

- Rahaman, M. N. (2006). *Ceramic Processing and Sintering*: CRC press, Taylor & Francis Group, UK.
- Rahmawati, F., Perkasa, N. M., Purwanto, A., & Nizam, M. (2014, November). The performance of LiFePO₄ battery with graphite waste as anode material. In Electrical Engineering and Computer Science (ICEECS), 2014 International Conference on (pp. 209-211). IEEE.
- Ramakrishna, P. V., Murthy, D. B. R. K., Sastry, D. L., & Samatha, K. (2014). Synthesis, structural and luminescence properties of Mn doped ZnO/Zn₂SiO₄ composite microphosphor. *Spectrochimica Acta Part A: Molecular and Biomolecular Spectroscopy*, 129, 274-279.
- Re, G. L., Lopresti, F., Petrucci, G., and Scaffaro, R. (2015). A facile method to determine pore size distribution in porous scaffold by using image processing. *Micron*, 76, 37-45
- Rong, S.-f., JI, Z.-s., Zhu, Y.-c., and Zhang, J.-q. (2007). Effect of Cu on Microstructure and Properties of Al₂O₃ Diphase Ceramic. *Journal of Iron and Steel Research*, *International*, 14(5), 90-93
- Rösler, J., Harders, H., and Baeker, M. (2007). *Mechanical behaviour of engineering materials: metals, ceramics, polymers, and composites*: Springer Science & Business Media.
- Rosso, M. (2006). Ceramic and metal matrix composites: Routes and properties. *Journal* of Materials Processing Technology, 175(1), 364-375
- Sbaizero, O., and Pezzotti, G. (2000). Influence of the metal particle size on toughness of Al₂O ₃/Mo composite. *Acta Materialia*, 48 (4), 985-992
- Sbaizero, O., and Pezzotti, G. (2001). Residual stresses and R-curve behavior of AlN/Mo composite. *Journal of the European Ceramic Society*, 21(3), 269-275
- Sbaizero, O., Pezzotti, G., and Nishida, T. (1998). Fracture energy and R-curve behavior of Al 2 O 3/Mo composites. *Acta Materialia*, 46 (2), 681-687
- Scott, D. A. (2002). *Copper and bronze in art: corrosion, colorants, conservation*: Getty publications, Los Angeles.
- Sedaghat, A., Taheri-Nassaj, E., Soraru, G., and Ebadzadeh, T. (2013). Microstructure Development and Phase Evolution of Alumina-mullite Nanocomposite. *Science of Sintering*, 45 (3)
- Seeber, B. S. M., Gonzenbach, Urs Thomas, Gauckler, Ludwig Julius. (2013). Mechanical properties of highly porous alumina foams. *Journal of Materials Research*, 28 (17), 2281-2287

- Sekino, T., Nakajima, T., and Niihara, K. (1996). Mechanical and magnetic properties of nickel dispersed alumina-based nanocomposite. *Materials Letters*, 29 (1), 165-169
- Sengphet, K. P., K Sato, Tsutomu Fauzi, MN Ahmad Radzali, O. (3,N 8,(2013) 2250-3153). Fabrication of Porous Clay Ceramics Using Kenaf Powder Waste. International Journal of Scientific and Research Publications, International Journal of Scientific and Research Publications.
- Serra, M., Conconi, M., Gauna, M., Suárez, G., Aglietti, E., and Rendtorff, N. (2016). Mullite (3Al₂O₃· 2SiO₂) ceramics obtained by reaction sintering of rice husk ash and alumina, phase evolution, sintering and microstructure. *Journal of Asian Ceramic Societies*, 4 (1), 61-67
- Shaw, T. M. (1993). Model for the Effect of Powder Packing on the Driving Force for Liquid-Phase Sintering. *Journal of the American Ceramic Society*, 76 (3), 664-670
- Smirnov, A., andBartolomé, J. (2014). Microstructure and mechanical properties of ZrO 2 ceramics toughened by 5–20vol% Ta metallic particles fabricated by pressureless sintering. *Ceramics International*, 40(1), 1829-1834
- Soltani, N., Bahrami, A., Pech-Canul, M., and González, L. (2015). Review on the physicochemical treatments of rice husk for production of advanced materials. *Chemical Engineering Journal*, 264, 899-935
- Somiya, S., Aldinger, F., Claussen, N., Spriggs, R. M., Uchino, K., Koumoto, K., (2003). Handbook of Advanced Ceramics, Volume II Processing and their Applications: Elsevier Inc.
- Studart, A. R., Gonzenbach, U. T., Tervoort, E., and Gauckler, L. J. (2006). Processing routes to macroporous ceramics: a review. *Journal of the American Ceramic Society*, 89(6), 1771-1789
- Studart, A. R. S., J. Xu, L. Yoon, K. Shum, H. C. Weitz, D. A. (2011). Hierarchical porous materials made by drying complex suspensions. *Langmuir*, 27(3), 955-964
- Täffner, U., Carle, V., Schäfer, U., and Hoffmann, M. (2004). Preparation and microstructural analysis of high-performance ceramics *Metallography and Microstructures* (pp. 1057-1066): ASM International.
- Tang, F. F., Hiroshi Uchikoshi, Tetsuo Sakka, Yoshio. (2004). Preparation of porous materials with controlled pore size and porosity. *Journal of the European Ceramic Society*, 24(2), 341-344
- Thostenson, E. T., Li, C., and Chou, T.-W. (2005). Nanocomposites in context. *Composites Science and Technology*, 65(3), 491-516

