

UNIVERSITI PUTRA MALAYSIA

EFFECT OF DWELL PERIOD ON FATIGUE LIFE OF 316L STAINLESS STEEL TUBE AT HIGH TEMPERATURE UNDER CREEP CONDITION

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By

KHAIRUL AZHAR BIN MOHAMMAD

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Abstract of thesis presented to the Senate of Universiti Putra Malaysia in fulfilment of the requirement for the degree of Doctor of Philosophy

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Chairman Faculty : Professor Mohd Sapuan Salit, PhD, P. Eng : Engineering

Components of engineering structures that operate at high temperatures, such as jet engines, pressure vessels, nuclear reactors, oil and gas plants, and steam and gas turbines, are subjected to significant thermal and mechanical loadings. As the external surfaces of structures always maintain contact with the environment and are exposed to weather, surface cracks may form due to imperfections during product fabrication. The main purpose of this study was to characterize and examine the relative importance of the mechanism of fatigue damage in tubular structures, which occur at high temperatures, and to develop an adaption equation of creep-fatigue life prediction based on failure mechanisms at high temperatures.

In this study, a 316L stainless steel grade tubular structure was employed to characterize the prediction fatigue lifetime of plant industry components at operating temperatures in the range of 500-650°C and in creep-fatigue loading conditions. Fatigue tests were performed in the finite life region of a rounded specimen with a constant load amplitude, a constant frequency of 5 Hz and the stress ratio R of 0.1 at room temperature. A constant load and high temperature of 565°C were imposed on the specimen during the creep test. The nature of a hold period (tensile or compressive) affects fatigue life and surface crack patterns. The creep-fatigue test, which is similar to the fatigue test, with five hold times at maximum tensile stress, was conducted using a rounded specimen of "Type 316L stainless steel" at 565°C. Optical and scanning electron microscopy was performed to characterize the metallurgical damage and explain the microscopic damage mechanics.

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Fatigue tests with and without hold periods were performed to assess the influence of creep-fatigue interaction on fatigue life. The results of the tests indicated that creep and temperature significantly impact fatigue behaviour. In several cases, fatigue lives were significantly reduced with an applied

hold time at high temperatures. Hold times are most damaging at high stress ranges and low fatigue lives. Many parameters affect the fatigue performance of structural components. Fatigue life is influenced by a variety of factors, such as the geometries and properties of specimens, stress, temperature, surface finish, direction of loading, presence of oxidizing or inert chemicals. Fatigue with a hold time and fatigue without a hold time at the fatigue limit were determined to be 39.2 MPa and 87.8 MPa, respectively, in this research.

The fatigue life of the 316L stainless steel was estimated by approaching the mean stress using a continuum damage mechanism (CDM). The continuum damage mechanism provides a reasonable prediction of fatigue response for high conditions. Based on the observation and characterization of fatigue life tubular steel pipes, the adaption equation for creep fatigue life prediction was proposed. Its simplicity gives it credibility and the adjustable use of alloy metal, which incorporates a number of cycles, applied stress and temperature, enables a power plant to predict the fatigue life of engineering components. The adaption equation can facilitate the damage tolerance design, which is the best design approach for reducing the cost and weight of heavy applications in the manufacturing process. A damage tolerant design philosophy for creep has been previously developed to improve the creep properties and elevated temperature fatigue crack growth resistance without sacrificing tensile strength and Low Cycle Fatigue (LCF) crack initiation life relative to the conventional microstructure (CM). Understanding the fatigue behaviour of steel pipes, which are derived from this study, are also useful in the design stage.

To validate the experimental results, the fatigue life prediction using finite element analysis (FEA) via Abaqus was employed. The simulation was performed by applying different stress levels to predict the stress of operation to measured life at the measured operation stress. The focus of the simulation is the importance of characterizing the fatigue limit using a comparison with experimental data. The fatigue limits for the simulation and the experiment are 150 MPa and 161 MPa, respectively, which correspond in terms of accuracy prediction; various aspects should be considered in the simulation. Additional developments in the analysis of creep-fatigue prediction test data are discussed, and expression for estimating fatigue life at high temperatures in stress-controlled tests are derived. To check the validity of adaption model proposed in this study, the assessment for creep fatigue interaction tests of austenitic 316L stainless steel under stress control at 565°C has been conducted. The life prediction results are within a factor of $\pm \times 1.5$, which indicates that the model is suitable for stress-controlled fatigue or creep fatigue life prediction of ductile material. The predicted results correspond with the experimental data.

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Abstrak tesis yang dikemukakan kepada Senat Universiti Putra Malaysia sebagai memenuhi keperluan untuk ijazah Doktor Falsafah

KESAN TEMPOH MASA PADA JANGKA HAYAT KELESUAN TIUB 316L KELULI TAHAN KARAT DI SUHU TINGGI DALAM KEADAAN RAYAPAN

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Komponen struktur kejuruteraan yang beroperasi pada suhu tinggi seperti enjin jet, kontena/ bejana tekanan, reaktor nuklear , loji minyak dan gas, dan gas turbin dan wap adalah tertakluk kepada muatan haba dan bebanan mekanik yang tinggi. Sejak permukaan luaran struktur sentiasa bersentuhan dengan persekitaran dan terdedah kepada cuaca dan terkena keretakan dan juga kerana ketidaksempurnaan semasa pembuatan produk, retak permukaan mungkin wujud. Tujuan utama kajian ini adalah untuk mencirikan dan memeriksa kepentingan relatif mekanisme kerosakan keletihan dalam struktur silinder berlaku pada suhu yang tinggi dalam usaha untuk membangunkan satu formula adaptasi rayapan-keletihan ramalan kehidupan berdasarkan mekanisme kegagalan pada suhu tinggi.

Dalam kajian ini masa ini, struktur berbentuk tiub 316L keluli tahan karat gred digunakan untuk mencirikan meramalkan jangka hayat kelesuan terutamanya bagi komponen industri loji dengan suhu operasi dalam lingkungan 500-650°C menjalani bebanan rayapan-kelesuan. Ujian kelesuan telah dijalankan dengan amplitud malar beban, kekerapan malar 5 Hz dan nisbah tekanan, R bersamaan dengan 0.1 pada suhu bilik untuk spesimen berbentuk tiub di kawasan hayat komponen. Sementara itu beban tetap dan suhu tinggi sebanyak 565°C akan dikenakan to spesimen untuk ujian rayapan. Sifat tahan lama (tegangan atau mampatan) mempengaruhi hayat kelesuan dan corak permukaan retak. Ujian rayapan-kelesuan adalah sama dengan ujian kelesuan dengan masa lima minit pada tegasan tegangan maksimum telah dijalankan menggunakan spesimen berbentuk tiub "Jenis 316L keluli tahan karat" pada suhu 565°C. Ujian optik dan imbasan mikroskop elektron akan dijalankan untuk mencirikan kerosakan metalurgi dalam memahami mekanik kerosakan mikroskopik.

Ujian kelesuan dengan dan tanpa tempoh pegangan telah dijalankan untuk menilai pengaruh interaksi rayap -kelesuan pada hayat lesu. Ia telah mendapati bahawa rayapan dan suhu memberi kesan yang utama ke atas tingkah laku kelesuan. Dalam beberapa kes, hayat kelesuan menyusut terutamanya dengan masa tegasan pada suhu yang tinggi. Ia telah mendapati

bahawa masa tegasan adalah paling merosakkan pada tekanan tinggi dan hayat kelesuan berkurang. Banyak parameter mempengaruhi prestasi kelesuan komponen struktur. Hayat kelesuan dipengaruhi oleh pelbagai faktor, seperti geometri dan sifat-sifat spesimen, tekanan, suhu, kemasan permukaan, arah bebanan, kehadiran pengoksidaan atau lengai bahan kimia dan lain-lain. Kelesuan dengan dan tanpa masa tegasan untuk daya ketahanan kelesuan didapati sebanyak 39.2 MPa dan 87.8 MPa masingmasing dalam kajian ini.

