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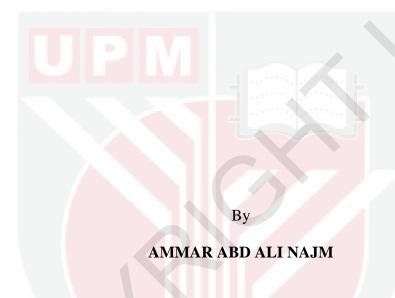
STRUCTURAL, ELECTRICAL AND MAGNETIC PROPERTIES OF BISMUTH FERRITE CERAMICS SUBSTITUTED WITH YTTRIUM AND INDIUM

AMMAR ABD ALI NAJM

FS 2016 42



STRUCTURAL, ELECTRICAL AND MAGNETIC PROPERTIES OF BISMUTH FERRITE CERAMICS SUBSTITUTED WITH YTTRIUM AND INDIUM



Thesis Submitted to the School of Graduate Studies, Universiti Putra Malaysia, in Fulfillment of the Requirement for the Degree of Master of Science



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DEDICATION

I dedicate this thesis to my father, mother, my brothers, my sisters and friends for their love and concern.



Abstract of thesis presented to the Senate of Universiti Putra Malaysia in fulfillment of the requirement for the degree of Master of Science

STRUCTURAL, ELECTRICAL AND MAGNETIC PROPERTIES OF BISMUTH FERRITE CERAMICS SUBSTITUTED WITH YTTRIUM AND INDIUM

By

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October 2016

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Faculty : Science

Multiferroic materials demonstrate the simultaneous presence of ferromagnetic, ferroelectric, or ferroelastic orderings. BiFeO₃ (BFO) is one of the significant multiferroic materials with high $T_C \sim 1103$ K and $T_N \sim 643$ K at room temperature. BFO suffers high leakage current and weak ferromagnetic and ferroelectric properties. This study was aimed to synthesize BiFe_{1-x}M_xO₃ (M = Y³⁺, In³⁺) samples; where x = 0, 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.8 and 1.0, investigate their phase formation due to Y³⁺ and In³⁺ substitution as well as their magnetic and electrical properties. Solid-state technique was used to synthesize BiFe_{1-x}M_xO₃, (M = Y³⁺, In³⁺) using Bi₂O₃, Fe₂O₃, Y₂O₃ and In₂O₃ as raw materials. XRD, SEM and EDX were used to determine the crystal structure, morphology of the grain size and elemental compositions respectively. Their leakage current, dielectric and magnetic properties were quantified by Keithley source measure unit, Impedance analyzer and VSM respectively.

XRD revealed the hexagonal single phase of pure BFO. The phase changed to cubic with Y^{3+} substitution and BFO remains the primary phase until x=0.2. Substitution of In^{3+} promotes the growth of $Bi_{25}FeO_{40}$, and BFO remains the primary phase until x=0.4. For SEM results, the average grain size of pure BFO decreased from 2.04 to 0.29 μ m with Y^{3+} substitution, while it decreases to 0.32 μ m for In^{3+} substitution. EDX revealed no impurities in the pure and substituted samples. From magnetic analysis, pure BFO shows antiferromagnetic behavior. A maximum M_s value of 2.9 emu/g and M_r of 0.09 were observed with Y^{3+} substitution at x=0.2. The magnetic properties showed nonlinear dependent on In^{3+} substitution. The highest M_s value of 0.0405 emu/g and M_r of 6.22×10^{-4} emu/g was achieved at x=0.3. The dielectric measurement showed that the ε'_r of the samples increased from 26.5 at x=0 to 105 at x=0.4, with Y^{3+} substitution. The values also improved with In^{3+} substitution and reached an optimum value of 372 at x=0.6. The J-E measurement revealed that the leakage current density, J of x=1.0 (4.6 × 10⁻⁸ A/cm²) substituted with Y^{3+} is decreased significantly by about four order of magnitude compared to that of x=0

 $(9.24 \times 10^{-4} \text{ A/cm}^2)$. Moreover, the J of $x = 1.0 \ (1.51 \times 10^{-6} \text{ A/cm}^2)$ substituted with In³⁺ is decreased significantly by about three order of magnitude compared to that of $x = 0 \ (9.24 \times 10^{-4} \text{ A/cm}^2)$.