- Tomba, A., Camerucci, M., Urretavizcaya, G., Cavalieri, A., Sainz, M., and Caballero, A. (1999). Elongated mullite crystals obtained from high temperature transformation of sillimanite. *Ceramics International*, 25(3), 245-252
- Tripathi, H. S., Mukherjee, B., Das, S. K., Ghosh, A., and Banerjee, G. (2003). Effect of sillimanite beach sand composition on mullitization and properties of Al₂O₃-SiO₂ system. *Bulletin of Materials Science*, 26(2), 217-220
- Trusty, P., andYeomans, J. (1997). The toughening of alumina with iron: effects of iron distribution on fracture toughness. *Journal of the European Ceramic Society*, 17 (4), 495-504
- Vekinis, G., Sofianopoulos, E., and Tomlinson, W. (1997). Alumina toughened with short nickel fibres. *Acta Materialia*, 45(11), 4651-4661
- Veljović, D., Jančić-Hajneman, R., Balać, I., Jokić, B., Putić, S., Petrović, R. (2011). The effect of the shape and size of the pores on the mechanical properties of porous HAP-based bioceramics. *Ceramics International*, 37 (2), 471-479
- Venkataraman, R., Das, G., Singh, S., Pathak, L., Ghosh, R., Venkataraman, B. (2007). Study on influence of porosity, pore size, spatial and topological distribution of pores on microhardness of as plasma sprayed ceramic coatings. *Materials Science and Engineering: A*, 445, 269-274
- Verhaeghe, B., Courtois, C., Petit, F., Cambier, F., Guérin, J.-D., Leriche, A. (2014). Lighter tableware ceramic by controlling porosity: Effect of porosity on mechanical properties. *Ceramics International*, 40(1), 763-770
- Wang, J., Xie, H., Guo, Z., Guan, L., and Li, Y. (2014). Improved thermal properties of paraffin wax by the addition of TiO 2 nanoparticles. Applied Thermal Engineering, 73 (2), 1541-1547
- Wang, S.-r. G., Hao-ran Hui, Lin-hai Wang, Ying-zi. (2007). Reticulated porous multiphase ceramics with improved compressive strength and fracture toughness. *Journal of Materials Engineering and Performance*, 16(1), 113-118
- Wang, S., Jia, D., Yang, Z., Duan, X., Tian, Z., and Zhou, Y. (2013). Effect of BN content on microstructures, mechanical and dielectric properties of porous BN/Si₃N₄ composite ceramics prepared by gel casting. *Ceramics International*, 39(4), 4231-4237
- Wang, X.-G., Guo, W.-M., Kan, Y.-M., Zhang, G.-J., and Wang, P.-L. (2011). Densification behavior and properties of hot-pressed ZrC ceramics with Zr and graphite additives. *Journal of the European Ceramic Society*, 31(6), 1103-1111
- Webb, P. A. (2001). Volume and density determinations for particle technologists. *Micromeritics Instrument Corp*, 2(16), 01