Hayat Kelesuan daripada keluli tahan karat 316L dianggarkan dari penghampiran tekanan separa menggunakan Continuum Damage Mechanism (CDM). Continuum Damage Mechanism menyediakan ramalan yang respon munasabah kelesuan untuk keadaan yang tinggi. Berdasarkan pemerhatian dan pencirian hayat lesu paip keluli berbentuk tiub, formula adaptasi rayapan kelesuan ramalan hayat telah dicadangkan. Ia adalah kesederhanaan memberikan ia kredibiliti, penggunaan laras untuk logam aloi yang menggabungkan bilangan kitaran, tekanan di kenakan dan suhu dan boleh membenarkan loji kuasa untuk menerima / adaptasi pakai dan hanya menggunakannya untuk meramal hayat lesu komponen kejuruteraan. Formula adaptasi boleh memudahkan reka bentuk toleransi kerosakan, yang mana merupakan pendekatan reka bentuk terbaik untuk kekal kos dan mengukuhkan simpanan untuk aplikasi berat dalam proses pembuatan. Falsafah reka bentuk toleransi kerosakan untuk rayap sebelum ini dibangunkan dengan tujuan untuk memperbaiki sifat-sifat rayapan dan rintangan kelesuan pertumbuhan retak suhu tinggi tanpa mengorbankan kekuatan tegangan dan LCF retak permulaan hidup berbanding dengan mikrostruktur konvensional (CM). Pemahaman dan pengetahuan tentang tingkah laku kelesuan paip keluli yang diperolehi daripada kajian ini juga berguna dalam peringkat reka bentuk.

Untuk keputusan pengesahan eksperimen, ramalan hayat lesu menggunakan Analisis Unsur Terhingga (FEA) melalui Abagus telah digunakan. Simulasi dilakukan dengan tahap tekanan yang berbeza untuk meramalkan tekanan operasi untuk mengukur hayat di tekanan operasi. Penekanan simulasi memberi tumpuan kepada kepentingan mencirikan had kelesuan berbanding dengan data uji kaji. Perbandingan had kelesuan antara simulasi dan eksperimen adalah 150 MPa dan 161 MPa, masing-masing yang akan menyediakan pertalian baik dari segi ketepatan ramalan walaupun pelbagai aspek perlu diambil kira dalam perkembangan simulasi. Pembangunan selanjutnya dalam analisis data ujian ramalan rayap-kelesuan dibincangkan dan ungkapan yang diterbitkan untuk menganggarkan hayat lesu pada suhu yang tinggi dalam ujian tekanan terkawal. Dalam usaha untuk memeriksa kesahihan model adaptasi yang dicadangkan dalam kajian ini, penilaian bagi ujian interaksi kelesuan rayapan 316L keluli tahan karat di bawah kawalan tekanan pada 565°C telah dijalankan. Hasil ramalan hayat berada dalam factor $\pm \times 1.5$ dan model dapat di kaitkan untuk tekanan terkawal kelesuan atau kelesuan rayapan ramalan hayat bahan mulur. Keputusan meramalkan berada dalam pertalian yang baik dengan data uji kaji.

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LIST OF ABBREVIATIONS

AGR	Advanced gas cooled reactor			
	American Iron and Steel Institute			
ASME	American Society of Mechanical Engineers			
ASTM	American Society for Testing and Materials			
Ω	Atomic volume			
Au	Gold			
С	Carbon			
°C	Celcius			
CDM	Continuum Damage Mechanisms			
CM	Conventional Microstructure			
CI	Chlorine			
Co	Cobalt			
Cr	Chromium			
CTOD	Crack Tip Opening Displacement			
DLDR	Double Linear Damage Rule			
EDX	Energy Dispersive X-ray			
Fe	Ferum			
FEA	Finite element analysis			
F	Fluorine			
FS	Frequency Separation Technique			
FSW	Friction Stir Welding			
HCF	High Cycle Fatigue			
HTHP	High Temperature High Pressure			
HTLCF	High Temperature Low Cycle Fatigue			
HCL	Hydrogen chloride			
kN	Kilo Newton			
LDR	Linear Damage Rule			
LEFM	Linear Elastic Fracture Mechanics			
LCF	Low Cycle Fatigue			
Mn	Manganese			
MPa	Mega Pascal			
IM	Melting Temperature			
Mo	Molybdenum			
Na	Natrium			
N	Newton			
HNO ₃				
0 DOD	Oxygen Demistant Olin Dand			
PSB	Persistent Slip Band			
SEM	Scanning Electron Microscopy			
51	Silicon			
SEFS	Strain Energy Frequency Modified Approach			
	Stress Intensity Factor			
5-N	Stress Life			
	Sulphul Tanaga Nasional Parkad Bassarah			
	renaya Nasional Demau Research Weight Dereentege			
WL%	Vergin Percentage			
αĭ	T LLEI DIUTTI			

 \bigcirc

YS Yield Strength



LIST OF SYMBOLS

Т	Absolute Temperature		
ΔF	Activation Energy		
Q_g	Activation Energy Of Grain Boundary Diffusion		
D	An Allowable Damage Value		
σ	Applied Stress		
R	Bolztmann's Constant		
b	Burgers Vector's Magnitude Of The Dislocation		
Ι	Cavity Spacing		
Р	Coefficient Of The Second Stage Of The Fatigue Life		
$\Delta \varepsilon_{_{pc}}$	Compressive Creep Strain		
$\Delta \varepsilon_{cc}$	Compressive Creep Strain		
$\Delta arepsilon_{pp}$	Compressive Plastic Strain		
$\Delta arepsilon_{cp}$	Compressive Plastic Strain		
α	Constant Of Order Unity		
t _r	Creep Rupture Time At The Same Stress Level In Static Creep Tests		
<i>E</i> _{cr}	Creep Strain		
$1/n_{c}$	Creep Strain Hardening Exponent		
N _I	Damage Caused By Crack Initiation,		
N _{II}	Damage Caused By Crack Propagation.		
E _{mod}	Elastic Modulus		

G

- \mathcal{E}_{el} Elastic Strain,
- $\Delta \mathcal{E}_{e}$ Elastic Strain Range
- σ_{e} Endurance Limit
- $\dot{\mathcal{E}}_{cr}$ Equivalent Creep Strain Rate
- σ_{v} Equivalent Saturation Creep Stress.
- *c* Fatigue Ductility Exponent
- *b* Fatigue Strength Exponent
- Fatigue Ductility Coefficient \mathcal{E}_t
- Fatigue Strength Coefficient σ_f
- Flow Stress At 0 K
- v Frequency
- *GPa* Giga Pascal
- *D_g* Grain Boundary Diffusivity
- $\sigma(t)$ Instantaneous Tensile Relaxation Stress During Hold Time
- F_{max} Maximum Strength
- M₂₃C₆ Metal Carbide
 - *N*_f Number Of Cycles To Failure
- N_f^{Hold} Number Of Cycles To Failure At A Given Strain Range For Fatigue Tests With Hold Periods
- N_{f}^{NoHold} Number Of Cycles To Failure At The Same Strain Range For Continuous Fatigue Tests
 - Plastic Strain