In conclusion, substituted BFO ceramics possess improved dielectric, magnetic properties and has reduced the leakage current. The prepared ceramics could be employed for several applications such as disk read/write heads and ceramic pressure sensor.



SIFAT STRUKTUR, ELEKTRIK DAN MAGNET SERAMIK BISMUT FERIT DENGAN PENGGANTIAAN ITERIUM DAN INDIUM

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Bahan multiferoik menunjukkan kehadiran serentak sifat feromagnet, feroelektrik, dan feroelastik. Salah satu bahan multiferoik yang mempunyai suhu curie, T_C ~1103K dan suhu Neel, T_N ~643 K yang tinggi pada keadaan suhu bilik ialah BiFeO₃ (BFO). Walaubagaimanapun, limitasi BFO ialah kebocoran arus yang tinggi di samping mempunyai sifat feromagnetik dan feroelektrik yang lemah. Oleh itu, kajian ini bertujuan untuk mensintesis selain mengkaji sifat magnet serta elektrik bagi bahan multiferoik BiFe_{1-x} M_x O₃ (M = Y³⁺, In³⁺) dengan penggantian Y³⁺ dan In³⁺ berdasarkan perubahan pembentukan fasa (x = 0, 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.8 dan 1.0).

Teknik keadaan pepejal konvensional telah digunakan untuk mensintesis BiFe_{1-x}M_xO₃, (M = Y^{3+} , In³⁺) dengan menggunakan Bi₂O₃, Fe₂O₃, Y₂O₃ dan In₂O₃ sebagai bahan mentah. XRD, SEM dan EDX telah digunakan untuk mengenal pasti struktur kristal, morfologi saiz butiran dan komposisi elemen. Manakala sifat dielektrik, sifat magnet dan kebocoran arus bahan telah diukur menggunakan penganalisis impedans VSM dan unit ukuran sumber Keithley.

Analisis menunjukkan fasa XRD tunggal heksagon bagi Walaubagaimanapun, fasa tersebut berubah kepada kubik dengan penggantian Y³⁺ namun BFO kekal sebagai fasa primer sehingga x = 0.2. Manakala, penggantian In³⁺ menggalakan pertumbuhan Bi₂₅FeO₄₀ namun BFO turut kekal sebagai fasa primer sehingga x = 0.4. Keputusan kajian SEM menunjukkan nilai purata saiz butiran BFO tulen menurun daripada 2.04 kepada 0.29 µm dengan penggantian Y³⁺. Berbeza dengan penggantian In³⁺ yang menurun sehingga 0.32 µm. Selain itu, analisis EDX mengesahkan tiada bendasing terdapat dalam sampel tulen mahupun sampel yang telah didopkan. Tambahan pula, BFO tulen mempunyai sifat antiferomagnetik dengan nilai maksimum M_s ialah 2.9 emu/g dan M_r ialah 0.09 emu/g apabila Y^{3+} digantikan pada x = 0.2. Sifat-sifat magnet menunjukkan kebergantungan tidak linear ke atas penggantian In³⁺. Nilai tertinggi M_s ialah 0.0405 emu/g dan M_r ialah 6.22 ×

 10^{-4} emu/g, masing-masing dicapai pada x=0.3. Ukuran diaelektrik membuktikan bahawa ϵ'_r sampel meningkat daripada 26.5 pada x=0 kepada 105 pada x=0.4, dengan penggantian Y^{3+} . Nilai tersebut juga semakin bertambah baik dengan penggantian In^{3+} sehingga mencapai nilai optimum iaitu 372 pada x=0.6. Pengukuran J-E mendapati bahawa kebocoran ketumpatan arus, J ialah 4.6×10^{-8} A/cm² pada x=1.0 dengan penggantian Y^{3+} telah menurun dengan ketara kira-kira sebanyak empat turutan magnitud berbanding dengan keadaan pada x=0 iaitu 9.24×10^{-4} A/cm². Tambahan itu, nilai J pada x=1.0 ialah 1.51×10^{-6} A/cm² dengan penggantian In^{3+} telah menurun dengan ketara kira-kira sebanyak tiga turutan magnitud berbanding dengan keadaan pada x=0 iaitu 9.24×10^{-4} A/cm².