- Wei, G., Hongbin, L., and Chunxia, F. (2010). Influence of La 2 O 3 on preparation and performance of porous cordierite from rice husk. *Journal of rare earths*, 28(4), 614-617
- William and David . (2010). Materials Science and Engineering. book eight edition (John Wiley & Sons, Inc), chapter 12
- Wu, C.-S. (2003). Physical properties and biodegradability of maleatedpolycaprolactone/starch composite. *Polymer Degradation and Stability*, 80(1), 127-134
- Xu, G., Li, J., Cui, H., He, Q., Zhang, Z., and Zhan, X. (2015). Biotemplated fabrication of porous alumina ceramics with controllable pore size using bioactive yeast as pore-forming agent. *Ceramics International*, 41(5), 7042-7047
- Xu, G., Ma, Y., Cui, H., Ruan, G., Zhang, Z., and Zhao, H. (2014). Preparation of porous mullite–corundum ceramics with controlled pore size using bioactive yeast as pore-forming agent. *Materials Letters*, 116, 349-352
- Yan, W., Li, N., and Han, B. (2010). Effects Of Sintering Temperature On Pore Characterisation And Strength Of Porous Corundum-Mullite Ceramics. *Journal of Ceramic Processing Research*, 11(3), 388-391
- Yang, J. F. G., Ji Qiang Zhang, Guo Jun Hayashi, Ichiro Ohji, Tatsuki. (280 (2004) 1231-1236). *Effects of pore morphology on the fabrication and mechanical properties of porous* _{Si3N4} *ceramics*. Paper presented at the Key Engineering Materials.
- Yi, F. W., Kun Yao, Jianfeng Webley, Paul Smart, Simon Wang, Huanting. (2013).
 Effect of the addition of polyvinylpyrrolidone as a pore-former on microstructure and mechanical strength of porous alumina ceramics. *Ceramics International [P]*, 39(7), 7551-7556
- Yin, L. Z., Xingui Yu, Jinshan, Wang, Honglei. (2016). Preparation of high porous silicon nitride foams with ultra-thin walls and excellent mechanical performance for heat exchanger application by using a protein foaming method. *Ceramics International*, 42(1), 1713-1719
- Yoshida, K. T., Hironori Murakami, Akira Miyata, Hiroshi. (2008). Influence of Pore Size on Fracture Strength of Porous Ceramics. *Journal of Solid Mechanics and Materials Engineering*, 2(8), 1060-1069
- Yu, P. W., JZ Yu, FL Yang, JF. (2011). Effect of pure β-Si₃N₄ powder on microstructure and mechanical properties of porous Si₃N₄ ceramics. Paper presented at the IOP Conference Series: *Materials Science and Engineering*.
- Zahedi, M., Khanjanzadeh, H., Pirayesh, H., and Saadatnia, M. A. (2015). Utilization of natural montmorillonite modified with dimethyl, dehydrogenated tallow quaternary ammonium salt as reinforcement in almond shell flour–

polypropylene bio-nanocomposites. *Composites Part B: Engineering*, 71, 143-151

- Zeng, T., Dong, X., Mao, C., Zhou, Z., and Yang, H. (2007). Effects of pore shape and porosity on the properties of porous PZT 95/5 ceramics. *Journal of the European Ceramic Society*, 27(4), 2025-2029
- Zhang, J.-y. Y., Feng. (2010). Effect of agarose content on microstructures and mechanical properties of porous silicon nitride ceramics produced by gelcasting. *Journal of Zhejiang University SCIENCE A*, *11*(10), 771-775
- Zhang, J., and Malzbender, J. (2015). Mechanical characterization of micro-and nanoporous alumina. *Ceramics International*, 41(9), 10725-10729
- Zhang, R. F., Daining Chen, Xiangmeng Pei, Yongmao. (2012). Effect of pre-oxidation on the microstructure, mechanical and dielectric properties of highly porous silicon nitride ceramics. *Ceramics International*, 38(7), 6021-6026
- Zhou, J., Fan, J.-p., Sun, G.-l., Zhang, J.-y., Liu, X.-m., Zhang, D.-h., (2015). Preparation and properties of porous silicon nitride ceramics with uniform spherical pores by improved pore-forming agent method. *Journal of Alloys and Compounds, 632*, 655-660
- Zhou, Z.-h. R., Jian-ming Zou, Jian-peng Zhou, Zhong-cheng Shen, Xiong-jun. (2007). Synthesis and structural characterization of macroporous bioactive glass. Journal of Central South University of Technology, 14, 301-304
- Zhu, L., Dong, Y., Hampshire, S., Cerneaux, S., and Winnubst, L. (2015). Waste-toresource preparation of a porous ceramic membrane support featuring elongated mullite whisker with enhanced porosity and permeance. *Journal of the European Ceramic Society*, 35(2), 711-721
- Zhu, Z., Wei, Z., Shen, J., Zhu, L., Xu, L., Zhang, Y. (2017). Fabrication and catalytic growth mechanism of mullite ceramic whisker using molybdenum oxide as catalyst. *Ceramics International*, 43(2), 2871-2875
- Žmindák, M., and Dudinský, M. (2012). Computational Modelling of Composite Materials Reinforced by Glass Fibers. *Procedia Engineering*, 48, 701-710
- Zuo, K. H., Jiang, D. L., Lin, Q. L., andZeng, Y.-p. (2007). Improving the mechanical properties of Al₂O₃/Ni laminated composites by adding Ni particles in Al₂O₃ layers. *Materials Science and Engineering: A, 443*(1), 296-300.