 ${\cal E}_{pl}$

	· E	Secondary Creep Strain Rate,
	μ	Shear Strength
	$\varepsilon - N$	Strain Life
	E	Strain Rate
	$\sigma_{_a}$	Stress Amplitude
	A_{o}	Stress Co-Efficient,
	п	Stress Exponent
	$\sigma_{\scriptscriptstyle FL}$	Stress of Fatigue Limit
	$\Delta arepsilon_p$	Sum of a Plastic Strain Range
	Т	Temperature
	Ta	Absolute Temperature
	Q	Thermal Activation Energy Of Creep
	δ	Thickness Of A Grain Boundary
	t	Time
	т	Time Exponent
	\mathcal{E}_{tot}	Total Strain
	t	Total Time In Fatigue Hold-Time Tests At A Given Stress Level
	$\Delta \varepsilon_t$	Total Width Of The Loop
	${\cal E}_f$	True Fracture Strain,
	$\pmb{\sigma}_{\scriptscriptstyle f}$	True Fracture Strength,

- $\Delta \mathcal{E}_t \qquad \text{Total Range Of Strain}$
- $\Delta\sigma$ Total Stress Range
- F_{yield}, Yield Strength
 - E Young's Modulus



CHAPTER 1

INTRODUCTION

1.1 Scenario of Fatigue Life in Heavy Industries at High Temperatures

Creep-fatigue life prediction is a complex and essential criterion in the design of components that operate at high temperatures and are subjected to alternate loadings. The mechanical properties of most materials are dependent on the temperature. The ultimate tensile strength, yield strength and modulus of elasticity usually decrease with increasing temperature. The effect of high temperature on mechanical properties is associated with transformations of the material structure due to diffusion processes, ageing, dislocation restructuring (softening), and recrystallization. These processes imply that plastic deformation can easily occur at an elevated temperature. For high temperature applications in the melting point temperature range of 1400-1500°C, with the introduction of hold time, creep deformation can occur if favourable stress and temperature combinations are maintained. At elevated temperatures that span one-third to two-thirds of the melting point and at low-imposed stresses, the majority of metals, alloys and ceramics exhibit creep deformation grain boundary cavitations. In many engineering alloys, creep cavities nucleate below the nominal stress level of 100 MPa. Creep deformation with the introduction of hold time can occur if favourable stress and temperature combinations are maintained for high-temperature applications in the melting point temperature range of 1400-1500°C.

At high temperatures, the majority of metals show damage in the form of grain boundary voids and wedge cracks. A grain boundary cavity may nucleate as a result of either sliding or slip impingement and is sustained by either a stressassisted diffusion, additional grain boundary sliding or a combination of these factors (Sandhya et al., 2005). Creep damage, which is a time-dependent process, is primarily dependent on the history of the stress and temperature that was applied to the component, whereas fatigue damage is generated by the cyclic stress and is primarily dependent on time-independent plastic strain (Gaoa et al., 2005).

A few decades ago, the prevailing viewpoint was that brittle material did not undergo fatigue (as brittle materials have limited dislocation motion); however, brittle materials do exhibit both mechanical fatigue and thermal fatigue under repetitive loading (Bhowmik et al. 2007; Wachyman et al. 2009).

Engineering components that are employed in oil, gas, aviation and nuclear industries usually involve high-temperature environments; in these conditions. As creep-fatigue is a significant failure mechanism, the life of these components should be adequately predicted before they are fabricated and implemented. Therefore, existing damage tolerant life prediction methods need to incorporate the creep phenomenon in high-temperature applications.

1.2 Issues Concerning of High-Steel Industries

Many researchers have performed studies on the type of steel that is applied for the global transmission of oil and gas. Defects that occur over the lifetime of a pipeline are problematic to engineers, researchers and manufacturers. Similar to any engineering structure, pipelines occasionally fail. In numerous cases, the only significant load in a pipeline's breakdown is the internal pressure (Cosham and Hopkins, 2004). Circumstances in which additional loads are possible include high temperature high pressure (HTHP), environmental damage (corrosion fatigue), elevated temperatures (creep fatigue) or sliding/physical contact (fretting and rolling contact fatigue) (Ritchie, 1999).

Different types of failure scenarios may occur on a pipeline, in which the failure of fittings (flanges and valves) are not considered.

- i. failure of a defect-free pipe,
- ii. failure of a pipe, which consists of a 'workmanship' imperfection (i.e., an imperfection in the pipe body or the possibility of a careless weld that is satisfactory to the relevant specifications or standards), and
- iii. failure of a pipe, which consists of a defect/flaw that is not tolerable at the workman-ship level (such as a crack or a dent on a weld) (Cosham et al., 2008).

In addition, the damage/blow of steel is also categorized by the interaction between a defect (which causes a reduction in the burst strength) and a defect in the pipe fitting (pipework, fitting, and elbows). Thus, a study of Type 316L stainless steel is conducted to identify the mechanism that is susceptible to failure under temperature and variable stress. Continuous research and development are required to understand the causes and cures of cracking in steel.

A general tendency towards severe operating conditions is observed, i.e., higher mechanical loadings and temperatures to increase the efficiency of gas and steam turbines, internal combustion engines, heat exchangers, conventional and nuclear electric power generation equipment and other engineering components and devices. This trend has caused the initiation, growth and interaction of complex damaging processes within the materials of these components. They can cause the failure of a component and an entire structure, which limits their lifetime. Therefore, the safe assessment of the lifetime of a component is important for the prevention of failures, which may have disastrous consequences. Conservative predictions, however, may unnecessarily increase the cost of production and maintenance of systems.

1.3 Significance of Service Lifespan Prediction

Service lifespan prediction techniques are important in any industry that involves the daily use of metal components to assess its service damage, which is caused by high pressure and elevated temperature creep. A precise life prediction methodology that focuses on metal life is essential to ensure that the hot section component can be rejuvenated and to minimize irreparable damage and replacement costs associated with steel components.

The main failure mode for tubular steel is known as high temperatures at low cycle fatigue as their utilization has experienced high global demand. Many researchers have focused on the service life prediction and extension of tubular steel, which is challenging due to the geometric shapes of specimens and the complexity of the phenomena. To determine and characterize the accuracy of a life prediction, both the upper bound and the lower bound are main aspects of engineering components at elevated temperatures. A material will be more unsafe and dangerous when it is overpredicted, whereas m will affect the inherent risk of damaged components or offset the economic benefits (Bernstein, 1982). No specific life prediction model has gained global acceptance among the majority of plant industries. Each industry performs a separate life prediction according to the situation and application. The foremost difficulty in the prediction on any material is accounting for the contributions by creep and/or environmental attack of the fatigue process.



Seamless Steel Tubes for Heat Exchanger and Condensers



UHT treatment of milk



stainless steel pipes



Food grade and sanitary grade applications



Finned Tubular Heater Construction Styles Finned Tubular Heaters with Copper, Stainless Steel and Incoloy Sheaths



1.4 Problem Statement

Life prediction is a complex and essential aspect of designing a component that is subject to high temperatures and alternative loading. Many researchers have focused on tolerate design to determine the life of a heavy-duty pipe that operates at high mechanical pressures and temperatures due to the repeated usage of cylindrical components and the extensive usage of this type of geometry. A main feature for industry is the provision of a high-strength material, which is challenging due to unknown compositions of industrial materials and the inevitable presence of defects in their substructures, such as holes, cavities and cracks. As a result, the consideration of fracture mechanics in the design of metallic structures is crucial. Both fatigue and creep can cause a structure to fail. Numerous efforts to understand why and how materials fail have been achieved. An increased demand for stainless steel products can be attributed to the increased use of stainless steel in the automotive industry (which uses stainless steel liners for auto exhaust systems), the maintenance an upgrading of oil refining and chemical plants, and the extensive use of stainless steel equipment in the pulp and paper industry and fast food service industry. The consumption of stainless steel that is used to fabricate kitchen utensils and appliances and the decorative use of coloured stainless steel on building facades by the construction industry has also increased.