Kesimpulannya, pendopan seramik BFO dapat menambah baik sifat dielektrik, sifat magnet selain dapat mengurangkan kebocoran arus turut dapat kepada nilai yang agak rendah. Seramik yang telah disintesis boleh digunakan untuk beberapa aplikasi seperti cakera membaca/menulis atau sensor tekanan seramik.

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This thesis was submitted to the senate of Universiti Putra Malaysia and has been accepted as fulfillment of the requirement for the degree of Master of Science. The members of the Supervisory Committee were as follows:

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LIST OF ABBREVIATIONS AND SYMBOLS

BFO BiFeO₃

XRD X-ray Diffraction

SEM Scanning Electron Microscope

EDX Electron Dispersion X-ray

VSM Vibrating Sample Magnetometer

 Y_2O_3 Yttrium Oxide In_2O_3 Indium Oxide

Bi₂O₃ Bismuth Oxide

Fe₂O₃ Iron Oxide

T Temperature

T_C Curie temperature

T_N Neel temperature

 H_c Coercivity Field

M_r Remnant magnetization

M_s Saturated magnetization

ME Magnetoelectric

FM Ferromagnetic

FE Ferroelectric

E Electric field

J Leakage current density

θ Diffraction angle

λ Wavelength

ICSD Inorganic Crystal Structure Database

PVA Polyvinylalcohol

SCLC Space Charge Limited Current

CHAPTER 1

INTRODUCTION

1.1 Background of multiferroic

In the current age of device miniaturization, multiferroics are technologically significant. It involves the coexistence of two or more ferroic order parameters viz. ferroelectricity, ferromagnetism and ferroelasticity in a single phase (Schmid, 1994). As a logical definition to the term multiferroic, it is any material presenting two of these three ferroic properties. However, the most interesting combination was thought to be materials presenting ferroelectricity and ferromagnetism simultaneously. Nevertheless, the most significant is to involve a strong coupling interaction between these two-ferroic orders. In multiferroic materials, the coupling interaction between the different order parameters can yield additional functionalities, such as a magnetoelectric (ME) effect (Ma *et al.*, 2011). Magnetoelectric effect gives place to extra degrees of freedom, which may permit magnetization to be switched by an electric field and polarization to be switched by a magnetic field (Yatom and Englman, 1969; Eerenstein *et al.*, 2006; Chu *et al.*, 2007).

Although there are multiferroic materials that are not magnetoelectric and vice versa, for fundamental reasons, the magnetoelectric coupling in single-phase materials is largest in multiferroic materials. For this reason, the development of these classes of materials is intimately related. Ferroelectric and ferromagnetic materials are characterized by their spontaneous polarization (electric or magnetic, respectively).

However, most materials do not exhibit a spontaneous order, but they do interact with applied fields. An electric field (E) produces an electric dipole moment and hence electric polarization (P) in the material. Conversely, a magnetic field (H) produces magnetization (M) and stress (s) produces stain (ε) (Velev *et al.*, 2011).

Multiferroism is observed in very few naturally available single-phase multiferroic systems. Moreover, commercial device engineering considerations impose further restrictions on the materials to exhibit ferroelectric/magnetic ordering at room temperature (RT) or close to RT. Boracites were possibly the first multiferroics materials identified (Khomskii, 2006), while others were soon to be found in nature, or synthesized artificially. Initially, most of the focus was on materials such as BiFeO₃, which have ferroelectric and magnetic transition temperatures close to or above RT (Roy *et al.*, 2012).

1.2 Multiferroic materials

Complex oxide materials display a varied range of properties especially due to various interactions that occur among the electronic degrees of freedom, structural and magnetic properties. H. Schmidt initially invented the expression "multiferroic" in 1994 to define multiferroic as a single phase material that has either two or three order parameters of ferroic which coexistence at the same phase, such ferroic are ferroelectricity, ferromagnetism, ferroelasticity and ferrotoroidic order which shows a strong coupling between the two ferroic orders (Schmid, 1994; Fiebig, 2005). Specifically, ferroelectric and ferromagnetic are materials with high technological relevance, and can be used in magneto-electric sensors driven magnetic data storage and recording devices (Spaldin and Fiebig, 2005).