The designer/analyst requires information about the stress concentration factor that is created by axial wall misalignment of adjacent tubular sections. This misalignment is usually caused by a change in wall thickness, oversized cross-sections and/or imprecise fabrication controls (Connelly and Nettlemoyer, 1993) Tubular components such as pressure vessels, pipes, borers, and driving shafts are common and useful parts of engineering structures. Due to an extensive range of usage for this type of structure, such as the transmission or storage of fluid produced in pipes and high-pressure vessels, these structures should be assessed under different conditions, such as various material, temperature and loading conditions. These conditions are components (Becker et al., 1986).

In the safety consideration of the mode of failure of pressure vessels and pipes, crack initiation, crack propagation and fracture, as well as fatigue failure, should be considered. Pipes that are subjected to high temperatures will have a shorter life than those with creep damage and fatigue damage that are separately incurred. In a safety assessment of pressure vessels and pipes, fatigue crack propagation and fracture should be considered among the possible modes of failure. The geometry of the component and the load conditions are the two most important parameters that affect the mechanisms of failure and crack propagation. The combination of temperature and loading fatigue causes more damage and life reduction in engineering components. Many researchers have noted that the application of fatigue and creep at high temperatures in the estimation of life and safety in a component is an essential part of design and application in heavy industries. The mechanism of the parameter in life prediction

for a tubular structure under different conditions, such as creep, fatigue or creepfatigue, is not well understood.

Therefore, this study investigated the fatigue life due to creep-fatigue and the estimation of life and verifies effective parameters and factors for tubular structures regarding certain parameters that can be published to improve the development of this modern arena.

1.5 Research Objectives

The objectives of this research are as follows:

- To characterize the fatigue life and endurance limit of tubular steel 316L stainless steel at high temperatures.
- To determine the mechanism of fracture, which is associated with creepfatigue that is induced by temperature and hold time.
- To investigate the metallographic microstructural characterization of the metal in terms of the microstructure's fractured surface based on the results of fatigue and creep fatigue tests.
- To suggest the simple constitutive models of creep-fatigue interaction, which can be used to predict the lifetime for complicated situations of creep-fatigue, using simple creep and fatigue test data.

1.6 Thesis Layout

This thesis is divided into five chapters. Following this introductory chapter, which consists of the background of the research, chapter two provides a critical review of relevant literature based on an overview of fatigue, creep, fracture mechanics and creep-fatigue. A review of fatigue studies that have explored topics such as the historical background and creep fatigue life prediction models, is also presented in chapter two. Chapter three outlines the underlying theory and the experimental techniques that are employed in this study. Chapter 3 describes the experimental work, including the utilized materials, the testing techniques and the equipment. Chapter four is a placeholder to present the results of the current study and discuss the correlation between the obtained results and the existing theory. Observations of crack initiation sites and details of the transition from Stage I to Stage II and the transition from Stage II to Stage III using scanning electron microscopy are provided. The microstructural and chemical composition study that employs SEM are discussed. The test results are analysed in Chapter 5, which also presents the conclusion and recommendations for future studies.

1.7 Scope of Study

The scope of this study focuses on stainless steel grade 316L, which is extensively used in the petrochemical and construction industries and for power generation in pressure vessels, boilers, and steam turbines; its application in the oil and gas industries, such as pipe lines for carrying natural gas and sour gas, was not addressed in this study. The specimen preparation and fatigue tests were conducted in accordance to ASTM E606-92 and ASTM E466-96, respectively. The fatigue tests were performed under constant amplitude cyclic stress fluctuation loadings. A push-pull fatigue test was conducted to develop a stress-life curve for the hourglass of stainless steel specimens. In the creep test, which was conducted in accordance with ASTM E-139, specimens were subjected to high-temperature conditions and constant stress. The environment design was created to mimic the real scenario of actual piping in plant industries; thus, a combination of fatigue and creep tests have been performed to examine the mechanism of creep fatigue interaction on steel subjected to stress and elevated temperature. A simulation using Abagus was performed to compare and validate the experimental data. In this study, a new contribution has been achieved by discovering the adaption of the equation from the continuum damage mechanism (CDM), which can be employed to predict the fatigue life of the engineering components of 316L stainless steel at high temperatures. Scanning electron microscopy (SEM) was employed to analyse the variations in the microstructure pattern and the fracture surface of the steel fatigue specimen and a macrostructure analysis was employed to identify the basic stages crack development using optical microscopy. The composition elements of the specimen fatigue of the structure were analysed by energy dispersive X-ray (EDX) analysis.

REFERENCES

- Ali, A., Brown M.W. and Rodopoulos C.A. (2008). Modelling of crack coalescence in 2024-T351 al alloy friction stir welded joints. *International Journal of Fatigue* 30(10-11): 2030-2043
- Anderson, T.L. (1995) Fracture Mechanics-Fundamental and application, 2nd ed. CRC Press, Boca Raton, Fla., pp. 88
- Anonymous, (1963), NPL Manual, Modelling Creep in Toughened Epoxy Adhesives, Implementation of the Creep Model in a Finite Element Analysis, Measurements for Materials System Programme on Design for Fatigue and Creep in Joined Systems of Mechanical Engineers, pp.3-47
- Anonymous, (1998). ASME B31.3 Process Piping, The American Society of Mechanical Engineers, 345 East 47 Street, New York, New York 10017.. Pp. 14-22.
- Anonymous, (2010c). Stainless Steel Tube and Pipe, Intergranular corrosion of stainless steel tubes. http://www.stainless-steel-tube.org/Intergranular-Corrosion-of-Stainless-Steel-Tubes.htm Retrieved 28 April 2011
- Anonymous, (2015) American Foundrymen's Society (AFS), Unites States Department Of Energy Welcome To Homepage of the Afs Inclusion Atlas, Inclusion Formation During Foundry Processing, Inclusion Formation And Removal, 2014 Accessed On 14 April 2015
- ASME, (1986). Boiler and Pressure Vessel Code, Code Cases, Nuclear Components N-47-24, Class 1 Components in Elevated Temperature Service, Section III, Div. I.
- ASME, (1986b). Boiler and pressure vessel code, Preliminary Materials Selection Issues for the Next Generation Nuclear Plant Reactor Pressure Vessel, Components in Elevated Temperature, Nuclear Engineering Division
- Baik, S. (1982). Mechanisms of creep fatigue interaction, *Phd Thesis*. Cornell University

Bannantine, J. A., Comer, J. J. and Handrock, J. L. (1990). *Fundamentals of Metal Fatigue Analysis,* Prentice-Hall, Inc.