Considering materials performing multiferroic properties, a coupling interaction that arises between ferroic parameters which yield additional characteristics, include magnetoelectric (ME) effect (Ma *et al.*, 2011). The incidence of ferromagnetic and ferroelectric orders in a material with a single-phase crystal structure is based on three conditions namely:

- i. Symmetry conditions.
- ii. The existence of sufficient structural building blocks which allows offcenter ion displacement, related to the ferroelectric spontaneous polarization or other different mechanism for ferroelectricity lone pair: BiFeO₃, BiMnO₃ or geometric thwarting e.g. YMnO₃ (Hill, 2000).
- iii. Magnetic-interaction pathways for the magnetic order, more commonly of super-exchange type (Gheorghiu *et al.*, 2013).

The expression "magnetoelectric", has lately become widespread, this term consists of not only ferroelectromagnets, but also with the materials upon which any two ferroic parameters coexist such as ferroelectric materials, antiferroelectric materials, ferromagnetic materials, antiferromagnetic materials, ferromagnetic materials, ferroelastic materials and ferrotoroidic materials (Eerenstein *et al.*, 2006).

1.3 Types of multiferroics

To comprehend all fundamental phenomena in the multiferroic field, it is important to classify multiferroics according to the different basic mechanisms into two types. Recently, multiferroic materials have been categorized into two sorts: Type I and Type II (Khomskii, 2009). The magnetism and ferroelectricity in Type I resulted from different sources and the influence independent of each other, but unfortunately the degree of coupling between the magnetism and ferroelectricity, is often weak. The spontaneous polarization (P) of such materials is usually large of the order (10 - $100~\mu\text{C/cm}^2$) and the best example of such materials is bismuth ferrite (BiFeO₃) (T_C $\approx 1103\text{K}$, T_N $\approx 643~\text{K}$, P ~ 88 - $100~\mu\text{C/cm}^2$) (Xie *et al.*, 2014), Yttrium manganite

(YMnO₃) ($T_C \approx 914~K$, $T_N \approx 76~K$, $P \sim 6~\mu C/cm^2$). In type II multiferroic materials such as TbMnO₃, Ni₃V₂O₆ and MnWO₆, magnetism causes coexistence of ferroelectricity, which implies a strong coupling between the two. A much smaller polarization ($10^{-2}~\mu C/cm^2$), is displayed in the presence of its magnetized state (Kimura *et al.*, 2003).

1.4 Multiferroic bismuth ferrite

Bismuth ferrite (BiFeO₃: BFO) is a prototype multiferroic among all the novel multiferroic materials that are currently in use. BiFeO₃ has been widely considered in the form of ceramics, thin films and nano-powders (Simões et al., 2011). Generally, BiFeO₃ is denoted by BFO in the field of materials science. BFO consists of bismuth (Bi), iron (Fe) and oxygen (O) which is considered an inorganic compound. BFO displays multiferroic properties especially at room temperatures. In addition, BFO also shows high transition temperature (1103 K) and in particular, a single crystal of BFO displays high electric polarization (~ 100 μC/cm²) when compared to other ferroelectric materials. It is often a great challenge to produce a single phase BFO. Previously, the difficulties of preparing single phases of BFO has been reported elsewhere (Xie et al., 2014), where the characteristic single phase of BFO with ferroelectric Curie temperature (T_C) of 1103 K and Neel temperature (T_N) of 653 K has been reported (Xie et al., 2014). The special arrangement of R3c group in the crystal lattice of BFO, which has rhombohedrally distorted perovskite structure, allows spontaneous ferroelectric polarization that can be either any of the eight diagonal [111] directions as revealed in Figure 1.1. Usually BFO polarization comes from A-site which is mainly due to the lone pair of Bi ions (6s² orbital), in the same manner the magnetization result from Fe³⁺ at B-site.