Basquin, O.H. (1910). The exponential law of endurance tests. Proceeding America, Society for Testing Materials 10: 625-630

Becker, F.L., Walker, S.M., and Viswanathan, R. (1986). Guideline for Evaluation of Seam Welded Steam Pipe. EPRI Report, CS-4774

- Beden, S.M., Abdullah, S., Ariffin, A.K., Al-Asady, N.A. and Rahman, M.M. (2009). Fatigue life assessment of different steel-based shell materials under variable amplitude loading. *European Journal of Scientific Research* 29(1): 157-169
- Bennett, J. A. (1946). A study of the damaging effect of fatigue stressing on X4130 steel", Proceedings, American Society for Testing and Materials 46 : 693-714
- Berling J.T. and Conway, J.B. (1970). Effect of hold time on the low cycle fatigue resistance of 304 stainless steel at 1200°F, In : 1st international conference on pressure vessel technology, part II, New York , Asme : p. 1233-1246
- Bernstein, H. L. (1982). Low cycle fatigue and life prediction, ASTM STP 770,
 C. Amzallag, B.N. Leis and P. Rabbe, Eds., *American Society for Testing* and Materials, Philadelphia, PA, p. 105
- Bhowmick, S. Melendez-Martinez, J.J and Lawn, B.R. (2007). Bulk is silicon is susceptible to fatigue, Applied Physic Letter., 91, 201902
- Carter, P. (2005) Analysis of cyclic creep and rupture. Part 1: bounding theorems and cyclic reference stresses, International Journal of Pressure Vessels and Piping 82(1):15-26.
- Chandler, H.D. (1984), Relationships between cyclic creep and hysteresis loop energies in metalsActa Metallurgical 32: 1253–1257
- Chen, L. J., Yao, G., Tian J.F., Z.G. Wang and Zhao, H.Y. (1998). Fatigue and Creep Fatigue Behavior Of A Nickel-Base Superalloy At 850°c, International Journal Fatigue 20(7): 543-548
- Christensen, R.H. (1959). Fatigue Cracking, Fatigue Damage and Their Detection. Metal Fatigue, Sines, G. and Waisman, J.L. Eds., Mcgraw-Hill Book Co. New York :p. 376
- Claudio, R.A., Burgess, A., Branco, C.M. and Byrne, J. (2008). Failure analysis of scratch damaged shot peened simulated components at high temperature. *Engineering Failure Analysis* 16(4): 1208-1220.
- Coffin, L.F. Jr., Carden, A.E., Manson, S.S., Severud, L.K. and Greenstreet W.L. (1977). Time Dependent Fatigue of Structural Alloys: A General Assessment, Report ORNL-5073.Components N-47-24, Section III, Div.1.
- Coles, A., Hill, G.J., Dawson R.A.T., and Watson, S. J. (1967). The High Strain Fatigue Properties of Low Alloy Creep Resisting Steel, The Metals and Metallurgy Trust, Monograph and Report Series No 32, pp. 270-294.
- Commerce, J.B. (1945). The effect of overstress in fatigue on the endurance life of steel", Proc. American Society for Testing and Materials 45 : 532-541

- Connely L.M. and Zettlemoyer, N. (1993) Stress concentration at girth welds of tubulars with axial wall misalignment. In: *Proceedings of The International Conference Symposium on Tubular Structures, Nottingham,UK*: p. 309-322
- Cosham, A. and Hopkins, P. (2004). An overview of the pipeline defect assessment manual (PDAM). 4th International Pipeline Technology Conference, Oostende, Belgium, 9-13 May.
- Cosham, A., Hopkins, P., and Spiekhout, J. (2008). Quantifying the probability of failure during the precommissioninghydrotest. *Journal of Pipeline Engineering* 7: 21-38
- Darveaux, R. (1993). Crack Initiation and Growth in Surface Mount Solder Joints. In :*Proceedings of the 26th ISHM International Symposium on Microelectronics*, Baltimore, p. 86-97
- Darveaux, R. (1997). Solder joint fatigue life model. in: Design and Reliability of Solders and Solder Interconnections. Mahidhara, R. K., Frear, D. R., Sastry, S. M. L., Murty, K. L., Liaw, P. K., Winterbottom, W. (Hrsg.). The Minerals an Materials Society, S.: p. 231-218
- Dawson, R. A. T., Elder, W. J., Hill, G. T. and Price, A. T. (1967). High-strain Fatigue of Austenitic Steels. International Conference on Thermal and High Strain Fatigue, Metals and Metallurgy trust, London, England. 239-269
- Degallaix, S., Degallaix, G., and Foct, J. (1988). Influence of Nitrogen Solutes and Precipitates on Low Cycle fatigue of 316L Stainless steels, Low Cycle Fatigue, ASTM STP 942, H. D. Solomon, G. R. Halfard, L.R. Kaisand, andB. N. Leis, Eds., ASTM, Philadelphia, pp.798-811
- Dobes F. and Miller, J. (1976) Relation Between Minimum Creep Rate And Time And Time To Fracture, Meterial Science 10 : 382-384
- Dong, P., Zhang, J. and Rawls, G. (2003). Crack growth behavior in residual stress field in vessel type structures. *Pressure Vessel and Piping Conference*, July 20-24, 2003 Cleveland, OH (US): 1-19.
- Edmunds, H.G. and White, D.J. (1966). Observation of the effect of Creep Relaxation on high strain fatigue. *Journal of Mechanical Engineering Science* 8(3): 310-321
- Ellison, E. G. and Patterson, A. J. F. (1976). Creep fatigue interactions in a 1 CrMOV steel, Proceeding of the Institution of Mechanical Engineers, Vol. 190, pp. 321-350

- Ellyin, F. (1997) *Fatigue Damage, Crack Growth and Life Prediction*.: Chapman and Hall, London ETNU-CT92-0056 *Evaluation of the Effect of Neutron Irradiation on Mechanical*
- Eshelby, J.D., (1961). Elastic Inclusion and Inhomogeneties, in Progress in solid mechanics, 2 north Holland Amsterdam, pp 130-133
- Evans, A.G. and Rana, A. (1980). *High Temperature Failure Mechanisms In Ceramics,* Acta Mettalurgical 28: 129-141
- Evans, W. J., Jones, J. P., and Williams, S., (2005). The Interactions Between Fatigue, Creep and Environmental Damage in Ti 6246 and Udimet 720 Li, *International Journal of Fatigue* 27(10-12) : 1473-1484.
- Fan, Z. C., Chen, X. D., Chen, L., and Jiang, J.L. (2005). Fatigue-creep interaction behavior of 1.25Cr0.5Mo steel and condition free from creep invalidation analysis, *Multiscale Damage Related to Environment Assisted Cracking*, Fracture Mechanics and Applications: 171-176.
- Fan, Z. C., Chen, X. D., Chen, L., and Jiang, J.L. (2006). A CDM-Based Study of Fatigue–Creep Interaction Behavior, *International Journal of Pressure Vessels and Piping* 86 (9) : 628–632
- Fan, Z. C., Chen, X. D., Chen, L., and Jiang, J.L. (2007). Fatigue-creep behavior of 1.25CrMo steel at high temperature and its life prediction. International journal of Fatigue 29: 1174-1183
- Fatemi, A. and Yang, L. (1998). Cumulative fatigue damage and life prediction theories: a survey of the state of the art for homogeneous materials. *International Journal of Fatigue* 20(1): 9-34.
- Forsyth, P.J.E. (1969). The Physical Basis of Metal Fatigue. Structural Fatigue and Metals, Blackie & Son. American Elsevier Publishing Co., New York.
- Frost, H.J. and Ashby, M.F. (1982). Deformation-Mechanism Maps: The Plasticity and Creep of Metals and Ceramics, Oxford, UK: Pergamon Press.
- Fuchs, H.O. and Stephens, R.I. (1980). *Metal Fatigue in Engineering*, John Wiley & Sons Inc. United States of America. *Fusion Reactor Materials*, September 27- October 1 1993
- Gao, N., Brown, M.W., and Miller, K.J. (1995). Short coalescence and growth in 316 stainless, Fatigue Fracture Engineering Material Structure 18(12): 1423-1441.
- Gaoa, N., Brown, M.W., Miller, K. J., and Reed, P. A. S. (2005). An investigation of crack growth behaviour under creep-fatigue condition. *Materials Science and Engineering A* 410-411: 67-71