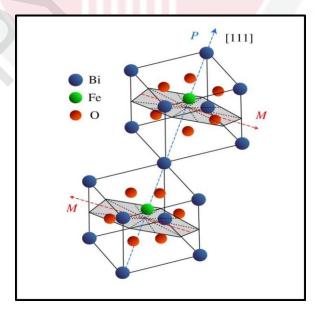


Figure 1.1: Representative image of the atomic structure of BiFeO₃ and the direction of the polarization along [111]. Source: Adapted from (Velev *et al.*, 2011).

1.5 Phase diagram of BiFeO₃

Figure 1.2 shows the phase diagram of bismuth oxide (Bi₂O₃)/iron oxide (Fe₂O₃) (Morozov *et al.*, 2003; Palai *et al.*, 2008). The preparation of bismuth ferrite (BiFeO₃) is always from the equal mixtures of raw materials i.e., Bi₂O₃ + Fe₂O₃ (1:1), at high temperatures, the mixture has the tendency to decompose back to its starting (raw) materials based on the following equation (1.1).

$$Bi_2O_3 + Fe_2O_3 \rightarrow 2BiFeO_3 \tag{1.1}$$

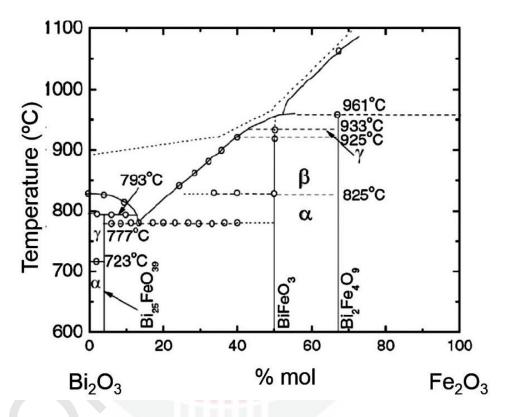


Figure 1.2: Phase diagram of BiFeO₃. Source: Adapted from (Catalan and Scott, 2009).

BFO is likely to show parasitic phases in which they nucleate together in the form of impurities at grain boundaries (Valant *et al.*, 2007). Previously, BFO has been reported to be truly metastable in atmospheric air, especially due to its optically visible impurity that are commonly found well below the melting point (Catalan and Scott, 2009). The remnant magnetization artificially improves due to the impurities and oxygen vacancies (Bea *et al.*, 2005; Lou *et al.*, 2007). When 200 kV/cm of electric field are applied, the BFO decomposes to produce a by-product Fe₃O₄ (magnetite) at room temperature (Leontsev and Eitel, 2009) as shown in the equation (1.2) below:

$$6BiFeO_3 \rightarrow 2Fe_3O_4 + 3Bi_2O_3 + O$$
 (1.2)

The phase Bi_2O_3 was possibly undetectable due to its known glass-forming compound, or because of its vaporizing capability within thermal decomposition. In addition, Bi_2O_3 reaches a melting point at temperature above 800 °C (Palai *et al.*, 2008).

1.6 Problem statement

There are some problems associated with BiFeO₃. The major problem in the preparation of bismuth ferrite (BiFeO₃) is the presence of impurities, non-perovskite phases such as Bi₂₅FeO₃₉ (sillenite) and Bi₂Fe₄O₉ (mullite) using solid-state reaction techniques (Muneeswaran *et al.*, 2014). These secondary phases (impurities) occur due to volatility and the dynamics phase formation. BiFeO₃ also has high leakage current because of oxygen vacancies and the oxidation status of Fe (ions) is fluctuation that appears (Fe²⁺, Fe³⁺) inside the perovskite matrix (Adhlakha *et al.*, 2013). In addition, BiFeO₃ also exhibits weak ferromagnetic (FM) properties due to the deformation of structure of spin spiral with G-type antiferromagnetism (Pradhan *et al.*, 2005; Godara *et al.*, 2015).

To overcome these difficulties, a modification has been employed, which is the substitution of B-site of BFO with rare-earth and transitional element, yttrium (Y³⁺) and Indium (In³⁺) using solid state reaction method. This is due to their larger ionic radius than iron, which can alter the crystal structure and improve their electrical and magnetic properties. Moreover, this will result in the reduction of high leakage current as well as remove the impurities. The following questions may arise for further research from the above-mentioned phenomenon:

- 1. At which percentage of substitution can one have an excellent reduction of leakage current?
- 2. At which specific phase formation of samples is a better substitution?
- 3. Will the influence of substitution increase/decrease the magnetic and electric properties of the materials?

1.7 Objectives of the study

General objective

The aim of this research is to study the influence of substitution of yttrium (Y^{3+}) and indium (In^{3+}) at the Fe-site of BiFeO₃ ceramics prepared via conventional solid-state reaction method. The effects of various concentrations on the structural, electrical and magnetic properties are investigated.

Specific objectives

The specific objectives are as follow

- 1. To synthesize $BiFe_{1-x}M_xO_3$ (M = Y^{3+} and In^{3+}) where (0.0, 0.1, 0.2, 0.3, 0.4, 0.5, 0.6, 0.8 and 1).
- 2. To study the phase formation of samples due to Y^{3+} and In^{3+} substitution.
- 3. To investigate the effect of substitution of Y^{3+} , with ionic radius of (1.04 Å) and In^{3+} , with ionic radius of (0.94 Å) on magnetic and electric properties of $BiFeO_3$.

1.8 Importance of the study

Multiferroics including BiFeO₃ are materials with high attractive interest subjects that have been deeply investigated in the field of material science research, due to their physical-chemical properties and numerous applications such as information storage devices, disk read/write heads, spin valves that are used in magnetic sensors and microelectronic devices. The material usually produces an enormous magnetoelectric coupling response above the room temperature. Moreover, it grants basic control of the electric polarization with a magnetic field or controlling charge by an electric field, which makes them valuable in the area of technological applications (Nan et al., 2008). These applications include spintronics, data storage, sensors and microelectronic devices (Arnold et al., 2010). The magnetoelectric (ME) influence is relatively significant for data storage applications that would allow magnetic information to be composed electrically and for the magnetic utility later (Smolenskiĭ and Chupis, 1982; Eerenstein et al., 2006; Bibes and Barthélémy, 2008). BiFeO₃ is a possible candidate for magnetoelectric and spintronic application. In spintroncs devices, information is written electrically and read magnetically (Chen et al., 2006). BiFeO₃ is found to be essential as a tunneling barrier layer, where the magnetic field can control the ferroelectric states, while the tunneling resistance controls the direction of polarization as well (Yin et al., 2015).

Given these evidences, multiferroics are currently being employed in several commercial applications, such as magnetic memory systems, sensor, spintronics (Béa *et al.*, 2005; Gajek *et al.*, 2005) and tunable microwave devices (Kadomtseva *et al.*, 2006): thus rendering the potential to revolutionize electromagnetic material's applications.

In this research study, the result obtained from yttrium and indium substituted BFO have added new contribution that can be adapted in many devices as mentioned above. Moreover, these new result can be considered new acknowledgments for researchers working in this field of research. In addition, the work conducted here using Y³⁺ and In³⁺ substitution on Fe-site of BFO has not been conducted by any other researcher and are not available in the literature.

1.9 Thesis outline

Synthesis and characterization of BiFe_{1-x} M_xO_3 (M = Y^{3+} , In³⁺) by conventional solidstate reaction is the main feature of evaluation in this research. Summary of multiferroic materials and bismuth ferrite in addition to the problem statement, importance and objectives of the study were presented in Chapter 1. A brief discussion on the general background of multiferroic materials, perovskite structure, magnetic and electrical properties of BiFeO₃, methods followed to synthesize BiFeO₃ and the effect of substitution on BiFeO₃ are presented in Chapter 2. Chapter 3 explains multiferroics, ferromagnetism, hysteresis loop, ferroelectrics, leakage current and dielectric properties. Information on the clarification of the procedures involved in the synthesis of $BiFe_{1-x}M_xO_3$ ceramics by solid-state synthesis technique were described in Chapter 4. The results and discussion of all the characterizations scanning electron microscopy (SEM), energy dispersive X-ray (EDX), X-ray diffraction (XRD), vibrating sample magnetometer (VSM), Impedance Analysis and Keithley source measure unit were explained in Chapter 5. Summary of allimportant results presented in the dissertation with recommendations for future research were presented in Chapter 6. Finally, Chapter 6 was followed by list of references, appendices, list of publications (research articles, conferences papers/posters by the author) and bio-data of the student.

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