- Geiss, P. L. (2011). Handbook of Adhesion Technology, Creep Load Conditions: 875-902
- Ghonem, H. and Zheng, D. (1992). Frequency interactions in high temperature fatigue crack growth in superalloys. *Material Science engineering* 150: 151– 60
- Golwalkar, S.V. (1984). Creep-Fatigue-Environmental Interactions in Ni-Base Superalloys, Ph.D Thesis, University Microfilms International.
- Goswami T. (1995) Creep-fatigue life prediction-a ductility model. *High Temperature Materials and Processes* 14(2): 101-114.
- Goswami, T. (2004). Development of generic creep-fatigue life prediction models. *Materials and Design* 25: 277–288
- Goswami, T. and Hanninen, H. (2001A). H. Dwell effects on high temperature fatigue behavior Part I. *Materials and Design* 22: 199-215
- Goswami, T. and Hanninen, H. (2001B). Dwell effects on high temperature fatigue behavior Part II. Materials and Design 22: 217-236.
- Goswami, T., 1999, "Low Cycle Fatigue–Dwell Effects and Damage Mechanisms," *International Journal of Fatigue*, 21(1), pp. 55-76.
- Grover, P.S. and Saxena, A., 1999, "Modeling the effect of creep-fatigue interaction on crack growth," *Fatigue and Fracture of Engineering Materials and Structures*, 22(2): 111122.
- H. L. Bernstein: Low Cycle Fatigue and Life Prediction, ASTM STP 770, C. Amzallag, B.N. Leis and P. Rabbe, Eds., American Society for Testing and Materials, Philadelphia, PA, 1982, p105.
- Haigh, B.P., Report on alternating stress tests of a sample of mild steel received from the British Association Stress Committee. 1915. p. 163-170.
- Hales, R. (1980). A quantitative metallographic assessment of structural degradation of type 316 stainless steel during creep-fatigue. *Fatigue and Fracture of Engineering Materials and Structures* 3(4): 339-356
- Hales, R., (1980). A Quantitative Metallographic Assessment Of Structural Degradation Of Type 316 Stainless Steel During Creep-Fatigue, Fatigue and Fracture of Engineering Materials & Structures 3(4): p. 339-356.
- Hall, P. (1987). Creep and Stress Relaxation in Solder Joints of Surface-Mounted Chip Carriers. *Components, Hybrids, and Manufacturing Technology, IEEE Transactions* 10(4): 556-565.
- Hashmi, M.S.J. (2014). Comprehensive materials processing 13 Volumes Set, Amsterdam. Netherlands, ISBN 978-0-08-096532-1 USA published 26 -09-2014

- Hong, J.W., Nam, S.W. and Rie, K.T. (1985). A model for the life prediction in low cycle fatigue with hold time. *Journal of material science* 20(10): 3763-3779
- Hull, D. and D.E. Rimmer, (1959). The growth of grain-boundary voids under stress, *Philosophical Magazine* 4 : p. 673-687
- Huynh, J., Molent, L., and Barter, S. (2008). Experimentally derived crack growth models for different stress concentration factors. *International Journal of Fatigue* 30(10-11): 1766–1786.
- Jaske, C.E., Mindlin, H., and Perrin, J.S. (1973). Cyclic stress-strain behaviour of two alloys at high temperature, *Cyclic Stress Strain Behaviour Analysis, Experimental and Failure Prediction*, ASTM STP 519, American Society for Testing and Materials :p.13-27
- Julie, A., Bannantine, Jess, J. and James, L. (1990). Handbook, Fundamentals of metal fatigueanalysis Prentice Hall.
- Kachanov, L.M. (1986) Introduction to Continuum Damage Mechanics, Mechanics of elastic stability 10: 121-128
- Kain, R.M. (1987). Evaluation of crevice corrosion, *Corrosion*, ASM Metal Handbook, 9th Ed., America Society for Metals, USA 13: 303-310
- Kashyap, B.P. and G.S. Murty, *Experimental constitutive relations for the high temperature deformation of a Pb-Sn eutectic alloy*. Materials Science and Engineering, 1981. 50(2): p. 205-213.
- Kashyap, B.P. and Murty, G.S. (1981). Experimental constitutive relations for the high temperature deformation of a Pb-Sn eutectic alloy. *Materials Science and Engineering* 50(2): 205-213
- Kennedy, A. J. (1963). Processes of Creep and Fatigue in Metals, John Wiley & Sons Inc. New York. p. 480.
- Kim, D.W., Chang, J.H., Ryu, W.S. (2008). Evaluation of the creep-fatigue damage mechanism of type 316l and type 316ln stainless steel. *International Journal of Pressure Vessels and Piping* 85(6) : 378-384.
- Kodama, M., Morisawa, J., Asano, K., Shima. S. and Nakata, K. (1993). Stress corrosion cracking and intergranular corrosion of austenitic stainless Irradiated at 323K, Journal Nuclear Material 212-215 : 1509- 1514
- Kommers, J.B. (1945). The effect of overstress in fatigue on the endurance limit of steel, Proceedings of ASTM, Vol. 45, pp. 532-541.
- Krempl, E. and Walker, C.D. (1969). Effect of creep rupture ductility and hold time on the 1000F strain-fatigue behaviour of a 1Cr-1Mo-0.25V steel.

Fatigue at High Temperature, ASTM-STP 459, American Society for Testing and Materials, Philadelphia :p. 75-99

- Krempl, E. and Wundt, B.M. (1971) *Hold-time Effects in High-Temperature Low Cycle Fatigues,* ASTM STP 489, American Society for Testing and Materials,
- Lagerberg, G. and Egnell, L. (1970). Canning materials for fast reactor fuel rods Nuclear Engineering International 15(166): 203-207
- Latha, S., Methew, M.D., Parameswaran, P., Rao, K.B.S. and Mannan, S.L. (2008). Thermal creep properties of alloy D9 stainless steel and 316 stainless steel fuel clad tubes. *International Journal of Pressure Vessels* and Piping 85(12): 866-870.
- Lee, S.B. and Kim, J.Y. (1996). Creep-fatigue crack growth behavior of 304 stainless steel at 650°C. *Theoretical and Applied Fracture Mechanics* 24: 181-188 Lemaitre, J., and Desmorat, R., 2005, *Engineering Damage Mechanics:Ductile, Creep, Fatigue, and Brittle Failures*, Springer-Verlag, Berlin, German.
- Lenntech, B.V. (1998). Water Treatment Solutions from http://www.lenntech.com/stainless-steel-316l.htm Retrieved 5 December 2011
- Ling, J. (2000). The evolution of the ASME boiler and pressure vessel code. Journal of Pressure Vessel Technology 122(3): 242-246
- Liu, X., Kang, B., and Chang, K.M. (2003). The effect of hold-time on fatigue crack growth behaviors of Waspaloy alloy at elevated temperature. *Materials Science and Engineering* A 340: 8-14.
- Lloyd, G.J. (1979). Predominantly elastic crack growth under combined creepfatigue cycling. *Metal Science* 13(1): 39-47
- Lloyd, G.J. and Wareing, J. (1979). Stable and unstable fatigue crack propagation during high temperature creep fatigue in austenitic steels: The role of precipitation. *Journal of Engineering Materials and Technology*. Transaction of ASME, 101: 275-283.
- Lloyd, G.J. and Wareing, J. (1981). Life-prediction methods for combined creep– fatigue endurance metals technology, 8(1): 297-305
- Lu, S. K., Yi, X.H., Yua, L., Jiang, Y.L., and Wei, W.R. (2011). Comparison of the simulation and experimental fatigue crack behaviors in the aluminum alloy HS6061-T6. *Procedia Engineering* 12: 242–247
- Lua, Y.L., Chen, L.J., Wang, G.Y., Benson M.L., and Liaw, P.K. (2005). Hold time effects on low cycle fatigue behavior of haynes 230 superalloy at high temperatures. *Materials Science and Engineering A* 409(1-2): 282-291

- Maeng W.Y. and Kang, Y.H. (1999). Creep-fatigue and fatigue crack growth properties of 316LN stainless steel at high temperature, *Transactions of the 15thInternational Conference on Structural Mechanics in Reactor Technology (SMiRT-15)*, August 15-20, 1999, Seoul, Korea.
- Maiya P. S. and Majumdar, S. (1977). Elevated Temperature Low Cycle Fatigue Behavior of Different Heats of Type 304 Stainless Steel, *Metallurgical transaction* 8A 1651-1660.
- Mannan, S.L. and Valsan, M. (2006). High-temperature low cycle fatigue, creepfatigue and thermomechanical fatigue of steels and their welds. *International Journal of Mechanical Sciences* 48(2): 160-175
- Manson, S. (1953). Behavior of materials under conditions of thermal stress. NACA-TN-2933 Manson, S. S., and Halford, G. R., 2009, *Fatigue and Durability of Metals at High Temperatures*, ASM International, Materials Park, OH.
- Manson, S., Halford, G., and Hirschberg, M. (1971). Creep–fatigue analysis by strain-range partitioning. in :Design for Elevated Temperature Environment, Zamrik S. and Jetter, R. Editors., ASME: New York : p. 12-28.
- Manson, S.S. (1973). A challenge to Unify Treatment of High Temperature Fatigue, fatigue at elevated temperature, ASTM STP 520 : 744-775
- Manson, S.S. and Hirschberg, M.H. *Fatigue: An Interdisplinary Approach,* Syracuse University Press, Syracuse, N.Y., 1964, p.133
- Marriott, D.L. (1983a). Estimation of creep deformation in components from short-term proof test data. *Journal of Pressure Vessel Technology* 105(2): 179-184.
- Marriott, D.L. (1983b). Lower-bound assessment of creep buckling strength. Journal of Pressure Vessel Technology 105(3): 216-221
- Mclean, D. and Pineau, A. (1978). Grain-boundary sliding as a correlating concept for fatigue hold-times. *Metal Science* 12(7): 313-316
- Metallography, (2002). Faculty of Engineering. Department of Mechanical and Production Engineering, National University of Singapore. accessed http://www.me.nus.edu.sg. Retrieved 17 December 2013
- Milella, P. P. (2013) Morphological Aspects of Fatigue Crack Formation and Growth, fatigue and corrosion in metals, ISBN:978-88-470-2335-2, 844 p. 763
- Min, B. K. and Raj, R. A. (1979). Mechanism of intergranular fracture during high temperature fatigue. *Fatigue Mechanisms*. Proceeding of an ASTM-NBS-NSF Symposium, Kansas City, Mo., May 1978, J.T. Fong, Ed., ASTM STP 675, American Society For Testing and Materials, Philadelphia,: p. 569-591

- Min, B.K. and Raj, R. (1978). Hold time effects in high temperature. *Acta Metallurgica*26 : p. 1007-1022
- Monkman F. C. and Grant, N. J. (1956). An empirical relation between rupture life and minimum creep rate in creep rupture test. Proc. ASTM 56 : 593-620
- Mukai, M., Kawakami, T., Hiruta, Y., Takahashi, K., Kishimoto,K., Shibuya, T. (1998). In: Transactions-American Society of Mechanical Engineers Journal Of Electronic Packaging; American Society Mechanical Engineers 120(2): 207-212
- Niesłony, A., Kurek, A., Dsoki, C., and Kaufmann, H. (2012). A study of compatibility between two classical fatigue curve models based on some selected structural materials. *International Journal of Fatigue* 39: 88-94.
- Olsson, C.O.A. and Landolt D. (1993). *Cavities Under Creep-Fatigue Interaction.* Progress In Materials Science, 37(5): 239-245
- Olsson, C.O.A. and Landolt, D. (2003) Passive films on stainless steelschemistry, structure and growth, Electrochim. Acta 48 :1093–1104
- Oshkour, A. A. (2009). *Fatigue crack propagation in aluminium 6063 Tubes*, Master Thesis, Universiti Putra Malaysia,
- Ostergren, W. J., 1976, "Damage Function and Associated Failure Equations for Predicting Hold Time and Frequency effects in Elevated Temperature, Low Cycle Fatigue," *Journal of Testing and Evaluation*, 4(5), pp. 327-339.
- Pao, Y.-H., Badgley, S., Jih, E., Govila, R., Browning, J. (1993). Constitutive behavior and low cycle thermal fatigue of 97Sn3Cu solder joints. ASME *Journal of Electronic Packaging* 115: 147-152.
- Pawel, J. E., Alexander, D. J., Grossbeck, M. L., Longest, A. W., Rowcliffe, A. F., Lucas, G. E., Jitsukawa, S., Hishinuma, A., and Shiba, K., "Fracture Toughness of Candidate Materials for ITER First Wall, Blanket, and Shield Structures," in Fusion Reactor Materials Semiannual Progress Report for Period Ending September 30, 1993, DOE/ER-0313/15, p. 173
- Picker, C., Tavassoli, A.A. and Dietz, W. (1991). EFR structural materialshighlights and progress. *International Conference on Fast Reactors and Related Fuel Cycles*, October 28-November 1, 1991, Kyoto.
- Picker, C., Tavassoli. A.A. and Dietz, W. (1992). Survey about failures in EuropeanFBRs. *Europe-CEI Exchange Meeting about Sodium Technology*, December14-18 1992, Bensberg Siemens.
- Plumbridge, W.J., Dalski, M.E., Castle, P.J. (1980). High strain fatigue of a type 316 stainless steel. *Fatigue of Engineering Materials and Structures* 3: 177-188. *Properties of Internals of PWR*. February 1994. CEC Study Contract

- Plumbridge, W. J., Dean, M. S., and Miller, D. A. (1982). The importance of failure mode in fatigue-creep interactions. *Fatigue and Fracture of Engineering Materials and Structures* 5(1): 101-114.
- Raj, R. and Ashby, M.F. (1975). Intergranular fracture at elevated temperature. *Acta Metallurgica* 23: 653-66
- Raj, R., *Flow and Fracture at Elevated Temperatures* (1983), ASM: Metals Park, Ohio. pp. 215-249.
- Raman, S. G. S., and Radhakrishnan, V.M. (2002). On cyclic stress-strain behaviour and low cycle fatigue life. *Journal of Materials and Design* 23: 249-254
- Rasidi, N., Soehardjono, A. M.D., and Dewi, S. M. (2011) Performance of Steel Structures under Fatigue Cyclic, Loading Journal of Civil Engineering and Architecture 5(3): 265-272
- RCC-MR, (2002) Design and Construction Rules for Mechanical Components of FBR Nuclear Islands and High Temperature Applications, Appendix A16: Guide for Leak Before Break Analysis and Defect Assessment, AFCEN.
- Riedel, H. and Rice, J.R. (1980). Tensile cracks in creeping solids. Fracture Mechanics: Proceedding of the Thirteenth National Symposium on Fracture Mechanicms: A Symposium American Society for Testing and Materials. ASTM Special Technical Publication 743: 112-130.
- Ritchie, R.O. (1999) Mechanisms of Fatigue-Crack Propagation In Ductile And Brittle Solid, International Journal of Fracture 100; 55-83
- Robinson , E. L.(1952) Effect of Temperature Variation on The Long Time Rupture Strength On Steels, ASME Transactions, 74: 777-781
- Robinson, E. (1952). Effect of temperature variation on the long-time rupture strength of steels. *Transaction of ASME* 74(5): 777–81
- Rodriguez, P. and BhanuSankara R. K. (1993) *Nucleation and growth of cracks* and cavities under creep-fatigue interaction. Progress in Materials Science,. 37(5): p. 403-480.
- Roylance, D. (2008) Strength of Materials & Solid Mechanics,_Mechanical Properties of Materials Book In Chapter Chapter 5 Yield And Flow pp. 79-104
- Sadananda, K. and Shahinian, P. (1980). Effect of environment on crack growth behavior in austenitic stainless steels under creep and fatigue conditions. *Metallurgical and Materials Transactions A* 11(2): 267-276

- Sahm, P.R. and Speidel, M.O. (1973). High-temperature materials in gas turbines. *Proceedings of the Symposium on High-Temperature Materials in Gas Turbines*, New York, 12-13 March, 1974. pp. 207.
- Sandhya, R., Rao, K.B.S., and Mannan, S.L. (2005). Creep–fatigue interaction behaviour of a 15Cr–15Ni, Ti modified austenitic stainless steel as a function of Ti/C ratio and microstructure. *Materials Science and Engineering A* 392:326–334
- Schijve, J. (2004). Fatigue of Structures and Materials. Delft: Kluwer Academic Publishers. ISBN 0-7923-7014-7
- Schroeder, L.C. and Poirier, D.R. (1984). The mechanical properties of thermite welds in premium alloy rails. *Materials Science and Engineering* 63(1): 1-21.
- Shine, M.C. and Fox, L.R. (1988). Fatigue of solder joints in surface mount devices. Low cycle fatigue, ASTM special technical publication STP 942 : p. 588-610.
- Smith D. J., and Ellison E. G. (1992) Modelling Crack Growth for and Fatigue Loading International Journal Pressure Vessell and Piping 50 :231-241
- Stephens, R. I., Fatemi A., Stephens, R. R., and Fuchs, H. O. (2001). Metal Fatigue in Engineering, 2nd ed. John Wiley & Sons Inc. United States of America Structures. Journal of Nuclear Materials 212-215: 442-447

Suresh, S. (1991) Fatigue of Materials. Cambridge: Cambridge University Press.

- Tavassoli, A.A., Ould, P., Little, E.A., Vries, M. de, and Dietz, W. (1994). Evaluation of the Effect of Neutron Irradiation on Mechanical Properties of Internals of PWR. February 1994. CEC Study Contract ETNU-CT92-0056
- Tavassoli, A.A., Ould, P., Little. E.A., Vries, M. de. and Dietz, W. (1994). *Temperature Service*, Subsection NH-Class 1, Code Cases, Nuclear
- Terlain, A. (1993). EC Structural Materials Meeting, March 10-11, 1993, Garching.

Tavassoli, A.A. (1995). Assessment of austenitic stainless steels. *Fusion Engineering and Design* 29: 371-390

- Tong, J., Dalby, S., Byrne, J., Henderson, M.B., Hardy, M.C. (2001). Creep, fatigue and oxidation in crack growth in advanced nickel base superalloys. *International Journal of Fatigue*. 23(10): 897-902
- Tribula, D. and Morris, J.W. Jr., (1989). Creep in shear of experimental solder joints. Journal Electron. Package 112(2): 87-93

- Velay, V., Bernhart, G., and Penazzi, L. (2009). Thermal Fatigue of a Tool Steel: Experiment and Numerical Simulation. *Journal of Fatigue* 38: 793-814
- Venkatesh, V. and Rack, H. J. (1999). A Neural Network Approach to Elevated Temperature Creep-Fatigue Life Prediction," *International Journal of Fatigue*, 21(3): 225234.
- Venkateswaran, A., and Hasselman D. P. H. (1981) Elastic Creep Of Stressed Solids Due To Time-Dependent Changes In Elastic Properties, Journal Of Materials Science 16 :1627-1632
- Vogt, J.-B., Foct, J., Regnard, C., Robert, G. and Dhers, J. (1991). Lowtemperature fatigue of 316L and 316LN austenitic stainless steel, *Metallurgical Transactions A* 22(10): 2385-2392.
- Vries, M. de and Tavassoli, A.A. (1993). Irradiation Induced Degradation of the Fracture Toughness of Austenitic Stainless Steel Plate and Weld Materials at 700K, *ICFRM-6 on Fusion Reactor Materials*, September 27-October 1, 1993, Stresa, Italy.
- Wachtman, J.B, Cannon, W.R. and Matthewson, M. J. (2009). Mechanical Properties of Ceramics, 2nd Edition Wiley, Hoboken, New Jersey,
- Walker, C.D., "Strain-Fatigue Properties of Some steels at 950°F (510°C) with a Hold in the Tension Part of the Cycle", in *Joint International Conference on Creep, 1963, Papers,* The Institution of Mechanical Engineers, 1963, pp.3-47
- Wang, J. (1992). Low cycle fatigue and cycle dependent creep with continuum mechanics. *International Journal of Damage Mechanics* 1(2): 237-244
- Wang, J., Qian, Z., Zou, D., and Liu, S. (1998). Creep behavior of a flip-chip package by both FEM modeling and real time moire interferometry. *Journal Electron. Package* 120(2): 179-185
- Wareing, J. (1977). Creep-fatigue interaction in austenitic stainless steel. Metallurgical and Materials Transaction A 8(5): 711-721
- Wei, R.P. and Huang, Z. (2002). Influence of dwell time on fatigue crack growth in nickel-base superalloys. *Materials Science and Engineering A* 336: 209–214.
- Yaguchi, M., Nakamura, T., Ishikawa, A., and Ashada Y., (1996). Creep fatigue damage assessment by sequent fatigue starining, nuclear engieenring and design 162 :97-196
- Yao J. T. P. And Munse W. H. (1961) Low-Cycle Fatigue Of Metals–Literature Ship Structure Committee, National Academy of Sciences-National Research Council

- Yao, W.X. (1993). Field Intensity Approach for Predicting Fatigue Life. International Journal of Fatigue 15(3), 243-246.
- PYung, L.L., Pan, J., Hathaway, R.B., and Barkey, M.E. (2005). Fatigue Testing and Analysis. Theory and Practice. Elsevier Butterworth–Heinemann, United States of America.
- Zamrik, S.Y. (1993). Damage models for creep-fatigue interaction", in Technology for '90s - A Decade of Progress (Ed: Au-Yang M. K.), The ASME Pressure and Vessels and Piping Division, ASME, New York, pp. 84-107.
- Zhichoa, F., Chen, X., Chen, L., and Jiang, J. (2007). Fatigue-creep behavior of 1.25Cr0.5Mo steel at high temperatures and it life prediction *International Journal of Fatigue* 29 : 1174-1183